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1 **Synthesis and NO_x removal performance of anatase S-TiO₂/g-CN heterojunction formed from**
2 **dye wastewater sludge**

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18 **Abstract**

19 In this study, sludges generated from Ti-based flocculation of dye wastewater were used to retrieve
20 photoactive titania (S-TiO₂). It was heterojunctioned with graphitic carbon nitride (g-CN) to augment
21 photoactivity under UV/visible light irradiance. Later the as-prepared samples were utilized to remove
22 nitrogen oxides (NO_x) in the atmospheric condition through photocatalysis. Heterojunction between S-
23 TiO₂ and g-CN was prepared through facile calcination (@550 °C) of S-TiO₂ and melamine mix.
24 Advanced sample characterization was carried out and documented extensively. Successful
25 heterojunction was confirmed from the assessment of morphological and optical attributes of the
26 samples. Finally, the prepared samples' level of photoactivity was assessed through photooxidation of
27 NO_x under both UV and visible light irradiance. Enhanced photoactivity was observed in the prepared

28 samples irrespective of the light types. After one h of UV/visible light-based photooxidation, the best
29 sample STC4 was found to remove 15.18% and 9.16% of atmospheric NO, respectively. In STC4, the
30 mixing ratio of S-TiO₂, to melamine was maintained as 1:3. Moreover, the optical bandgap of STC4
31 was found as 2.65 eV, where for S-TiO₂, it was 2.83 eV. Hence, the restrained rate of photogenerated
32 charge recombination and tailored energy bandgap of the as-prepared samples were the primary factors
33 for enhancing photoactivity.

34 **Keywords**

35 Resource recovery, TiO₂/g-CN, NO_x photooxidation, Nitrate selectivity, Heterogenous photocatalysis

36 **1 Introduction**

37 In conventional wastewater treatment plants, coagulation and flocculation have been employed for
38 generations for the removal of particulate matters either in suspension or colloidal form. In general,
39 settling basin and mechanical filtration can easily separate the pollutants in suspension but found
40 ineffective in removing colloidal particles (Zhao and Li, 2019). Meanwhile, coagulation and
41 flocculation can readily remove various water-induced contaminants of the colloidal genre, e.g.,
42 materials causing turbidity, metal oxides, persistent toxic compounds, organic materials, stable
43 emulsions, etc. (Xia et al., 2018; Zhang et al., 2018b; Hussain et al., 2019). Traditionally, various types
44 of Al and Fe salts are used as coagulants; however, they are induced with the significant drawbacks of
45 unwanted sludge generation and effects of trace metals (Al/Fe) in subsequent effluent (Xu et al., 2018b).
46 Recently, Ti-based salts (TiCl₄, TiSO₄, and PTC) are found to perform on par with the Al and Fe based
47 coagulants (Wang et al., 2018a; Wang et al., 2018c; Chi et al., 2019). Ti (IV) coagulation has been
48 effectively applied to treat water and wastewater enriched with various types of pollutants, such as
49 inorganic colloids (Wang et al., 2018c), natural organic matter with varying molecular weight (Wan et
50 al., 2019), algae (Xu et al., 2018b; Chi et al., 2019), trace nanoparticles (Aziz et al., 2018), phosphorous
51 (Jeon and Ahn, 2018), arsenic (Sun et al., 2013) and many more. Additionally, Shon et al.
52 (2009b) affirmed equivalent pollutant removal comparing conventional coagulants (FeCl₃ & Al₂(SO₄)₃)
53 and steady titania recovery from the flocculated sludge (Shon et al., 2007). Photo-catalytically active

54 anatase TiO₂ were prepared through incineration of TiCl₄ flocculated sludge at a temperature of 600 °C
55 to 800 °C. Shortly after, Lee et al. (2008), studied the aquatic toxicity of residual Ti after TiCl₄
56 coagulation. They observed negligible toxicity at a chemical dose of 150 mg/L. Furthermore, the ready
57 titania from flocculated sludge showed low toxicity compared to commercially available TiO₂ (P-25).
58 Hence Ti-based coagulation possesses the potential to mitigate the human health and high chroma index
59 issues of residual Al and Fe after treatment (Lee et al., 2008; Shon et al., 2008; Okour et al., 2009b). To
60 date, titania nanoparticles (NPs) and nanofibers (NFs) have been developed by utilizing the coagulated
61 sludge of drinking water, seawater, biologically treated sewage effluent, wastewater from the printing
62 press, dye wastewater and algae bloomed surface water (El Saliby et al., 2009; Okour et al., 2009a; El
63 Saliby et al., 2010; Kim et al., 2010). The provision of resource recovery as titania (TiO₂) and negligible
64 aquatic toxicity of residual Ti established the background of the current study.

65 Initially, Fujishima and Honda (1972) announced the revolutionary study on TiO₂ electrode
66 sensitization for H₂O electrolysis. Since then, many research groups have examined the possibility of
67 using TiO₂ as photocatalysts for the hydrolysis of H₂O, towards numerous practical implications
68 (Haider et al., 2019; Wang et al., 2019). Moreover, TiO₂ is cost-effective, non-toxic as a photocatalytic
69 substance, and has reasonably good reactivity and stability in both aqueous and atmospheric
70 environments (Diebold, 2003). TiO₂ has mostly been developed for the application of highly dispersed
71 standalone photocatalysts, dye-sensitized solar cells, and the UV/visible light irradiated degradation of
72 organic and inorganic pollutants through photocatalysis (Asahi et al., 2014; Nasirian et al., 2017).
73 However, the main shortcoming of TiO₂ that has hindered its practical application is its wide energy
74 bandgap of ~3.2 eV (Nasirian et al., 2017). Hence, standalone utilization of TiO₂ for photocatalysis can
75 only be facilitated by light irradiation in the UV region. Moreover, due to having large energy bandgap,
76 the rate of recombination of the developed photoinduced electron/hole (e⁻/h⁺) pair is elevated, which
77 substantially suppresses the photoactivity of TiO₂ (Asahi et al., 2014). Hence, to overcome the
78 limitations associated with a large energy bandgap, a number of strategies were utilized to augment the
79 photocatalytic efficiencies of TiO₂. To date, the most common strategies that have been employed
80 includes self-structural adjustment, metal/noble metal/non-metal doping, sensitization, and

81 heterojunction with other suitable semiconductors (Asahi et al., 2014; Nasirian et al., 2017). Among the
82 techniques mentioned above, the heterojunction of semiconductors has stirred outstanding interest in
83 recent years. In the last decades, an N modified TiO₂ was synthesized through pyrolysis of the precursor
84 mix of TiO₂ and urea/melamine at 400 °C, which showed notable photoactivity within the visible light
85 spectrum (Kobayakawa et al., 2005). The aforementioned visible light activation was attributed to the
86 highly condensed melamine products such as melam, melem, and melon, which are the predecessors of
87 graphitic carbon nitride (g-CN) (Zhang et al., 2019a). These research results recommended that the
88 products created from further condensation of melamine can potentially sensitize titania for visible light
89 activation, which may have created an inspirational route of TiO₂/g-CN heterojunction formation.
90 Hence, an extensive research study has been conducted on visible light-sensitive g-CN, focusing on
91 their composition, and structural arrangement to manage the necessary transformation of light energy
92 into usable chemical energy (Ong, 2017; Kumar et al., 2018; Tan et al., 2018; Xiao et al., 2018).

93 The backbone of g-CN, a nonmetal-based semiconductor, is constructed as a conjugated polymeric
94 system and contains the earth abundant C and N atoms (Xu et al., 2018a). The polymeric layers are
95 being constructed with s-triazine (C₃H₃N₃) or heptazine (C₆H₃N₇) units interlinked by tertiary amine
96 groups, and the layer atoms are considered in the arrangement of honeycomb. (Kumar et al., 2018). The
97 C₃H₃N₃ or C₆H₃N₇ based 2D layers remain interconnected through weak van der Waals forces. Hence,
98 due to the presence of aromatic heterocycles, including prevailing van der Waals forces within the
99 polymeric sheets, g-CN is thermally durable up to 600 °C in atmospheric conditions (Zhang et al.,
100 2019a). Also, g-CN showed superior chemical stabilities in solvents like water, alcohols, ethers, and
101 several weak acids. Moreover, the optical bandgap of g-CN is narrower (~2.7 eV), and with respect to
102 normal hydrogen electrode at pH= 7, the conduction band (CB) and valence band (VB) location could
103 be estimated as -1.31 and +1.40 eV, respectively (Zhou et al., 2019). Hence, as a photocatalyst, the
104 extraordinary advantages of g-CN include favorable electro-optical properties and superior visible light
105 sensitivity due to narrow energy bandgap. For the preparation of photocatalytic heterojunction,
106 molecular level chemical modification can easily be applied to facilitate favorable surface properties of
107 g-CN. Moreover, several studies inferred that g-CN showed outstanding compatibility as a host

108 nanomaterial while composited with diverse inorganic NPs (Zhou et al., 2019; Huang et al., 2020; Yu
109 et al., 2020). However, the prompt recombination of the photogenerated e^-/h^+ pairs and the inferior
110 active surface area is still two vital disadvantages limiting the photoactivity of g-CN. Therefore,
111 developing novel answers to surmount these barriers are of significant importance.

