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Highly stable gold nanolayer membrane for efficient solar water evaporation under a harsh environment

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Highlights

- A novel strategy to fabricate a scalable gold nanolayer membrane is reported.
- The modified membrane demonstrates robust chemical and mechanical stability.
- High salt ion rejection of seawater was demonstrated.
- The gold nanolayer membrane showed high reusability and stability under wide pH ranges.

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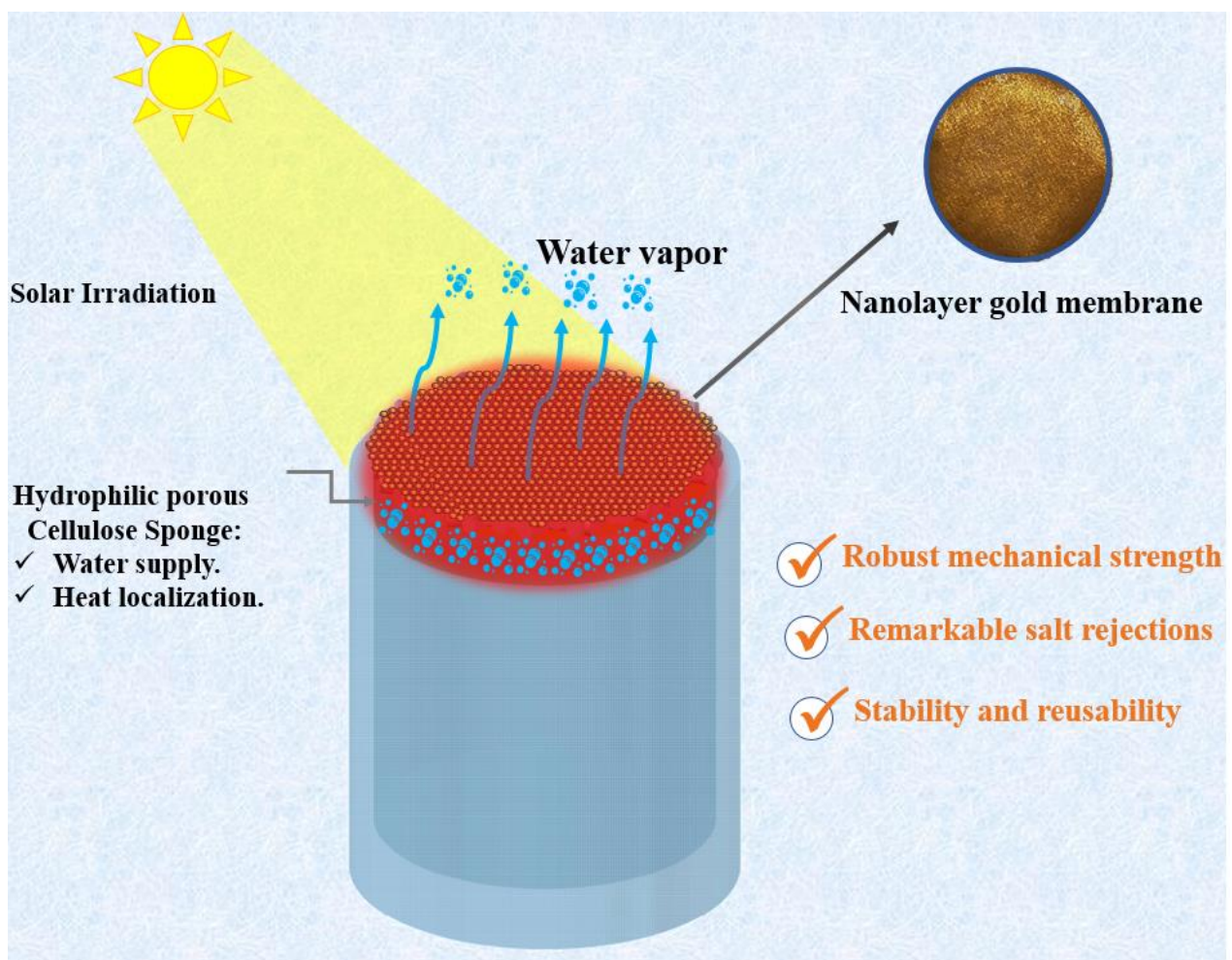
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38 **Graphical abstract**



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45 **Abstract**

46 Interfacial solar water evaporation has attracted tremendous attention for sunlight harvesting
47 for water purification. However, salt formation and stability of the photothermal materials
48 (PTMs) remain a challenge that need addressing before bringing this technology to real-world
49 applications. In this work, a nanoscale thin film of gold (Au) on a polytetrafluoroethylene
50 (PTFE) membrane has been prepared using a magnetic sputtering technique. The fabricated
51 membrane displays a robust mechanical strength and chemical stability arising from the
52 adhesiveness of the thin film Au nanolayer on the PTFE membrane as well as the chemical
53 inertness of the noble metal PTM. The Au nanolayer/PTFE membrane with cellulose sponge
54 substrate resulted in an evaporation rate of $0.88 \text{ kg m}^{-2} \text{ h}^{-1}$ under 1 sun intensity. Remarkable
55 salt ion rejection of 99.9% has been obtained, meeting the required standard for drinking water.
56 Moreover, the membrane exhibited excellent stability and reusability in natural seawater and
57 high salinity brine (150 g/L) and even in severe conditions (acidic, basic, and oxidized). No
58 noticeable salt formation was observed on the evaporator surface after the tests. These findings
59 reveal promising prospects for using a magnetron sputtering technique to fabricate a stable
60 photothermal membrane for seawater and high salinity brine desalination.

61

62 **Keywords:** Gold nanolayer membrane; photothermal conversion; solar water evaporation,
63 desalination; reusability; harsh environment.