112 Over the past decades, a substantial increase in atmospheric nitrogen oxides (NO_x ; $\text{NO} + \text{NO}_2$) has
113 been observed, and photooxidation of NO_x is one of the most prominent solutions for this problem (Ma
114 et al., 2016; Papailias et al., 2017). Many research groups nowadays have focused on this topic and
115 developed a number of efficient photocatalysts to oxidize NO_x into a more neutral product. The suitable
116 positioning of the CB and VB edge of anatase TiO_2 and g-CN, along with the 2D sheet-like structure of
117 g-CN, favors successful heterojunction. It has been reported that g-CN could form a thin layer on TiO_2
118 NPs, while heterojunction is formed (Jiang et al., 2018). Additionally, type II heterojunction has been
119 reported for anatase $\text{TiO}_2/\text{g-CN}$ composites due to the placement of their CB and VB edges (Ma et al.,
120 2016). Therefore, photogenerated e^- and h^+ can migrate between the band edges of TiO_2 and g-CN,
121 facilitating the hindered rate of recombination. Giannakopoulou et al. (2017) demonstrated an easy
122 solvent-free mechanical mixing followed by calcination technique to tailor the energy bandgap of
123 $\text{TiO}_2/\text{g-CN}$, to facilitate NO oxidation. They oxidized an initial concentration of 1 ppm NO under both
124 visible (7000 lux) and UV (10 Wm^{-2}) light irradiation. After 40 min of irradiance, they attained a
125 maximum of 38% and 42% of removal for visible and UV light, respectively. Papailias et al. (2017)
126 similarly removed NO_x using $\text{TiO}_2/\text{g-CN}$ with a removal rate of 18% and 22.5% after 30 min by using
127 visible and UV irradiation, respectively. This was attained by modifying the surface morphology of the
128 as-prepared photocatalysts with the assistance of CaCO_3 . Recently, Huang et al. (2019) used a simple
129 ultrasonic exfoliation technique and synthesized $\text{Ti}^{3+} \text{TiO}_2/\text{p-CN}$ composite for the purpose of NO_x
130 oxidation. Using an initial NO concentration of 400 ppb, a removal percentage of 25.8% after 30 min
131 of irradiation was reported.

132 Despite numerous studies conducted for NO_x removal using $\text{TiO}_2/\text{g-CN}$ composites, in most cases,
133 commercially available titania was used as the precursor of TiO_2 , which increases costs for
134 photocatalyst synthesis. Therefore, in this study, we have synthesized a novel $\text{TiO}_2/\text{g-CN}$ heterojunction

135 by using sludge generated titania (S-TiO₂) as the precursor of TiO₂. We have used a template-free
136 approach and utilized the precursor mix of S-TiO₂ and melamine for successful heterojunction.
137 Advanced chemical, morphological and optical characterizations of as-synthesized samples have been
138 conducted. ISO 22197-1 (2007) and ISO 17198-1 (2018) was followed to evaluate the prepared samples'
139 NO_x photooxidation capacity under UV and visible light, respectively. Besides, nitrate selectivity of the
140 ready samples was assessed and based on the advanced characterizations, a possible photooxidation
141 scheme was proposed.

142 **2 Materials and methods**

143 *2.1 Preparation of S-TiO₂ from dye wastewater sludge*

144 The wastewater produced by a dye wastewater treatment plant located in Daegu, South Korea, was
145 collected for the purposes of this study. The initial physicochemical properties were as follows: pH,
146 11.7; chemical oxygen demand (COD), 449 mg/L; total nitrogen (TN), 72 mg/L; total phosphorous, 3.2
147 mg/L. The wastewater was treated with an optimized dose of TiCl₄, which was first rapidly mixed within
148 the sample at a rate of 100 rpm for 1 min before being slowly mixed at a rate of 30 rpm for a further 20
149 min. The flocculated wastewater was subsequently stored to allow the flocks to settle, after which the
150 settled dye wastewater sludge was extracted dried overnight in a laboratory oven at a temperature of
151 100°C. A pestle and mortar was used to mill the dried material into a powder form before the anatase S-
152 TiO₂ was synthesized by placing the powder into an alumina crucible and calcining it at 600 °C
153 (@10°C/min) for 3 h.

154 *2.2 Preparation of S-TiO₂/g-CN heterojunction*

155 The pure g-CN and conjugated S-TiO₂/g-CN was prepared via a facile one-step synthesis scheme.
156 A box furnace was utilized for annealing the dry mixture of S-TiO₂ and melamine (Sigma-Aldrich, 99%)
157 for the sample preparation. The annealing was performed under atmospheric pressure. Various samples
158 were created by differing the ratio of wt.% of melamine to S-TiO₂, and a calcination temperature of
159 550 °C was applied. The temperature of the box furnace was fixed such that it rose at 10 °C/min with a
160 calcination duration of 3 h. Table 1 presents an overview of the nomenclature of the prepared samples
161 together with details of the mixing ratios of the precursors.

162 **Table 1.** Nomenclatures and salient features of the prepared samples. STC indicates the S-TiO₂/g-CN
 163 samples at various melamine concentrations.

ID	Amount(g)		Prepared sample (g)	Mixing ratio	Temperature (°C)	Ramp (°C/min)	Duration (h)
	S-TiO ₂	Melamine					
STC1	4	2	4.8	2:1	550	10	3
STC2	4	4	5	1:1	550	10	3
STC3	4	8	7.8	1:2	550	10	3
STC4	4	12	8.1	1:3	550	10	3
STC5	4	16	11	1:4	550	10	3
STC6	4	20	13.2	1:5	550	10	3
g-CN		5	1.1		550	10	3

164 **2.3 Characterization**

165 X-ray diffraction (XRD) analysis was performed to characterize the crystalline phase of the
 166 synthesized samples. XRD patterns were generated on an MDI Jade 5.0 X-ray diffractometer (D/MAX
 167 Ultima III, Rigaku, Japan) operating at 40 kV and 40 mA that was fed with a Cu K α radiation source.
 168 By using the Bragg-Brentano experimental method, the patterns of XRD were documented at a scanning
 169 rate of 0.02/sec. Besides, the patterns were generated within the scattered angle (2θ) array of 5° to 90°.
 170 The X-ray beam was operated with limiting parameters of 2/3° divergence, 10 mm divergent altitude,
 171 0.3 mm receiving slits, and 2/3° scattering.

172 A scanning electron microscope (S-4700, Hitachi, Japan) that was operated at 15 kV in a vacuum
 173 with a working gap of 7 mm was used to examine the morphology of the powdered S-TiO₂/g-CN. A
 174 scanning electron microscope that was supplied with an energy dispersive X-ray detector (55VP SEM)
 175 operating at 15 kV was employed to perform the elemental composition analyses. To further investigate

176 the samples' crystallinity and internal structure, a transmission electron microscope (FE-TEM, JEOL
177 Ltd., JEM-2100F, Japan) was utilized. The TEM samples were prepared by evaporating a droplet of the
178 powders placed on a Cu grid coated with holey carbon support films in ethanol. The as-prepared
179 samples' effective surface area was estimated using the Brunauer-Emmett-Teller (BET) theory. The
180 adsorption-desorption isotherms were generated in Belsorp mini II, BEL, Japan. The desorption data
181 was subsequently used in combination with the Barrett-Joyner-Halenda (BJH) framework to investigate
182 the pore volume distributions.

183 To confirm the presence of anatase titania and g-CN, the IR Prestige-21, Shimadzu, Japan, was
184 utilized to generate the Fourier transform infrared (FT-IR) patterns of the as-prepared samples. The
185 range of wavenumbers was prefixed between 400 cm^{-1} and 4000 cm^{-1} ; moreover, the samples were
186 prepared using the KBr pellet preparation method. The X-ray photoelectron spectroscopy (XPS) spectra
187 of the dry samples were generated on a MultiLab2000, VG, UK, to validate the various chemical bonds
188 present in the samples. The photoluminescence (PL) spectra were generated on a dual microplate
189 spectrofluorometer, Horiba, Piscataway, NJ, USA, to evaluate the charge recombination rates of the
190 prepared samples during photoactivity. A sample excitation wavelength of 300 nm was utilized in the
191 fluorometer. A UV-Vis-NIR spectrometer (Lambda 950, Perkin Elmer, KBSI Daegu center, South
192 Korea) that was fitted with a diffuse reflectance (DRA) accessory was used to obtain the UV-VIS
193 spectra, which were documented in a spectral range of between 400 and 800 nm.

194 The optical bandgaps of the as-prepared samples were determined by using the modified Kubelka-
195 Munk functions illustrated in the following equations,

$$196 \quad F(R) = \frac{K}{S} \quad (1)$$

$$197 \quad K = (1 - R)^2 \quad (2)$$

$$198 \quad S = 2R \quad (3)$$

199 Here, $F(R)$ is the Kubelka-Munk function, and the K , R , and S are molar absorption coefficient,
200 scattering factor, and reflectance data, respectively. Additionally, for sludge generated titania

201 (anatase), an indirect energy bandgap was considered, and it was determined through extrapolation of
202 $F(R)hv)^{1/2}$ vs. hv . In contrast, the bandgaps of the as-prepared samples were estimated considering
203 direct energy bandgap $(F(R)hv)^2$ vs. hv .

204 2.4 Photocatalytic activity

205 Following ISO 22197-1 (2007) and ISO 17198-1 (2018) conventions, the as-prepared samples'
206 NO_x removal capacity was investigated under UV and visible light irradiation, respectively. In a
207 continuous flow reactor, the samples were placed and irradiated (pretreatment) at a light intensity of 10
208 W/m^2 for 5 h. For the pretreatment, two UV-A lamps from Sanyo-Denki, Japan were used. The
209 experimental procedures that were followed are explained in previous work (Rhee et al., 2018).

210 Briefly, a rectangular sample holder with the surface area of 50 cm^2 was utilized for holding 1 g of
211 uniformly distributed samples. At the inlet of the flow reactor, the ambient parameters were prefixed
212 following the aforementioned ISO protocols. At the reactor's inlet, the moisture content and temperature
213 were maintained as 50% and $25\text{ }^\circ\text{C}$, respectively. For UV and visible light irradiation, UV-A lamps at a
214 light intensity of $10\text{ W/m}^2 \pm 0.5\text{ W/m}^2$ and white fluorescent lamps (with UV cut-off filter) at an
215 intensity of $6000\text{ Lx} \pm 300\text{ Lx}$ was utilized, respectively. To assess the prepared samples' photoactivity,
216 each time the light irradiation was maintained for 1 h. Besides, a constant airflow of 3 L/min was
217 retained at the inlet to facilitate a continuous flow of 1 ppmv of NO. The subsequent concentrations
218 NO_x , NO, and NO_2 were recorded on a NO_x analyzer (CM2041, Casella). For each sample, the
219 photocatalytic experiment was continued for 100 min, where 20 min of gas adsorption and 20 min of
220 gas desorption were considered at the beginning and end of light irradiation. A simple schematic of the
221 utilized experimental setup is illustrated in Fig. S1. The NO and NO_x removal rate along with the NO_2
222 generation rate and nitrate (NO_3^-) selectivity were estimated using the following equations,

$$223 \quad NO_{removal}(\%) = \frac{NO_{in} - NO_{out}}{NO_{in}} \times 100 \quad (4)$$

$$224 \quad NO_{2,generation}(\%) = \frac{NO_{2,out} - NO_{2,in}}{NO_{in}} \times 100 \quad (5)$$

$$225 \quad NO_{x,removal}(\%) = \frac{NO_{x,in} - NO_{x,out}}{NO_{in}} \times 100 \quad (6)$$

$$NO_3^- \text{selectivity}(\%) = \frac{NO_{x,removal}}{NO_{removal}} \times 100 \quad (7)$$

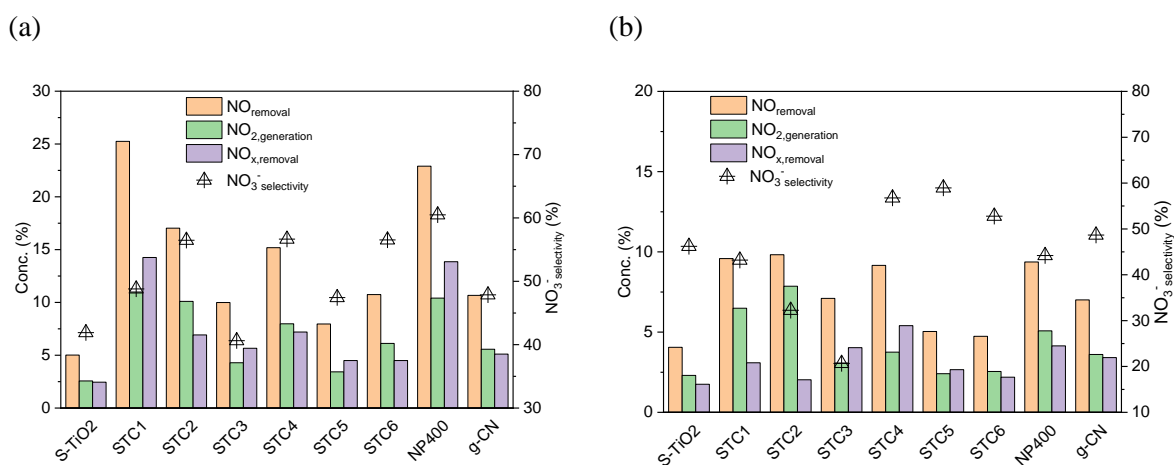
227 **3 Results and discussion**

228 *3.1 Photocatalytic activity*

229 The photoactivity of the synthesized samples was assessed through estimating the extent of NO
230 oxidation under UV and visible light irradiance. In addition, the photoactivity of the as-prepared
231 samples were compared with standalone use of S-TiO₂, NP400, and g-CN. Fig. S2 and S3 show the NO,
232 NO₂, and NO_x concentration trends of as-prepared samples, STC1 – STC6, S-TiO₂, NP400, and g-CN,
233 under light (UV/visible) irradiance. Moreover, the detailed results of total NO and NO_x removal along
234 with NO₂ generation during 1 h of light irradiation are tabulated in Table S1 (UV light) and S2 (visible
235 light). For each experiment, 7.58 ± 0.069 μmol NO was inserted in the reactor, and irrespective of the
236 light types (UV/visible), both S-TiO₂ and NP400 showed similar NO removal profile as illuminated in
237 Fig. S2(a) and (h), respectively. For all the samples examined, the NO and NO_x concentrations initiated
238 to decrease instantly at the light irradiance (UV/visible) and reached the minimum concentration of NO
239 and NO_x within 5 min of light irradiation. Negligible response lag was witnessed at the beginning and
240 the end of the light irradiation due to the instrument's default factory setup. The rapid degradation of
241 NO at the starting of the light irradiance could be attributed to the dual effects of adsorption and
242 photooxidation on the photocatalytic sites of the respective samples. Later, due to saturation of the
243 photocatalytic sites with the protoxidized by-products (i.e., NO₂, NO₃⁻), the NO removal efficiencies
244 tend to decrease gradually (Park et al., 2014). Fig. S2 and S3 show that the gradual decrease in NO and
245 NO_x removal with irradiation time is dominant in S-TiO₂ and NP400. Interestingly, compared to anatase
246 titania (S-TiO₂ and NP400), the as-prepared samples (STC1-STC6) and g-CN showed stable NO and
247 NO_x removal within 1 h of irradiation (UV/visible), which could be the indication of superior
248 photoactivity.

249 Fig. 1(a) and (b) reveal the corresponding NO removal, NO₂ generation, and NO₃⁻ selectivity of
250 the as-synthesized samples along with S-TiO₂, NP400, and g-CN under UV and visible light,
251 respectively. The S-TiO₂ prepared from flocculation of dye wastewater showed very limited

252 photoactivity under any kind of light irradiation, whether UV or visible light, which is in contrast with
 253 some related studies (El Saliby et al., 2012). Under 1 h of UV and visible light irradiation, S-TiO₂ only
 254 removed 0.378 μmol and 0.303 μmol of NO, respectively (see Table S1 and S2), which is around 5%
 255 of the total NO inserted in the reactor. On the other hand, with the same setup, NP400 showed around
 256 22.89% and 9.36% NO removal under UV and visible light, respectively. As Fig. 1(a) illustrates, the
 257 prepared samples under UV irradiation showed significant improvement in NO removal. STC1 showed
 258 the maximum NO removal of 25.25%, which is around 5 times higher compared to S-TiO₂. During the
 259 synthesis of the sample STC1, the precursor mix contained around 33.33% (Table 1) of melamine, and
 260 in the following samples, the melamine contents were increased as 50%, 66.67%, 75%, 80%, and
 261 83.33%. With increased melamine content in the precursor mix, at 50% melamine content, the NO and
 262 NO_x removal get reduced to 17.03% and 6.92%, respectively. Later, at 75% melamine content, the
 263 sample showed enhanced NO_x removal of 7.82%, and a further increase in melamine content within the
 264 precursor showed a consecutive reduction in NO and NO_x removal. One dominant trait that has been
 265 observed is, with increased melamine proportion in the precursor mix, the NO₃⁻ selectivity of the
 266 samples showed an increasing trend and begun to decline while the melamine proportion exceeded 75%
 267 of the initial mix.



268 **Fig. 1.** NO_x removal and NO₃⁻selectivity of the as-prepared samples under 1 h of (a) UV light, and (b)
 269 visible light irradiation.

270 Fig. 1(b) shows that STC4 achieved the maximum NO_x removal of 5.4% under visible light due to
271 superior NO₃⁻ selectivity of 56.76%. As illustrated in Table S2, in total, 7.63 ± 0.067 μmol NO_x was
272 inserted in the reactor under visible light irradiation of 1 h for each sample. At the end of 1h of visible
273 light irradiation, STC4 was found to remove around 0.41 μmol of NO_x, whereas S-TiO₂, NP400, and g-
274 CN removed around 0.13 μmol, 0.31 μmol, and 0.25 μmol, respectively. For NO₃⁻ selectivity under
275 visible light, a similar pattern was observed as photooxidation under UV light. Initially, selectivity
276 increased with the increase in the precursor mix's melamine content, and after 80%, it started to decrease.