64

65 1. INTRODUCTION

66 In recent years, solar water evaporation (SWE) (also called solar steam generation) has
67 attracted increased interest as a passive, decentralized, and sustainable desalination technology
68 for use in inland and off-grid areas where freshwater is scarce and sunlight is abundant (Li et
69 al., 2018a; Li et al., 2018b; Chen et al., 2021a). This is in contrast with the mainstream
70 centralized desalination technologies like reverse osmosis that require big infrastructures and
71 a high amount of fossil fuel-based energy sources (Ghafurian et al., 2020b; Li et al., 2021).
72 Small-scale SWE system can generate over $10 \text{ kg m}^{-2} \text{ day}^{-1}$ of clean water and can be scaled
73 up due to simplicity in setup and design (Li et al., 2020). In this study, we refer to SWE in the
74 context of interfacial solar water evaporation. The technology is based on photothermal
75 materials (PTMs) that absorb sunlight and convert it into heat at the air/water interface, which
76 effectively vaporizes water at the solar absorber surface, then condenses it to generate clean
77 water (Tao et al., 2018; Wu et al., 2019b; Ibrahim et al., 2020; Shi et al., 2020; Zhao et al.,
78 2020c; Liu et al., 2021; Wang et al., 2021). However, salt crystallization on the evaporation
79 area is the main issue that needs to be addressed to implement this technology for practical
80 applications. This results in poor light absorption, which reduces the overall evaporation
81 surface by blocking the water vapor channel, hence decreasing the water evaporation rate
82 (Chen et al., 2021b; Ibrahim et al., 2021b; Peng et al., 2021). To date, various strategies have
83 been proposed to overcome this issue by designing various PTMs including porous polymer
84 materials (He et al., 2020; Wang et al., 2020), bioinspired microstructure materials (Huang et
85 al., 2020a), Janus materials (Xu et al., 2018; Chen et al., 2020; Peng et al., 2021), and
86 hydrophobic materials (Wei et al., 2021). However, the chemical stability and resistivity to
87 harsh chemical conditions remains a question that must be addressed to ensure long-term
88 stability.

89 Among the reported PTMs for SWE, noble metals such as gold nanoparticles have been used
90 as firstly reported by Halas et al. in 2013 (Neumann et al., 2013), and have since attracted broad
91 interest, which led to an increase in research activities related to SWE. Noble metals are known
92 to exhibit excellent chemical stabilities under harsh environments such as for evaporation of
93 source water with highly acidic and basic conditions (Hammer and Norskov, 1995; Badawy et
94 al., 2010; Kiriarachchi et al., 2018). Previous studies on noble metal-based PTMs utilized either
95 complicated or small-scale synthesis strategies (Zhang et al., 2019). Furthermore, the
96 mechanical strength of their fabricated PTMs remains a challenge, hence hindering their
97 practical implementation. Thus, a scalable strategy to allow the fabrication of nanoscale noble
98 metal-based PTMs with excellent mechanical stability is essential for practical application of
99 SWE.

100 Magnetron sputtering (MS) is a promising technique that enables the deposition of nanoscale
101 thin film on various substrates (Grammatikopoulos et al., 2013). The method provides a cost-
102 effective thin film coating at relatively low temperatures and can produce uniform thin film
103 layers on large substrate areas with excellent adhesion (Greene, 2017; Barbosa et al., 2019). In
104 the present study, a well-designed Au nanolayer on PTFE membrane surface fabricated by an
105 MS technique for SWE application is demonstrated. Benefitting from the strong adhesion of
106 the Au nanolayer on the PTFE membrane surface, the current approach results in a membrane
107 with robust mechanical strength and excellent chemical stability even under harsh
108 environments. The light absorption capability of our fabricated membrane covers the whole
109 solar spectrum region. The designed solar evaporator of Au nanolayer-PTFE on cellulose
110 sponge resulted in excellent salt ion rejection, meeting the required standard for drinking water.
111 Furthermore, exceptional reusability and stability are demonstrated in natural seawater and
112 harsh chemical conditions. Our findings reveal the novelty of magnetic sputtering methods for

113 scaling up highly stable PTMs with excellent salt resistance for long-term durability solar
114 evaporators.

115 **2. Materials and methods**

116 **2.1 Materials**

117 Gold targets (Au >99.9% purity) were purchased from Kurt J. Lesker company. A commercial
118 polytetrafluoroethylene (PTFE, 0.45 μm pore size) membrane was obtained from Ningbo and
119 was used as a substrate for the formation of nanoscale thin film Au layer. A commercial
120 cellulose sponge was purchased from a local supermarket, and was utilized as the insulator and
121 floating support for the Au nanolayer membrane. Potassium hydroxide (KOH) was purchased
122 from Merck; hydrochloric acid was bought from Chem-Supply, and sulfuric acid and nitric
123 acid were received from Sigma Aldrich.

124 **2.2. Fabrication of Au nanolayer membrane**

125 Nanoscale Au thin films on PTFE membranes were formed using a magnetron sputtering
126 system equipped with a 75 mm Au diameter target and power supply (AJA DCXS-750,
127 Scituate, MA, USA). The high target purity was 99.99%. The distance between the target and
128 the substrate (PTFE membrane) was set at 70 mm. The deposition system was equipped with
129 rotary and cryogenic pumps and a controlled gas introduction system. A base pressure of 1×10^{-4}
130 Pa was attained in the chamber before the deposition. Argon gas was introduced into the
131 chamber prior to the deposition process. The deposition pressure (1.0 Pa) was set independently
132 of the gas flow by adjusting a throttle valve. DC power was set at 100 W. The deposition time
133 was 4 min, and the thickness of deposition nanolayer thin film was 50 nm (Ghafurian et al.,
134 2020a) and the change in the mass loading of the PTFE membrane after the gold nanolayer
135 deposition was $45 \mu\text{g}/\text{cm}^2$.

136 **2.3 Characterization**

137 The structure and the morphology of pristine PTFE and Au nanolayer membrane was
138 investigated using field emission scanning electron microscopy (FE-SEM, Zeiss Supra 55VP)
139 at an accelerated voltage of 10 kV. The surface wetting ability of the samples was investigated
140 by Theta Lite 100 (Attension) using a sessile drop method. A volume of 10 μ L water droplet
141 was vertically dropped on the substrate surface. The light absorption characteristics of the
142 pristine PTFE and Au nanolayer membranes were measured at a wavelength range of 300-2500
143 nm (i.e., ultraviolet-visible-near-infrared regions) using a spectrophotometer equipped with an
144 integrating sphere (950 PerkinElmer Lambda). Infrared (IR) images and surface temperatures
145 were recorded using an infrared camera (FLIR, E6). The concentrations of the positively
146 charged ions (K^+ , Na^+ , Mg^{2+} , and Ca^{2+}) in seawater before and after purification were measured
147 using an inductively coupled plasma mass spectrometer (ICP-MS, Agilent 7900), while the
148 negatively charged ions (NO_3^- , Cl^- , and SO_4^{2-}) were measured using ion chromatography (IC)
149 (Thermo Fisher, Australia).