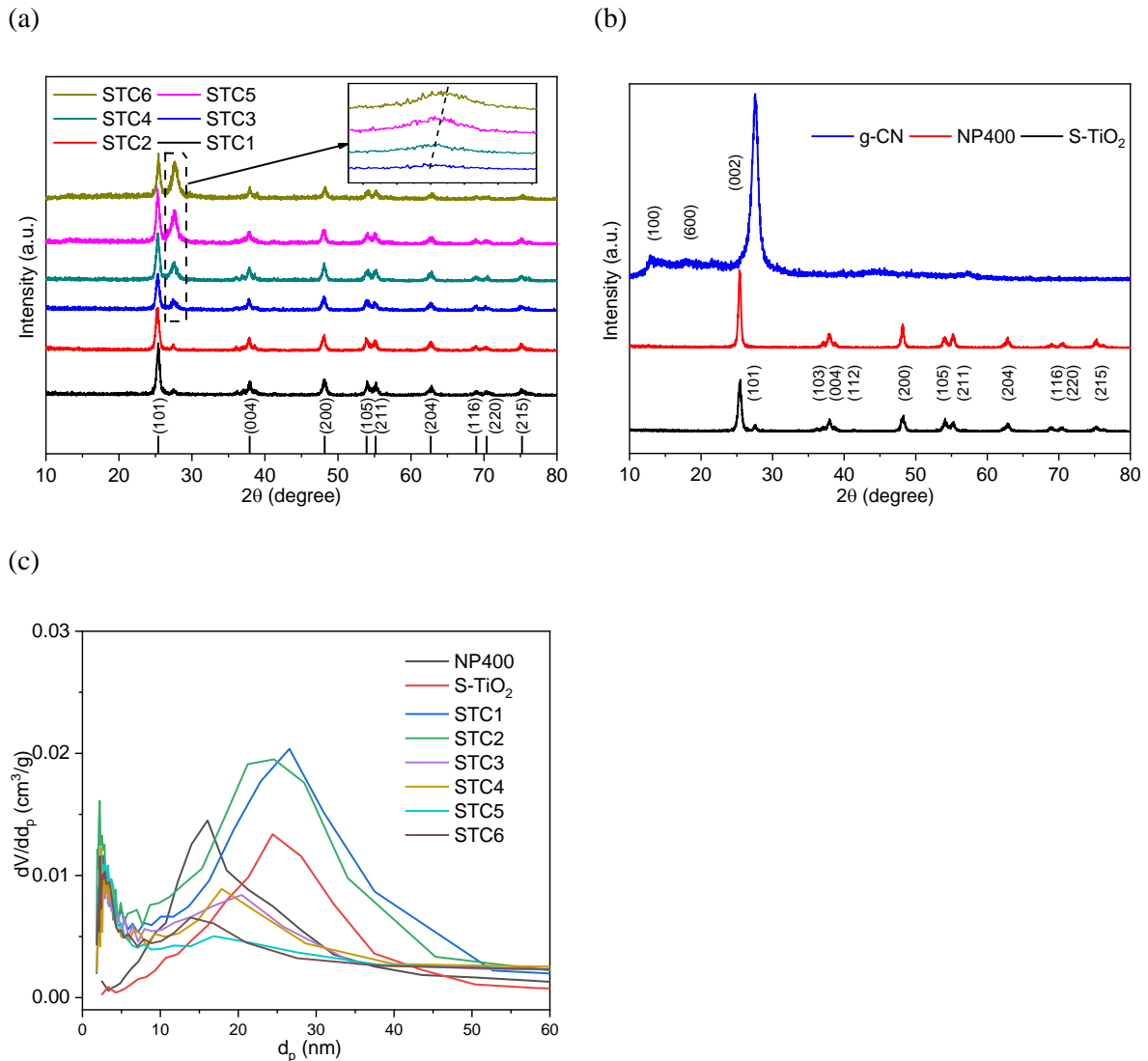
277 3.2 Morphological attributes

278 3.2.1 XRD and BET analysis

279 XRD was conducted to identify the phase configuration of the synthesized samples as tabulated in
280 Table 1, and the patterns were compared with commercially available NP400 and laboratory prepared
281 g-CN and S-TiO₂. Fig. 2(b) shows that the prepared S-TiO₂ using dye wastewater is mainly of anatase
282 form. The corresponding anatase crystal planes that were found are, (101), (103), (004), (112), (200),
283 (105), (211), (204), (116), (220), (215), and (224) (JCPDS No. 21-1272) (Xia et al., 2019; Zhang et al.,
284 2019b; Yu et al., 2020). On the other hand, the g-CN prepared in this study showed two characteristic
285 peaks around 2θ of 13.04° and 27.58°, representing the spacing between the adjacent lattice planes of
286 0.693 and 0.324 nm respectively (Wang et al., 2015). The dominant crystal planes observed in g-CN are
287 depicted in Fig. 2(b), where plane (100) corresponds to the N-bridged repetition of C₆H₃N₇ units and
288 (600) explains the inter-planer assembling of the conjugated aromatic system (Dong et al., 2015).

289 From Fig. 2(a), it is obvious that the synthesized S-TiO₂/g-CN heterojunction preserved the anatase
290 planes of S-TiO₂, and no significant shift in 2θ was observed. Additionally, the peaks for anatase planes
291 remained intact irrespective of the variations in weight ratios of S-TiO₂ and melamine in the precursor
292 mix, which are in coherence with relevant studies (Tan et al., 2018; Troppová et al., 2018; Wang et al.,
293 2018b). At lower melamine content of almost up to 50% of the precursor mix, no characteristic XRD
294 pattern of g-CN was found within the prepared samples. Presumably, this could be due to the formation
295 of negligible amounts of g-CN in the samples. However, once the melamine proportion was twice than
296 that of S-TiO₂ in the precursor mix, an obvious peak around 27.58° (2θ) depicted the presence of

297 characteristics (002) plane of g-CN. With a further increase in melamine content during the synthesis
 298 of the samples, the peak around 27.58° became wider and steeper. The enhancement in peak intensity
 299 for g-CN in the consecutive samples (STC1 to STC6) could be attributed to the better crystallinity
 300 (Dong et al., 2015).



301 **Fig. 2.** XRD images of the (a) as-prepared samples STC1 to STC6, and (b) NP400, g-CN and S-TiO₂;
 302 (c) evaluated pore size distribution (BJH adsorption) of the as-prepared samples.

303 Additionally, with the increased content of g-CN in the samples, the (002) lattice plane of g-CN
 304 tends to shift slightly toward high 2θ value, due to the reduced gallery distance of the basic aromatic
 305 sheets in g-CN. Moreover, due to slight increase in the total content (see Table 1) of the precursor mix,
 306 there could be a slight increase in the internal temperature during the synthesis, which could be the

307 reason behind the slight shift of the (002) plane of the composited g-CN (Dong et al., 2015; Wang et
308 al., 2015). More melamine in the precursor mix will also generate more ammonia, HCN and CN_x gases
309 which renders the atmosphere in furnace were more reductive. Compared to standalone calcination of
310 melamine, the calcination of S-TiO₂ mixed melamine tends to destroy the in-plane periodic C₆H₃N₇
311 units; hence, the lattice plane of g-CN around 13.04° was diminished in the as-prepared samples (Kočí
312 et al., 2017). From the above discussion and by using Fig. 2, it is evident that the prepared samples
313 exhibited superposition of the characteristics XRD peaks of both S-TiO₂ and g-CN. Hence, it can be
314 presumed that during the calcination of the dry mix of S-TiO₂ and melamine, g-CN was assembled on
315 the facade of S-TiO₂, without affecting the crystal lattice arrangement of either of the components.

316 Moreover, to evaluate the effects on crystal size of the as-synthesized samples, scherrer's equation
317 (Song et al., 2016; Giannakopoulou et al., 2017; Papailias et al., 2017) was utilized at the dominant
318 plane (101) of the anatase TiO₂, and the results are reported in Table 2.

$$319 \quad D = \frac{K\lambda}{FWHM \times \cos\theta} \quad (8)$$

320 Here, D represents the crystal size, while K, λ, FWHM, and θ are the Scherrer's constant, wavelength,
321 full width at half maxima, and Bragg's angle. From the crystal sizes reported in Table 2, incorporation
322 of g-CN with S-TiO₂ did not show any remarkable changes. The average crystal size of the synthesized
323 samples was determined as 15.42 ± 0.66 nm, which was similar to the pristine S-TiO₂ (15.40 nm).

324 The effective surface area (S_{BET}) and pore size (d_p) distribution of the as-prepared samples and
325 pristine S-TiO₂ were determined by utilizing N₂ physisorption isotherms. Table 2 depicted the
326 respective S_{BET} values and pore volumes (V_T) of the as-synthesized samples. Furthermore, the d_p
327 distribution curves evaluated using BJH technique were incorporated in Fig. 2(c). S-TiO₂ showed a
328 comparable surface area to commercially available NP400. From the S_{BET} values tabulated in Table 2,
329 a declining trend was observed with the increase of g-CN content in the synthesized samples. Besides,
330 from Fig. 2(c) it is evident that both S-TiO₂ and NP400 showed mesoporous traits, where integration of
331 g-CN in the samples showed the combination of mesoporous and microporous nature (pore size
332 distribution of STC1 – STC4).