150

151 **2.4 Solar water evaporation test**

152 The SWE set-up consisted of a solar simulator (PLS-SXE300) with a standard AM 1.5G filter,
153 a precision analytical weighing balance (IC-PX84/E, Ohaus, 0.1 mg accuracy) connected to a
154 computer, an infrared camera (FLIR E6), and Type-K thermocouples (RS PRO: RS172TK)
155 (**Fig. S1**). The Au/PTFE membrane solar evaporator (diameter = 4.5 cm) placed on top of a
156 cellulose sponge with a thickness of 1.2 cm (wet) (**Fig. S2**), was floated on a 100 ml beaker
157 filled with 80 mL deionized water (DIW). The effect of the cellulose sponge was investigated
158 at different thicknesses (0.6, 1.2 cm and 1.6 cm). During the experiment, simulated sunlight at
159 an intensity of 1 sun (1 kW m^{-2}) was illuminated on the solar evaporator which was placed on
160 an electronic balance. The change in weight with respect to time was continuously recorded at
161 4 min interval, with each evaporation experiment lasting for 1 h. Changes in membrane surface

162 temperature were measured by an IR camera at various intervals, and a thermocouple was used
163 to measure the bulk water temperature changes. SWE experiments at dark condition (without
164 simulated light) were also carried out. Pristine PTFE membrane without gold thin film coating
165 was also checked for its SWE performance, which served as the control for comparison. All
166 experiments were carried out at room temperature of 22 ± 2 °C and humidity of ~52.5%.

167

168 **2.5 Solar water desalination and stability test under harsh environment**

169 The potential of our Au nanolayer membrane for water desalination was evaluated by treating
170 a real seawater collected from Cronulla Beach, Sydney, Australia and similar protocols for
171 SWE tests were followed. In addition, a high salinity NaCl brine solution (at 150 g/L) was also
172 prepared. The stability of the gold PTMs was evaluated under a harsh environment utilizing
173 acidic (0.1 M H₂SO₄), basic (0.1 M KOH), and oxidizing conditions (0.1 M HNO₃), with
174 respective ionic strengths (*I*) of 0.3 M, 0.1 M, and 0.1 M (Zhao et al., 2020b). During the water
175 purification tests, the evaporated water was collected and analyzed using ICP and IC techniques
176 to measure the ions of the feedwater and the collected water. In addition, cycling stability tests
177 (10 cycles) were demonstrated.

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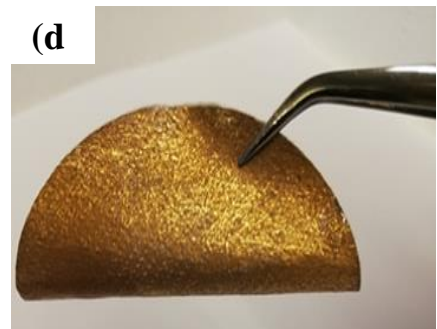
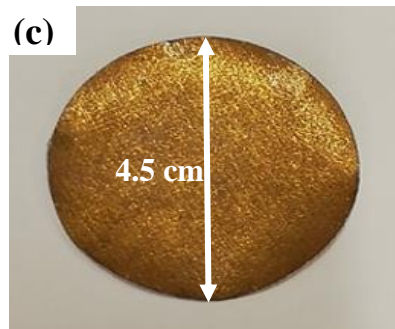
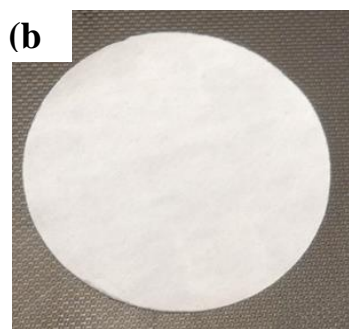
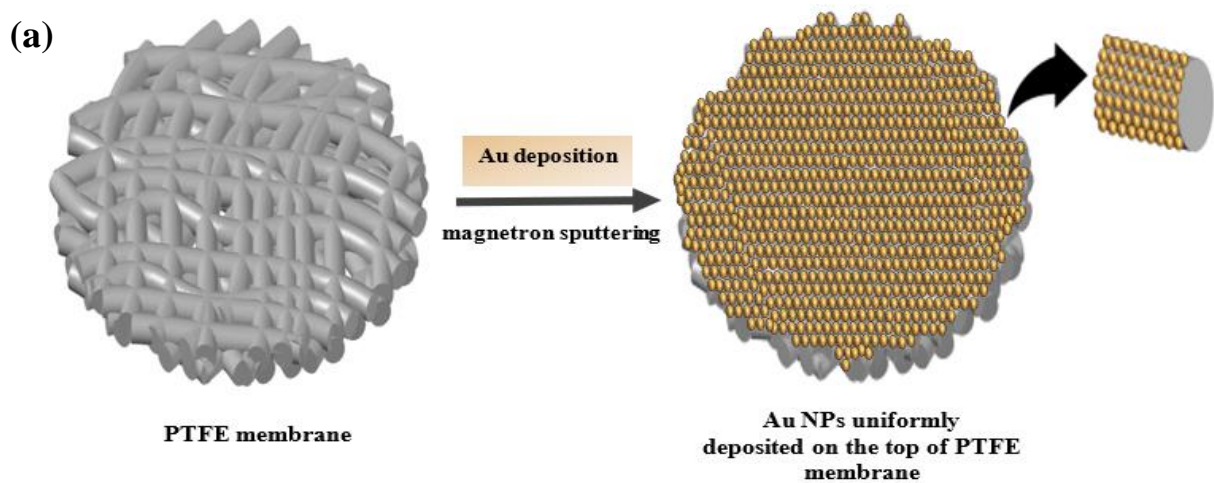
179 **3. Results and discussion**