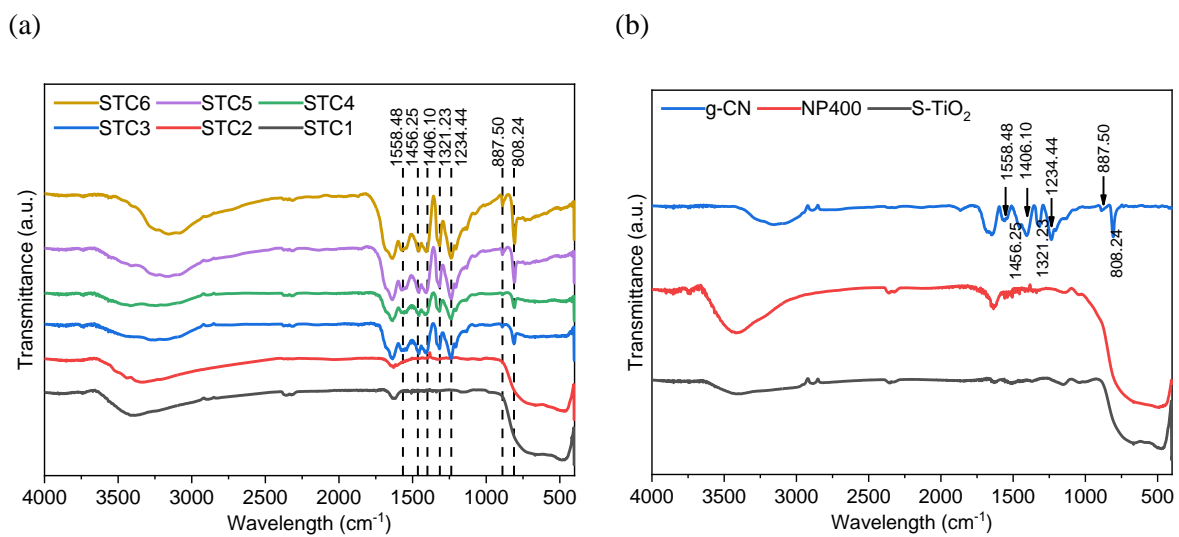
333 **Table 2.** Estimated FWHM and crystal size of the samples along the dominant anatase (101) plan of
 334 TiO₂, effective surface area (S_{BET}), and total pore volume (V_T).

ID	2θ (degree)	FWHM	Crystal size (nm)	S _{BET} (m ² /g)	V _T (cm ³ /g)
S-TiO ₂	25.41	0.529	15.40	51.69	0.31
STC1	25.31	0.475	16.23	45.45	0.31
STC2	25.28	0.548	14.87	52.33	0.31
STC3	25.31	0.533	15.29	35.0	0.29
STC4	25.33	0.501	16.26	33.69	0.26
STC5	25.34	0.550	14.80	30.47	0.23
STC6	25.44	0.540	15.08	30.40	0.23
NP400	25.40	0.370	21.87	66.65	0.34

335 3.2.2 FT-IR spectra

336 To further validate the occurrence of S-TiO₂ and g-CN in the as-synthesized samples, Fourier-
 337 transform infrared spectroscopy (FT-IR) analysis was conducted, and the findings were compared with
 338 individual FT-IR of S-TiO₂, NP400, and g-CN. The FT-IR spectra of the synthesized samples STC1 –
 339 STC6 are illustrated in Fig. 3(a), while Fig. 3(b) depicted the FT-IR patterns of S-TiO₂, NP400, and g-
 340 CN. For S-TiO₂ and NP400, the broad absorption peak across 400 – 700 cm⁻¹ (Fig. 3(b)) can be allocated
 341 to the Ti-O and Ti-O-Ti stretching forms (Liu et al., 2019). Additionally, the broad absorption band
 342 around 3000 – 3700 cm⁻¹ can be ascribed to the O-H stretching of the adsorbed H₂O on the samples
 343 (Kočí et al., 2017; Papailias et al., 2017). The prepared g-CN showed FT-IR peaks around 808.24 cm⁻¹
 344 due to the bending mode of skeletal triazine cycles (Zhang et al., 2018a). For g-CN, a series of
 345 absorption bands were found within the wavelength range of 1200 – 1600 cm⁻¹, as shown in Fig. 3(b),
 346 which are mainly attributed to the C-N stretching of the aromatic part (Troppová et al., 2018). Besides,
 347 the wide absorption band within 3000 – 3500 cm⁻¹ could also be due to the vibration of N-H from the
 348 remaining NH₂ groups (Du et al., 2019).

349 From Fig. 3(a), STC1 and STC2 only showed the characteristics absorption bands for Ti-O and Ti-O-
 350 Ti stretching modes, whereas presence of g-CN was not confirmed which corresponds well with the
 351 obtained XRD data. On the other hand, all other samples starting from STC3 to STC6 evidently showed
 352 the absorption bands for the aromatic C-N stretching and anatase TiO₂. Additionally, with increased g-
 353 CN content in the synthesized samples, the comparative intensity of the characteristics FT-IR peaks for
 354 g-CN became broader and steeper. Another interesting finding from assessing the FT-IR spectra of the
 355 prepared samples was a slight shift of the wide peak across 400 – 700 cm⁻¹ towards greater wavenumber
 356 (see Fig. 3(a)), which is in coherence with relevant composite studies (Song et al., 2016;
 357 Giannakopoulou et al., 2017). Presumably, the presence of g-CN in the sample modified the chemical
 358 environment on the surface of S-TiO₂, and relevant studies confirmed that interaction between the
 359 molecular orbitals of the separate components happens due to close heterojunction (Song et al., 2016).
 360 Hence, the shift in the Ti-O-Ti stretching modes of the as-synthesized samples may infer robust
 361 heterojunction at the edge of S-TiO₂ and g-CN.



362 **Fig. 3.** FT-IR patterns of the (a) as-prepared samples STC1 to STC6, and (b) NP400, g-CN and S-TiO₂.

363 3.2.3 Electron Microscopy

364 To authenticate the compositions and morphology of the as-prepared samples, scanning electron
 365 microscope (SEM) was conducted, and the SEM images of STC1 to STC6 are illustrated in Fig. S4(a –
 366 f). Additionally, for comparison purposes, SEM images of S-TiO₂ and g-CN are incorporated in Fig.

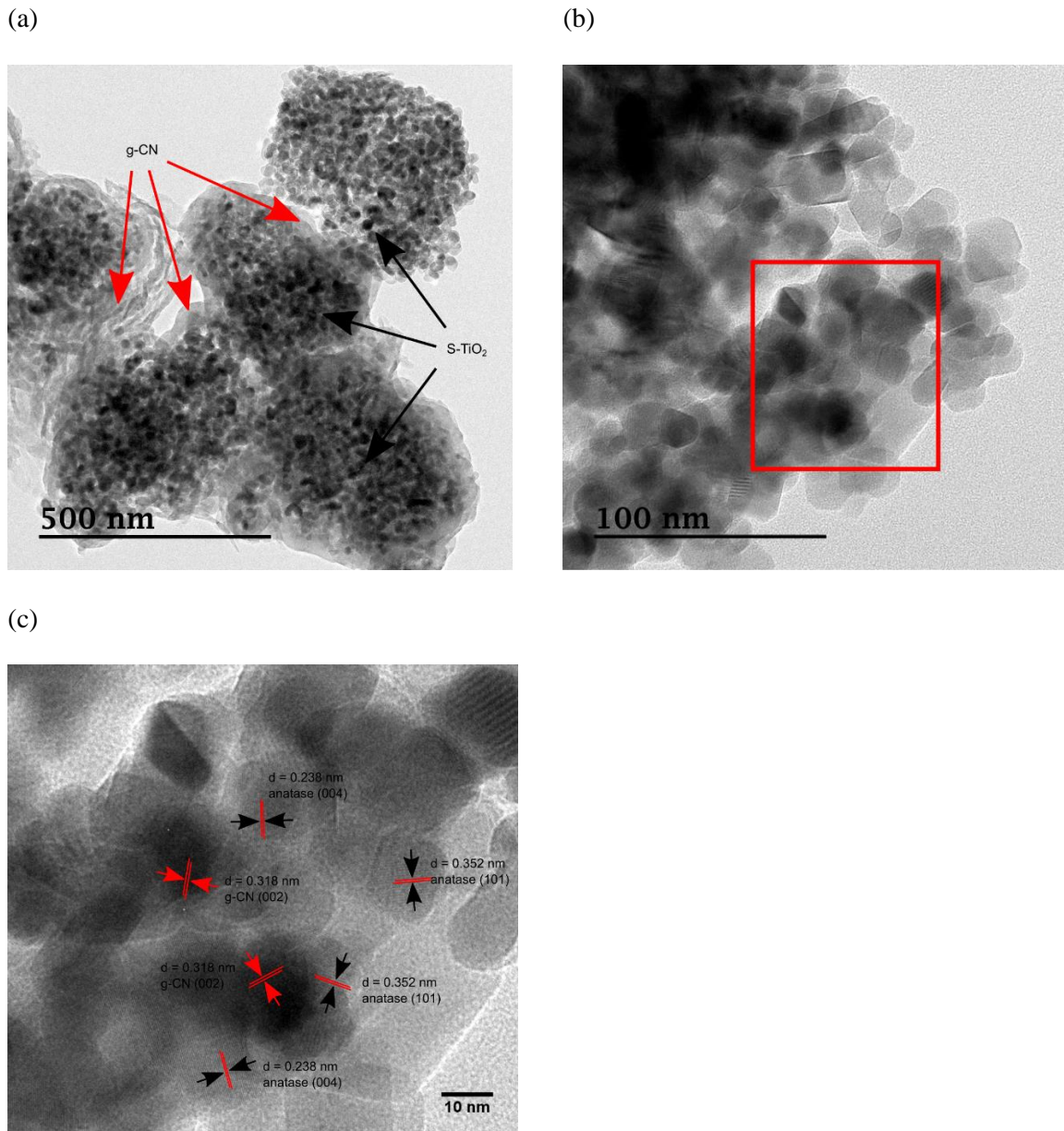
367 S4(g) and (h), respectively. As a representative of the as-prepared samples transmission electron
368 microscopy (TEM) image of STC4 is incorporated in Fig. 4 at varying magnifications. The prepared g-
369 CN exhibited the presence of a 2D layer sheet structure together with a fairly smooth and flat facade,
370 where the S-TiO₂ prepared from wastewater was found as irregular structure (Fig. 4(a)). From Fig. S5,
371 based on the energy-dispersive X-ray spectroscopy (EDX) results, S-TiO₂ showed the presence of the
372 C atom (8.50%), which is in coherence with former relevant research (Shon et al., 2007). In a number
373 of previous studies, recovered TiO₂ from wastewater sludge inferred the presence of doped C in the as-
374 prepared samples, and the presence of prevailing organic compounds was concluded as the source of
375 the doped C atoms (Shon et al., 2007; Shon et al., 2009a).