180 **3.1 Characterization of Au nanolayer-PTFE membrane**

181 **Fig 1.** shows the pristine PTFE membrane and the fabricated Au nanolayer films coated PTFE
182 membranes. A magnetron sputtering process was used to uniformly coat thin films of gold onto
183 PTFE membranes. The duration of the coating process was 4 min with film thickness of 50 nm.
184 This type of coating process ensures quick, highly scalable, and uniform deposition of materials
185 in a single step without the use of chemicals making it a practical fabrication approach. This
186 technology enables the fabrication of Au nanolayer membranes in large areas, which is vital

187 for practical applications (**Fig. S3**). A PTFE membrane was chosen as the substrate due to its
188 good mechanical strength, flexibility, chemical stability and high-temperature resistance
189 (Chang et al., 2018). During the magnetron sputtering process, strong physisorption was
190 expected between the partially ionized Au particles and the modified PTFE membrane surface
191 which led to their good adhesion with each other (Pinho and Piedade, 2013). As illustrated in
192 **Fig. 1a-c**, after deposition of the Au nanolayer, the circular white membrane top surface (inset
193 of Fig 1a) acquired a gold color, signifying the successful coating of the Au nanolayer. The
194 coated membranes displayed strong adherence of the thin-layer Au to the PTFE membrane.
195 The modified membranes can easily be folded and returned to their original state without
196 detachment of the coating or membrane damage (**Fig. 1d**). These features are beneficial for the
197 development of scalable and effective PTM for solar water evaporators.

198



199

200 **Fig. 1** (a) Schematic illustration of gold deposition on a pristine PTFE membrane substrate.
201 Images showing the (b) pristine PTFE membrane, (c) gold-coated PTFE membrane, and (d)
202 the flexibility of the Au-PTFE membrane.

203

204 The structural morphology, elemental compositions, surface analysis and wettability of the
205 fabricated Au nanolayer were examined. SEM images reveal that the surface morphology of
206 the pristine PTFE membrane contains unequal-length extended holes with open and distributed
207 channels (**Fig. S4**). After gold deposition, a homogeneously distributed gold layer with a
208 thickness of ~50 nm was observed on the PTFE fiber surfaces (**Fig 2a** and **2b**). The surface
209 structure and porosity of the PTFE were maintained after the coating process. This structure is
210 favourable for fast and extensive vapor diffusion and can facilitate good water evaporation
211 performance (Lu et al., 2019). Energy-dispersive X-ray spectroscopy (EDS) analysis (**Fig. S5**)
212 and elemental mapping (**Fig. 2c**) confirm the existence and the distribution of Au nanolayer
213 homogeneously on the PTFE membrane surface. Additionally, XPS analysis was conducted to
214 investigate the elemental composition and electronic structure of the Au-PTFE membrane
215 surface. The XPS survey spectrum confirms the presence of Au, C, F and O (**Fig. 2d**). High-
216 resolution XPS analysis revealed narrow doublet peaks of Au 4f_{5/2} (84.12 eV) and Au 4f_{7/2}
217 (87.75 eV), separated by 5 eV (**Fig. 2e**). The peaks at 680 eV and 284.59 eV match the F 1s,
218 and C1s, respectively (Kiriarachchi et al., 2018; Saravanakumar et al., 2020). The weak peaks
219 of F 1s, and C 1s suggest the successful Au nanolayer uniform coating on the PTFE membrane
220 surface.

221 The surface wettability plays a vital role in the anti-salt properties of PTMs for solar seawater
222 desalination (Ibrahim et al., 2021a). The dynamic wetting of water droplets on the pristine
223 PTFE and gold deposited PTFE membrane were measured using a contact angle measurement
224 (tensiometer) apparatus. The pristine PTFE membrane surface showed a water contact angle

225 of $145 \pm 3^\circ$, indicating that it is almost superhydrophobic (**Fig. 2f**). The thin layer of gold on
226 the PTFE surface has drastically reduced its wettability, with a water contact angle of $100 \pm 4^\circ$
227 (**Fig. 2g**). This hydrophobic behavior supported the self-floating ability of the membrane on
228 water. In addition, it can act as an anti-salt property by repelling the salt ion crystallization on
229 the evaporator surface, which is beneficial for the long-term durability of solar desalination.

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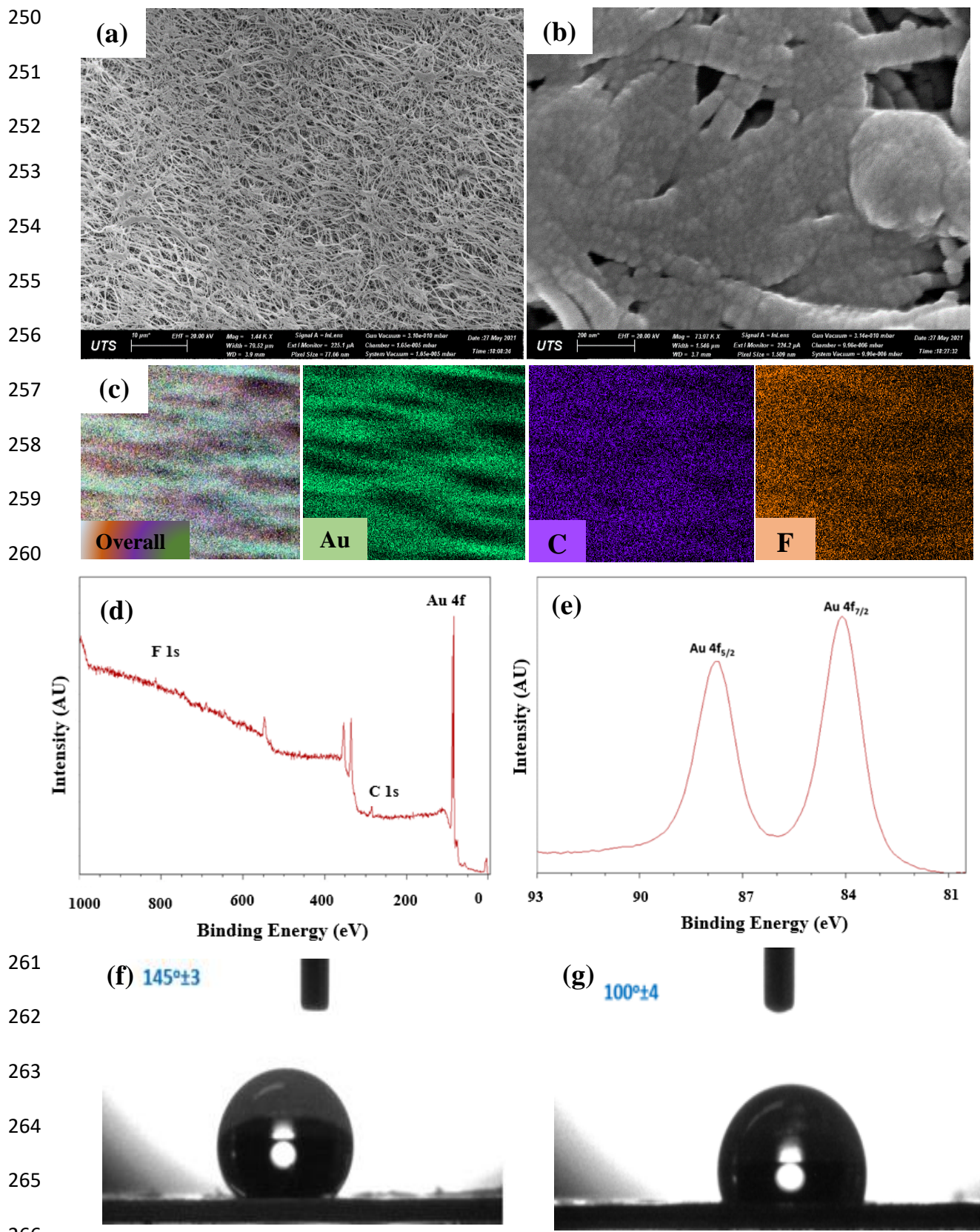
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267 **Fig 2.** (a) Low and (b) high magnification SEM images of the Au/PTFE photothermal
268 membrane; (c) EDS mapping results confirming the distribution of Au nanolayer on the PTFE

269 membrane surface; **(d)** X-ray photoelectron spectra (XPS) survey of Au-PTFE membrane; **(e)**
270 High resolution XPS spectra confirming the existence of the Au 4f; and surface wettability test
271 of the PTFE membrane **(f)** and Au nanolayer membrane **(g)** displaying hydrophobic properties,
272 and reduction in hydrophobicity being observed after the gold nanolayer deposition.

273

274 **3.2 Chemical and mechanical stability of Au nanolayer-PTFE membrane**

275 PTMs with chemical stability and robust mechanical strength are essential for practical
276 applications in SWE. The tensile strength and percentage strain results indicated that for
277 pristine PTFE membrane, i.e., without the thin film gold coating, only 6.71 MPa and 33.9%,
278 were obtained, respectively. After the coating process, the values have increased to 8.09 MPa,
279 and 36.48%, respectively. This phenomenon can be explained by the strong adhesion of the
280 coated Au nanolayer to the membrane, enhancing its mechanical properties. The gold did not
281 detach or fall off from the surface of PTFE membrane. Furthermore, the chemical stability was
282 assessed by heating the Au-PTFE membrane under acidic, basic, and oxidizing solution
283 conditions at a temperature of 40 °C for 30 hours, and then subjected the Au-PTFE membranes
284 under sonication for 40 minutes. The EDS analysis revealed that the gold on the treated
285 membranes maintained similar composition as with the original samples (**Fig. S6 a-d**) even
286 after exposure to these harsh environments. This signifies that the magnetic sputtering method
287 could be a superior technique to fabricate Au nanolayer-based solar absorber compared to those
288 techniques reported in literature, such as electrospinning (Wu et al., 2019a) and dip-coating
289 (Zhu et al., 2018). The obtained results demonstrated that the magnetic sputtering method
290 resulted in nanolayer gold formation with remarkable adhesiveness on PTFE membrane. This
291 could be a promising strategy for the fabrication of a stable photothermal membrane for solar-
292 driven water purification.