376 By observing the SEM image of STC1 to STC6 (Fig. S4), it is clear that the incorporation of
377 melamine in the precursor mix reduced the agglomeration of S-TiO₂ NPs and more uniform dispersion
378 was observed (Giannakopoulou et al., 2017). Accordingly, based on the EDX result presented in Fig.
379 S5, the presence of C, N, O, and Ti atoms was confirmed in the samples, and with increased melamine
380 content during synthesis, the proportion of g-CN tends to increase. Based on the EDX and SEM results,
381 no g-CN were formed in the STC1 and STC2; however, the presence of C atoms was observed in the
382 as-prepared samples. Initially, by comparing S-TiO₂ and STC1, the atomic proportion of C was reduced
383 from 8.50% to 5%. Further increase of melamine in STC2, enhanced the C content (7.37%) of the
384 sample.

385 In STC3, i.e., in the initial mix while the melamine content was 150% of S-TiO₂, the occurrence
386 of g-CN was observed in the TEM and EDX (Fig. S5) results. As shown in Fig. 4, the TEM of STC4,
387 depicted the presence of 2D sheet like g-CN along with a considerable amount of somewhat spaced
388 sphere-shaped NPs (S-TiO₂). Moreover, in the TEM image of STC4, transparent lattice fringes at a
389 distance of 0.352 and 0.238 nm were detected, which can be assigned to the anatase lattice planes (101)
390 and (004), respectively (Zhang et al., 2014; Wei et al., 2016).

391 The TEM image of STC4 (Fig. 4) and SEM image of STC3-STC6 (Fig. S4) explicitly showed the
392 successful grafting of S-TiO₂ over the g-CN nanosheets. In the as-prepared samples, S-TiO₂ NPs appear
393 denser and more compact due to the excellent heterojunction effect and found beneficial for the transfer

394 of photogenerated e^-/h^+ pairs. The TEM image shows laminar structure of g-CN, loaded with dense and
395 uniformly distributed S-TiO₂ NPs without any apparent accumulation, demonstrating a strong fusion
396 between S-TiO₂ and g-CN.

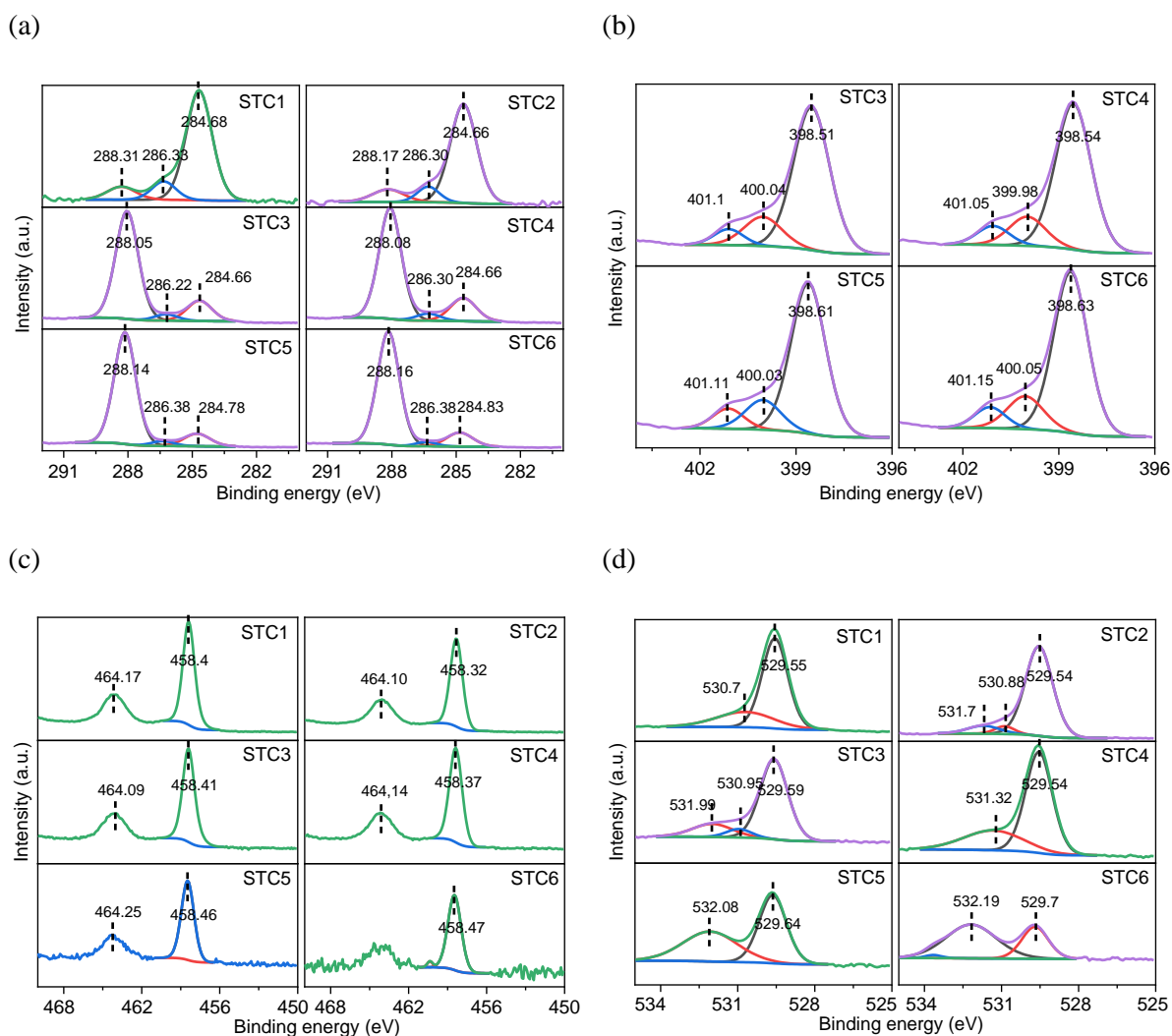


397 **Fig. 4.** TEM images of the as-prepared STC4 at varying scale ((a) 500 nm, (b) 100 nm, and (c) 10 nm)
398 showing the presence of S-TiO₂ and g-CN.

399 3.2.4 XPS spectra

400 To have an in-depth idea concerning the oxidation states and the chemical compositions of the
401 prevailing components in the as-synthesized samples, X-ray photoelectron spectroscopy (XPS) analysis

402 was conducted. Fig. S6 illuminated the wide-angle XPS survey spectra of the as synthesized samples.
 403 From the Fig. S6, the sharp photoelectron peaks for C 1s, N 1s, Ti 2p, and O 1s are clearly visible across
 404 the binding strengths of 286, 400, 458, and 530 eV (Li et al., 2015; Liu et al., 2019). Based on the survey
 405 spectra, peaks for Ti 2p and O 1s has significantly reduced in the samples STC5 and STC6. Besides,
 406 the prepared g-CN revealed the characteristics C 1s and N 1s peaks on the wide-angle survey scan.



407 **Fig. 5.** High resolution survey scan of (a) C 1s, (b) N 1s, (c) Ti 2p, and (d) O 1s for the synthesized
 408 STC1 to STC6.