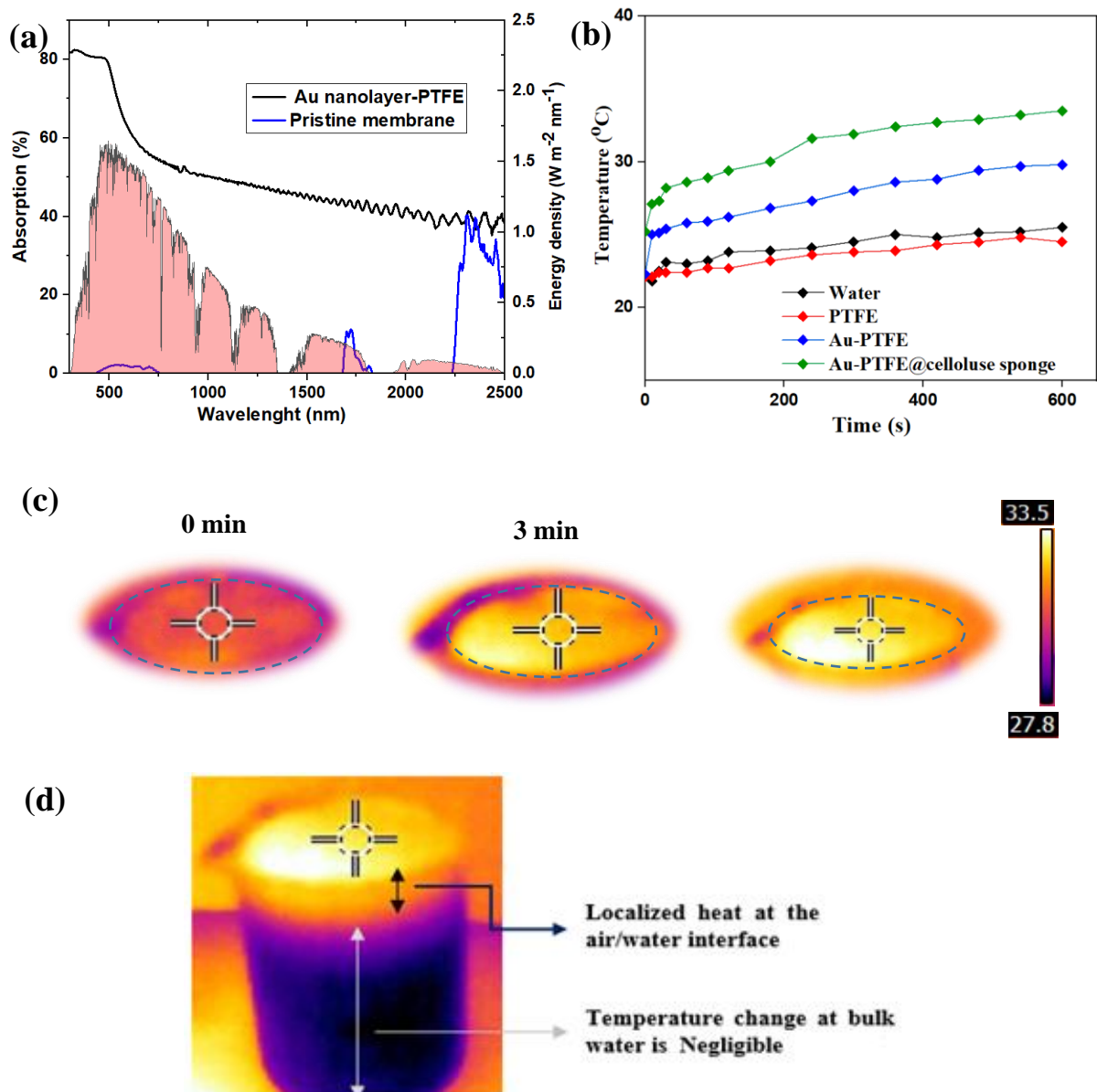
293 **3.3 Optical characteristics and solar-thermal conversion of gold nanolayer membrane**

294 The structure-dependent light absorption characteristics of the pristine PTFE and the designed
295 Au nanolayer-PTFE membranes were evaluated using a UV-Vis-NIR spectrometer in the
296 wavelength range of 300-2500 nm (**Fig. 3a**). The inset image in **Fig. 4a** shows the reference
297 solar spectrum (at AM 1.5 G) (Politano et al., 2017). The pristine PTFE displayed weak light
298 absorption characteristics in the whole solar spectrum regions. In contrast, the coated PTFE
299 membrane with Au nanolayer shows a much higher light absorption covering the whole solar
300 spectrum region, with light absorption of ~ 80% in the UV region (300-400 nm), ~ 55-80% in
301 the visible region (400-700 nm), and ~ 40-55% in the NIR region (700-2500 nm). Along with
302 good light absorption characteristics of the prepared gold nanolayer PTFE membrane, surface
303 plasmon resonance (SPR) effect from the gold nanolayer helps to generate heat in the PTMs
304 for the effective water evaporation to occur. Plasmonic metals such as gold, due to the presence
305 of high concentration of free carriers, enable free-carrier absorption (Zhao et al., 2020a). The
306 plasmon-induced heat generation occurs when Au nanolayer is illuminated at their resonant
307 wavelengths, which causes oscillation of the electron cloud of free electrons. The decay of the
308 hot electrons through electron-electron scattering creates and redistributes the hot electron
309 energy in the form of heat, which can rapidly increase the localized surface temperature of
310 plasmonic Au (Kim et al.; Linic et al., 2015). Furthermore, the rough surface of the PTFE
311 membrane and the holes with open and distributed channels play an effective role in reducing
312 light reflection and can trap the photon capture (Lu et al., 2019). Hence, the absorption is
313 considered as angle-independent due to the potential multiple scattering between the fibrous
314 membrane (Zhu et al., 2018). This enhancement in the optical performance is beneficial for
315 solar-thermal conversion for water evaporation application. The optical performance of our Au
316 nanolayer-PTFE membrane is comparable with some reported optical characteristics of noble
317 metal PTMs in SWE (Chen et al., 2018; Kiriarachchi et al., 2018; Huang et al., 2020b).

318 After the optical properties test, we designed a simple solar evaporator consisting of a Au
319 nanolayer membrane (Au-PTFE) placed on top of a hydrophilic porous cellulose sponge (CS)
320 (referred to as Au-PTFE@CS). The added CS has dual-function roles; (1) acting as thermal
321 insulator where it can localize the heat at the air/water interface, and can reduce the downward
322 heat losses, and, (2) it helps in supplying water to the top hydrophobic photothermal Au-PTFE.
323 This simple combination design allowed fast and continuous water supply from the bulk water
324 to the top hydrophobic Au nanolayer-PTFE membrane. Meanwhile, the hydrophobic Au-PTFE
325 can repeal the salt formation resulting in excellent salt ions rejections with long-term durability
326 and stable performance. For comparison, a pristine PTFE membrane was examined as well as
327 a free-standing Au nanolayer membrane (without CS), and a DI water only condition.

328 The surface temperature change of the designed solar evaporator was monitored over time
329 using an IR camera under 1 sun (1 kW m^{-2}) illumination. As illustrated in **Fig. 3b** and **3c**, the
330 surface temperature of the pristine PTFE membrane only showed a slight increase in
331 temperature from 22.1 °C to 24.8 °C within 10 min. In contrast, the surface temperature of the
332 Au-PTFE@CS rapidly increased from an ambient temperature of 25.2 °C to reach 30 °C within
333 3 minutes of illumination and further rose to 33.2 °C after 10 min. In addition, without CS, the
334 surface of free-standing Au nanolayer membrane increased to 29.3 °C. The surface temperature
335 of the Au-PTFE with CS is higher compared to without CS demonstrating its excellent solar
336 light-to-heat conversion. One reason for this is due to CS support, which acts as thermal
337 insulator and heat localizer at the air-water interface. To further investigate the effectiveness
338 of the CS in localizing heat and reducing the heat transmittance to bulk water, we measured
339 the bulk water temperature using a thermocouple. The side-view image using IR camera was
340 also used to visualize the heat distribution on the side of the solar evaporator. The thermal
341 images of Au-PTFE with CS revealed that heat is localized with low transmission to the bulk
342 water (**Fig. 3d**). Furthermore, the bulk water temperature change was relatively low with an

343 increase in temperature from 25.6 °C to 26.5 °C (i.e., 0.9 °C difference). Meanwhile, without
 344 CS, the bulk water temperature increased from 25.6 °C to 30.7 °C, which indicated the fast
 345 conduction of heat to the water. These observations demonstrate the effectiveness of adding
 346 thermal insulation in localizing the heat at the air-water interface and minimizing the downward
 347 heat losses, which is beneficial for solar water evaporation application.
 348



349
 350 **Fig 3. Solar absorption and solar-thermal conversion.** (a) Light absorption characteristics
 351 of the Au nanolayer membrane in the UV-vis and NIR regions. (b) The temperature changes

352 of the Au-PTFE/CS, pristine PTFE membrane, and water only at 1-sun irradiation. (c) Infrared
353 image of the heat distribution after surface exposure to 1 sun-irradiation from 0 to 10 minutes.
354 (d) Infrared photo confirming the capability of the device to localize heat at the air-water
355 interface.

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358 **3.4 Solar water evaporation performance**

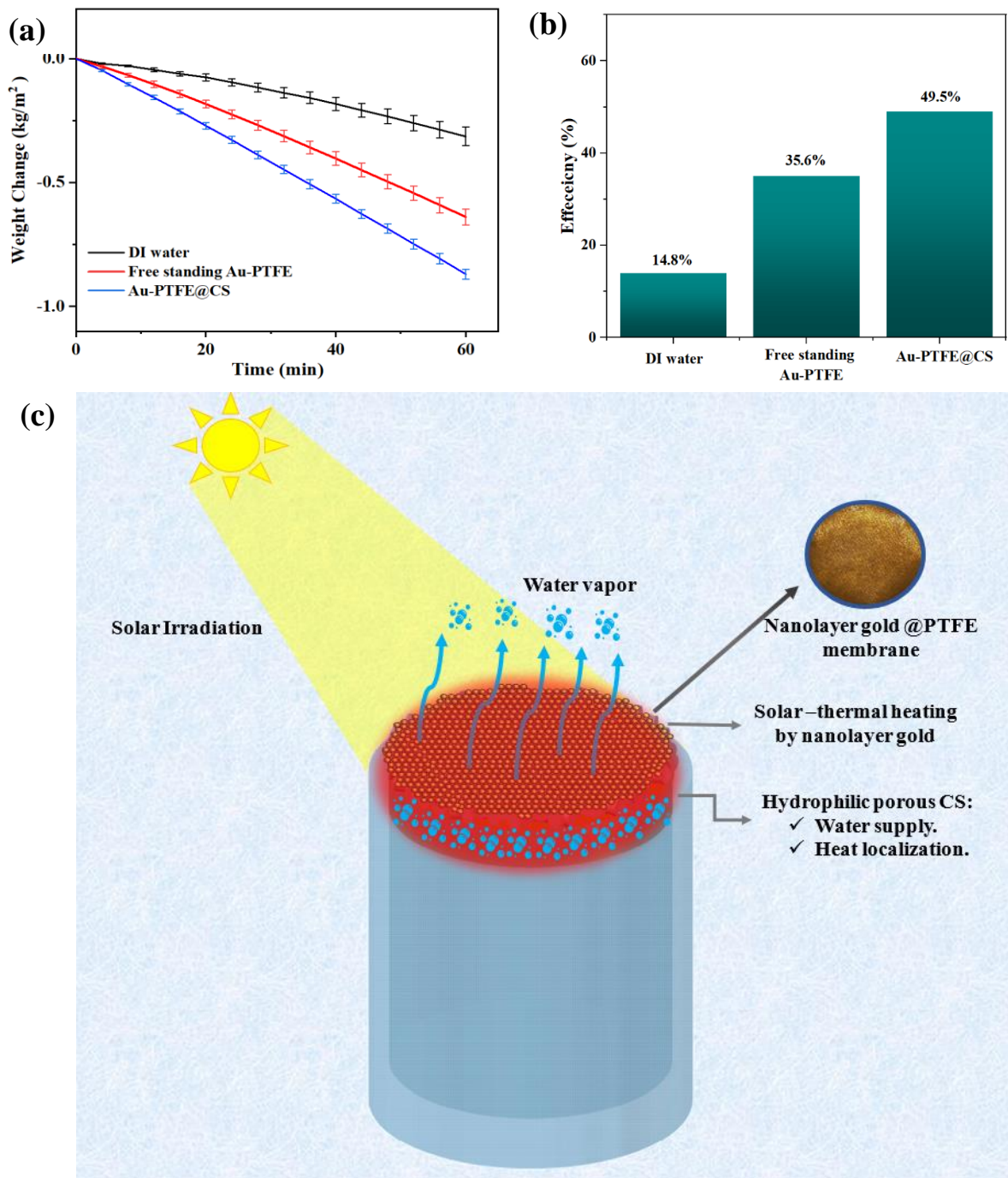
359 The solar water evaporation performance of the Au-PTFE/CS was investigated and compared
360 to those of DIW, and a freestanding Au-membrane (without CS support), under 1 sun solar
361 illumination. The solar heat-to-vapor efficiency (η) is calculated using the following equation
362 (Shan et al., 2019; Zhao et al., 2020d):

$$363 \quad \eta = \frac{\Delta m * He}{I * T} \times 100\%,$$

364 where Δm refers to $(m_i - m_o)$, m_i is the water mass change (kg/m^2) under solar light, and m_o is
365 the water mass change (kg/m^2) at dark condition (without sunlight). He is the total vaporization
366 enthalpy of the water-steam phase change ($\text{kJ}\cdot\text{kg}^{-1}$), I is the solar power energy density
367 (kW m^{-2}), and T (h) is the irradiation period. The evaporation rates at the dark condition were
368 0.081, 0.072, and 0.078 $\text{kg m}^{-2} \text{h}^{-1}$ for DIW, free-standing Au nanolayer-PTFE, and Au
369 nanolayer-PTFE/CS, respectively (**Fig. S7**).