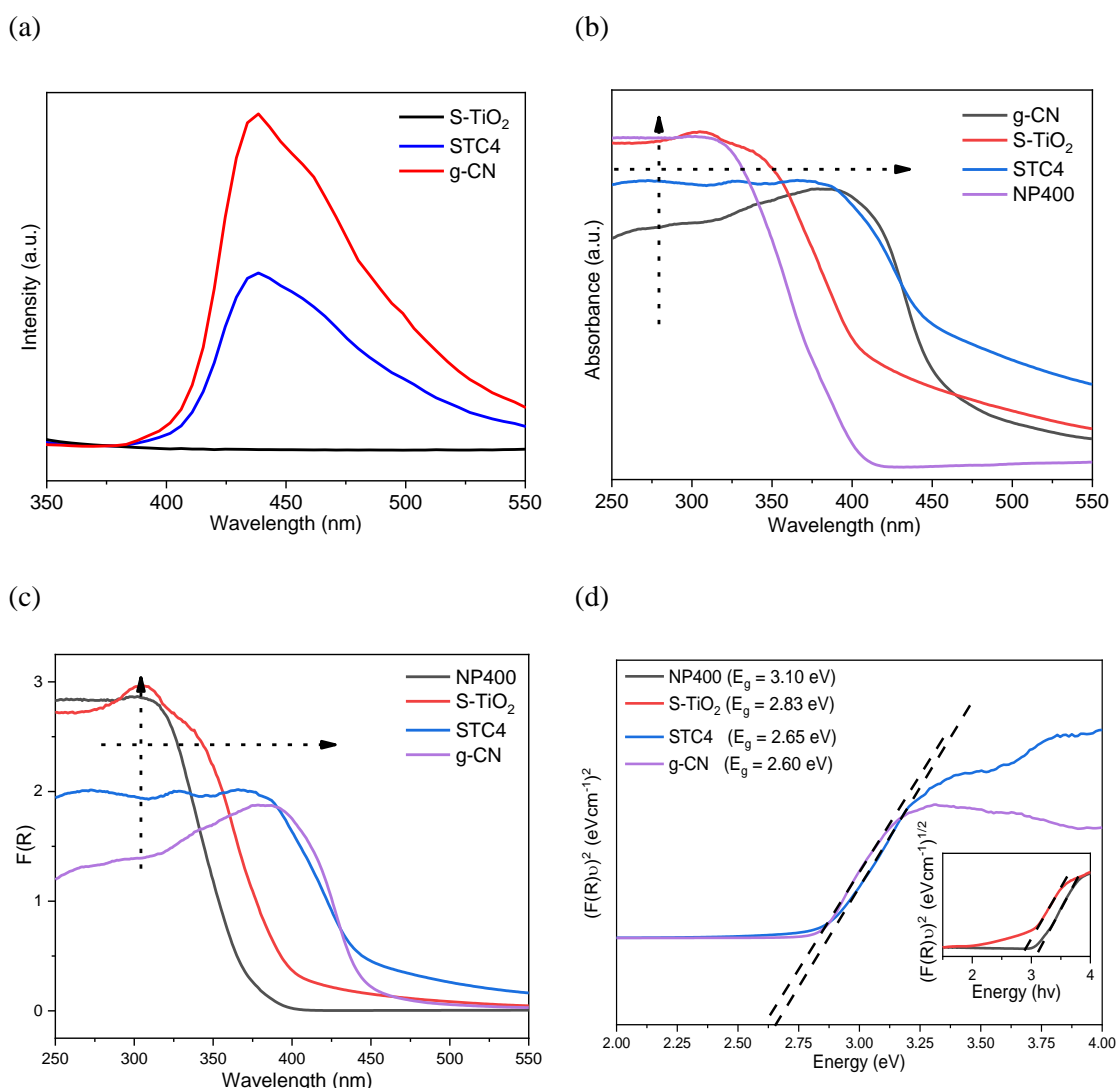
409 Fig. 5(a – d) shows the high-resolution survey spectra C 1s, N 1s, Ti 2p, and O 1s, respectively,
 410 for the as-prepared samples STC1 to STC6. Besides, the peak positions, along with the atomic
 411 proportions of the narrow scans for the corresponding samples, are tabulated in Table S3. Lorentz peak
 412 fitting was utilized for the deconvolution of the narrow scan XPS peaks. Based on the FT-IR and XRD

413 analysis, the presence of g-CN was not found in STC1 and STC2; hence the deconvoluted 3 C 1s peaks
414 shown in Fig 5(a) could be attributed to adventitious carbon as Sp^2 hybridized C-C (284.70 eV), C-O-
415 C (286.30 eV) and O-C=O (288.20 eV) bond present as defects (Zhou et al., 2014). While for the
416 prepared g-CN, STC3, STC4, STC5, and STC6, the dominant C 1s narrow scan peak was found around
417 ~ 288.10 eV, which can be assigned to N-C=N₂ coordination and characteristics peak for g-CN (Thomas
418 et al., 2008; Liu et al., 2019). Additionally, the absence of no visible peak for C-Ti confirmed that there
419 were no chemical reactions between S-TiO₂ and g-CN, rather successful heterojunction was prepared.

420 Except for STC1 and STC2, from the narrow survey N 1s spectra of the as-synthesized samples
421 dominant peak for triazine rings, i.e., sp^2 hybridized C=N-C was found around ~ 398.54 eV (Caudillo-
422 Flores et al., 2019). Moreover, peaks for tertiary N-(C)₃ and N-H group were found around ~ 399.98 and
423 401.05 eV (Li et al., 2019; Liu et al., 2019). Moreover, the peak positions were in coherence with the
424 laboratory-made g-CN (Table S3). In STC1 and STC2, characteristic peaks for g-CN were absent, and
425 very low concentration of N atoms was found in doped condition, which was 0.8% and 0.7%,
426 respectively, which could be the reason for enhanced NO_x removal beneath UV light. The O 1s spectra
427 of the as-prepared samples could mainly be fitted around 529.55 and 530.7 eV, which clearly confirms
428 the presence of lattice O²⁻ and O₂⁻ ions/OH⁻ groups respectively (Huang et al., 2019). With the increase
429 of g-CN in the prepared samples, the peak around 530.7 eV tends to shift towards higher binding energy,
430 which inferred that g-CN firmly combines with the oxygen groups presented on the facade of S-TiO₂.
431 Fig. 5(b) indicates the Ti 2p bands of the as-prepared samples STC1 to STC6. Sharp peaks around
432 458.66 and 464.41 eV has been observed in the samples, which are characteristic peaks of Ti 2p_{3/2} and
433 Ti 2p_{1/2} and confirms that Ti is present as Ti⁴⁺ species (Song et al., 2016). Compared to S-TiO₂, with
434 increased amounts of g-CN in the samples, a negative shift was observed in the prepared samples (Table
435 S3). This phenomenon can be attributed to the close electronic interaction between the used S-TiO₂ and
436 g-CN, which confirms compact sample formation. Additionally, the negative shift could be due to the
437 increased electron density on Ti from the sample formation.

438 3.2.5 Optical traits and photocatalytic mechanism

439 Based on the detailed morphological and chemical characterizations, STC3, STC4, STC5, and
 440 STC6 were the successful S-TiO₂/g-CN heterojunctions. Additionally, among the aforementioned
 441 samples, maximum photoactivity was observed by STC4 under both UV and visible light irradiance
 442 (see Fig. 1). Therefore, to further evaluate the photooxidation mechanism of STC4, PL analysis was
 443 performed, and the respective spectrum was compared with PL spectra of S-TiO₂ and g-CN (Fig. 6(a)).
 444 It is well established that the PL spectra of the semiconductors can infer valuable insights on the rate of
 445 photoinduced charge separation, while irradiated with light.



446 **Fig. 6.** (a) PL spectra, (b) UV-vis DRS (absorbance vs. wavelength), (c) UV-vis DRS (F(R) vs.
 447 wavelength), (d) converted Kubelka-Munk function vs. energy of absorbed light of STC4, S-TiO₂, g-
 448 CN, and NP400.

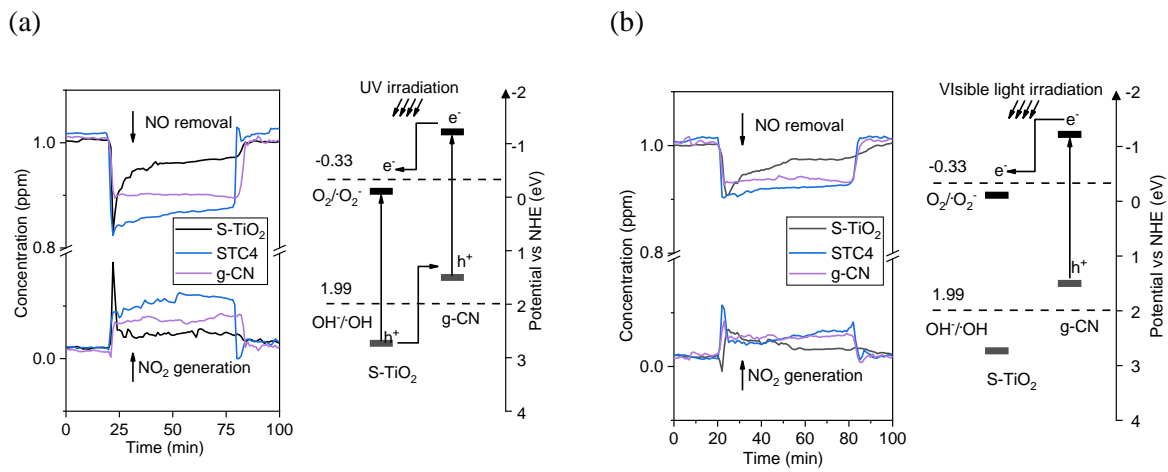
449 The prepared g-CN displayed a strong quasi-asymmetric PL spectrum, and the maximum peak was
 450 found around ~400 nm. This peak corresponds to the recombination of photoinduced e^-/h^+ pairs
 451 generated within the g-CN (Wang et al., 2013; Yu et al., 2013). Based on Fig. 6(a), compared to g-CN,
 452 STC4 showed much lower PL intensity at a similar peak position. Normally, lower PL intensity means
 453 a lower rate of recombination (Troppová et al., 2018). Moreover, the minimal recombination rate of S-
 454 TiO_2 is understandable as in the visible light spectrum, light absorption by S- TiO_2 was also minimal.

455 Additionally, for the determination of the optical energy bandgap (E_g) of STC4, UV-vis DRS was
 456 employed, and the light absorption data was reported in the UV-Vis range. By using the extrapolation
 457 technique on the modified Kubelka-Munk function, E_g for STC4 was determined, and the findings are
 458 depicted in Fig 6(b – d) (Giannakopoulou et al., 2017). Fig. 6(b) and (c) illustrate the absorbance spectra
 459 and the pattern of $F(R)$ function of the as-prepared superior sample (STC4), and the results were
 460 compared with g-CN, S- TiO_2 , and commercially available NP400. From Fig. 6(b), the prepared STC4
 461 showed superior light absorption in the UV region compared to g-CN hence showed significant
 462 improvement in removing NO_x under UV light irradiation. On the other hand, compared to NP400 and
 463 S- TiO_2 , STC4 showed expanded light absorption beyond the UV range. A superposition of the
 464 material's light absorption spectra was observed, which also confirmed successful heterojunction.
 465 Similar attributes were reported for the photocatalysts synthesized by heterojunction of commercially
 466 available titania and nitrogen enriched precursors like (melamine, urea etc.) (Song et al., 2016; Papailias
 467 et al., 2017). Additionally, Fig. 6(d) shows the $(F(R)hv)^n$ ($n = 1/2$ and 2) vs. hv graph of the samples,
 468 and the energy bandgaps were determined through the extrapolation of the corresponding graphs. The
 469 energy bandgaps were determined as 2.60, 2.65, 2.83, and 3.10 eV for g-CN, STC4, S- TiO_2 , and NP400,
 470 respectively. Besides, the CB and VB edge of S- TiO_2 and g-CN was estimated by using the following
 471 equations (Zhou et al., 2019),

$$472 \quad E_{CB} = E_{VB} - E_g \quad (9)$$

$$473 \quad E_{VB} = X - E_0 + 0.5 E_g \quad (10)$$

474 Here, E_{CB} and E_{VB} represents the CB and VB edge of the semiconductor. Additionally, X and E_0
 475 represents the average electronegativity and the energy of a free electrons next to NHE.



476 **Fig. 7.** Proposed photooxidation scheme for NO_x using $\text{S-TiO}_2/\text{g-CN}$ under UV (a) and visible light (b)
 477 irradiation.

478 Based on the relevant studies, for anatase TiO_2 and g-CN , the value of electronegativity can be
 479 assigned as 5.81, and 4.64 eV, respectively, and E_0 can be taken as 4.5 eV (Zhou et al., 2019). Hence,
 480 the E_{VB} of S-TiO_2 and g-CN can be estimated as +2.73 eV and +1.5 eV, respectively; also, the
 481 corresponding E_{CB} will be -0.11 eV and -1.22 eV. According to the above discussion, the primary factor
 482 that has been contributed towards the superior photoactivity of STC4 was the reduced energy bandgap
 483 due to the successful formation of heterojunction. Interestingly, despite having reduced S_{BET} , STC4
 484 ($33.69 \text{ m}^2/\text{g}$) showed extended NO_x oxidation compared to S-TiO_2 ($51.69 \text{ m}^2/\text{g}$). The possible reason
 485 for negligible photoactivity of the prepared S-TiO_2 could be the result of a greater amount of
 486 adventitious carbon or Sp^2 hybridized defects, as explained with EDX and XPS analysis. On the other
 487 hand, the reduced PL intensity of STC4 compared to g-CN confirms the reduced recombination rate;
 488 hence, enhanced photoactivity. Based on the assessment of the optical traits of the prepared STC4 , the
 489 photocatalytic reaction mechanism can be proposed as shown in Fig. 7.

490 Under UV irradiance, both S-TiO_2 and g-CN within the as-synthesized sample get excited, and the
 491 photoinduced e^- transfer from the VB towards the CB of the respective components (Troppová et al.,
 492 2018). Hence, the photoinduced e^-/h^+ pair gets generated, as shown in Fig 7(a). Following the charge

493 separation, the e^- generated on the CB of g-CN favors transferring towards the CB of S-TiO₂ due to less
 494 negative CB edge (Song et al., 2016). On the other hand, the h^+ generated on the VB of S-TiO₂ migrates
 495 towards the less positive VB of g-CN (Papailias et al., 2017). Consequently, the lifetime of the
 496 photoinduced e^-/h^+ pairs gets increased, leading to superior photoactivity. From the inhibited PL
 497 spectrum of the as-synthesized STC4, it is evident that the photoinduced charge separation rate got
 498 augmented, leading to an enhanced lifetime of e^-/h^+ pairs (Ma et al., 2016). Later, the photoinduced e^-
 499 can readily reduce the adjacent O₂ into $\cdot O_2^-$, while the h^+ on the VB reacts with adsorbed H₂O and
 500 generates $\cdot OH$ radicals (Ma et al., 2016; Papailias et al., 2017). It is well established that the active
 501 species $\cdot O_2^-$ and $\cdot OH$ are the two significant radicals that can promptly oxidize NO_x into more neutral
 502 nitrate salts. However, when STC4 was irradiated under visible light, only g-CN became excited, and
 503 the photoinduced e^-/h^+ pairs were generated on the respective CB and VB (Ma et al., 2016). Similarly,
 504 due to the less negative position of the CB of S-TiO₂, the e^- from g-CN migrates to the S-TiO₂ and
 505 restrain the rate of charge recombination. Later, the photoinduced e^-/h^+ pairs can readily produce the
 506 active species ($\cdot O_2^-$ and $\cdot OH$) for the effective photooxidation of NO_x. Hence, under UV and visible light
 507 irradiation the dominant photocatalytic reactions could be as follows,



513 One interesting phenomenon observed during the Photooxidation of NO under UV/visible light
 514 was an enhancement in the NO₃⁻ selectivity of the prepared heterojunctions compared to the precursor S-
 515 TiO₂. Under UV and visible light irradiation, the NO₃⁻ selectivity of S-TiO₂ was reported as 41.90% (see
 516 Table S1) and 42.20% (see Table S2), respectively, which was substantially increased for STC4 and
 517 were found as 56.66% and 56.76%, respectively. Perhaps the incorporation of g-CN played a vital role
 518 due to the favorable V_{CB}/V_{VB} position. Compared to S-TiO₂ (-0.11 eV), the CB edge of g-CN (-1.22 eV)

519 was found more negative; hence, the reduction potential of the photogenerated e^- in STC4 was higher
520 to reduce O_2 into $\cdot O_2^-$ (Wang et al., 2016). On the other hand, less positive V_{VB} of g-CN (+1.5 eV) in
521 STC4 reduced the generation rate of $\cdot OH$, as the redox potential of $OH^-/\cdot OH$ (2.37 eV) is higher (Wang
522 et al., 2016; Hossain et al., 2021). Therefore, it can be presumed that the $\cdot O_2^-$ played the primary role in
523 the complete photooxidation of NO. Moreover, the enhanced NO_3^- selectivity of STC4 depicted that
524 compared to $\cdot OH$, the oxidation potential of $\cdot O_2^-$ for NO to NO_3^- conversion is higher, explaining the
525 significant NO_2 generation during standalone use of S-TiO₂. The aforementioned phenomena are
526 coherent with most of the TiO₂/g-CN based atmospheric NO removal studies, where $\cdot O_2^-$ was claimed
527 to be the primary reactive species (Ma et al., 2016; Wang et al., 2016; Papailias et al., 2017; Jiang et al.,
528 2018).

529 **4 Conclusion**

530 The objectives of the study were to prepare TiO₂/g-CN samples by utilizing sludge generated TiO₂
531 and increase the removal of atmospheric NO_x under UV and visible light irradiation. Successful
532 preparation of S-TiO₂/g-CN samples was confirmed through XRD, SEM, TEM, FT-IR, and XPS
533 analysis. Detailed surface characterizations revealed, improved dispersion of S-TiO₂, while sampled
534 with g-CN. Moreover, g-CN had negligible impacts on the crystal structure and size of the utilized S-
535 TiO₂ precursor. Among the successfully prepared samples, maximum NO_x removal was reported for
536 STC4, wherein the precursor mix of the melamine content was 3 times higher than the S-TiO₂. UV-vis
537 DRS conducted on STC4 revealed that the energy bandgap was reduced to 2.65 eV from 3.20 eV from
538 S-TiO₂. Hence, the absorption band of STC4 was significantly increased in both UV and visible regions.
539 Besides, PL analysis showed inhibited recombination rate of generated e^-/h^+ pairs, which is favorable
540 to readily generate $\cdot O_2^-$ and $\cdot OH$ radicals for photodegradation of NO_x. Finally, based on the evaluated
541 optical traits, a simplified schematic is proposed for NO_x removal under UV and visible light irradiation.
542 By considering NO_3^- selectivity, the NO_x removal under UV light by S-TiO₂ was reported only 2.45%,
543 which was increased in STC4 and found as 7.2%. Again, under visible light, the NO_x removal of 1.75%
544 by S-TiO₂ was increased to 5.4% by utilizing STC4. However, compared to commercially available
545 NP400, the NO_x removal of STC4 was found inferior, but this synthesis route of TiO₂/g-CN

546 heterojunction has significant potential to promote sustainability through the utilization of unwanted
547 sludge from the water treatment plant and reduction in synthesis cost of the photocatalysts.

548 **Author Contributions**

549 Conceptualization and methodology, S.M.H., H.P. and J.S.M.; data analysis, investigation and data
550 curation, S.M.H., J.S.M. and I.R.; original draft writing, S.M.H.; review and editing, L.T., H.-J.K., Y.-
551 S.J. and H.K.S.; supervision, J.-H.K., H.K.S and Y.-S.J. All authors provided critical feedback and
552 helped shape the research, analysis and manuscript.

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