370 Under 1 sun illumination, the DIW displayed an evaporation rate of 0.314 $\text{kg m}^{-2} \text{h}^{-1}$,
371 corresponding to an evaporation efficiency of 14.8%. The evaporation rate was significantly
372 increased after placing free-standing Au nanolayer membrane at the air/water interface to
373 deliver evaporation rate of 0.64 $\text{kg m}^{-2} \text{h}^{-1}$, corresponding to an efficiency of 35.6% which is
374 ~ 2 times that of pure water. The vapor generation arises from the absorption of the solar light
375 through the gold nanolayer which generates a sufficient heat at the air-water interface. It was
376 noticed that the evaporation rate was further improved by placing cellulose sponge below the

377 Au nanolayer membrane to achieve an evaporation rate of $0.88 \text{ kg m}^{-2} \text{ h}^{-1}$, corresponding to
378 an efficiency of $\sim 49.5\%$. **Fig. 4a and 4b** display the water evaporation rate and solar- heat-to-
379 vapor efficiency results in this work. The reason for such further increase is due to the presence
380 of underlying hydrophilic CS, which offers great heat confinement and provides efficient solar
381 heat utilization. In addition, it delivers sufficient and continuous water supply to the
382 hydrophobic Au nanolayer membrane. The spaces between the hydrophobic Au nanolayer
383 membranes facilitate vapor escaping. Meanwhile, without CS (free standing Au nanolayer),
384 some of the generated heat by the Au nanolayer membrane were transmitted to the bulk water
385 via heat conduction, resulting in lower water evaporation. **Fig. 4c** describes the designed device
386 and the evaporation process. The obtained evaporation rate in this study is comparable to some
387 reported studies in the literature (**Table S1**). The effect of the CS thickness on the water
388 evaporation was investigated to find the optimal thickness. At 0.6 cm CS thickness, it generated
389 a water evaporation rate of $0.82 \text{ kg m}^{-2} \text{ h}^{-1}$ at one sun intensity, while the 1.6 cm CS thickness
390 resulted to an evaporation rate of $0.86 \text{ kg m}^{-2} \text{ h}^{-1}$, which is similar to the 1.2 cm thickness.
391 Hence, we chose 1.2 cm cellulose sponge as the optimized thickness in our current study (see
392 **Fig. S8**).



393

394 **Fig 4.** (a) Weight change of DIW, free standing Au-PTFE, and Au-PTFE supported CS under
 395 1 sun intensity, (b) their corresponding solar conversion efficiency, and (c) schematic
 396 illustration of the solar conversion and vapour generation process of Au nanolayer membrane
 397 on CS based solar evaporator.

398

399 3.5 Solar desalination performance

400 The feasibility of the Au-PTFE membrane for solar water desalination was also investigated.
401 For the desalination tests, real seawater samples from Cronulla Beach, Sydney were used.
402 Interestingly, the Au nanolayer-PTFE membrane displayed a seawater evaporation rate of
403 $0.859 \text{ kg m}^{-2} \text{ h}^{-1}$, close to the evaporation rate of DI water of $0.88 \text{ kg m}^{-2} \text{ h}^{-1}$ (**Fig. S10**). In
404 order to investigate the capability of the evaporator system for generating potable water, the
405 evaporated water was condensed, collected and analyzed using ICP-MS and compared to the
406 original seawater. Schematic illustration of the device used for the water condensation and
407 collection is provided in **Fig. 5a and Fig. S9**. The results indicated that a high amount of the
408 salt ions Na^+ , K^+ , Ca^{2+} , and Mg^{2+} in the purified water were significantly decreased to 7.79,
409 0.60, 0.62, and 0.86 mg/L, respectively compared to their respective initial concentrations of
410 8750, 1043.7, 348.3 and 398.3 mg/L (**Fig. 5b**). The obtained values are below the standard for
411 drinking water required by World Health Organization (WHO) and the U.S. Environmental
412 Protection Agency (EPA) (i.e., Na^+ : 200 mg/L, Ca^{2+} : 75 mg/L, K^+ : 12 mg/L, and Mg^{2+} : 50
413 mg/L) (WHO, 1993; EPA, 2012). This result proves the capability of Au nanolayer-PTFE
414 membrane-based solar evaporator for producing drinking water.

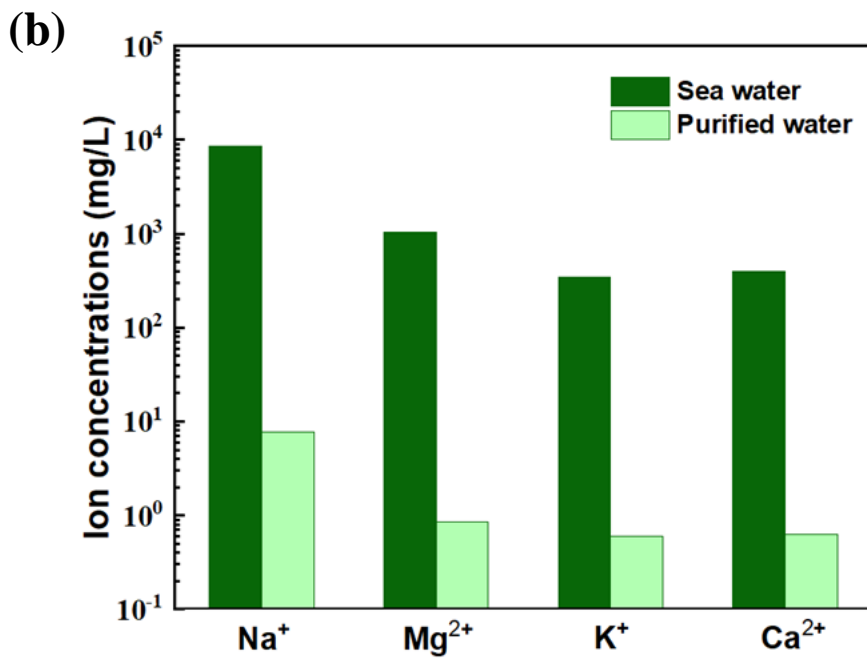
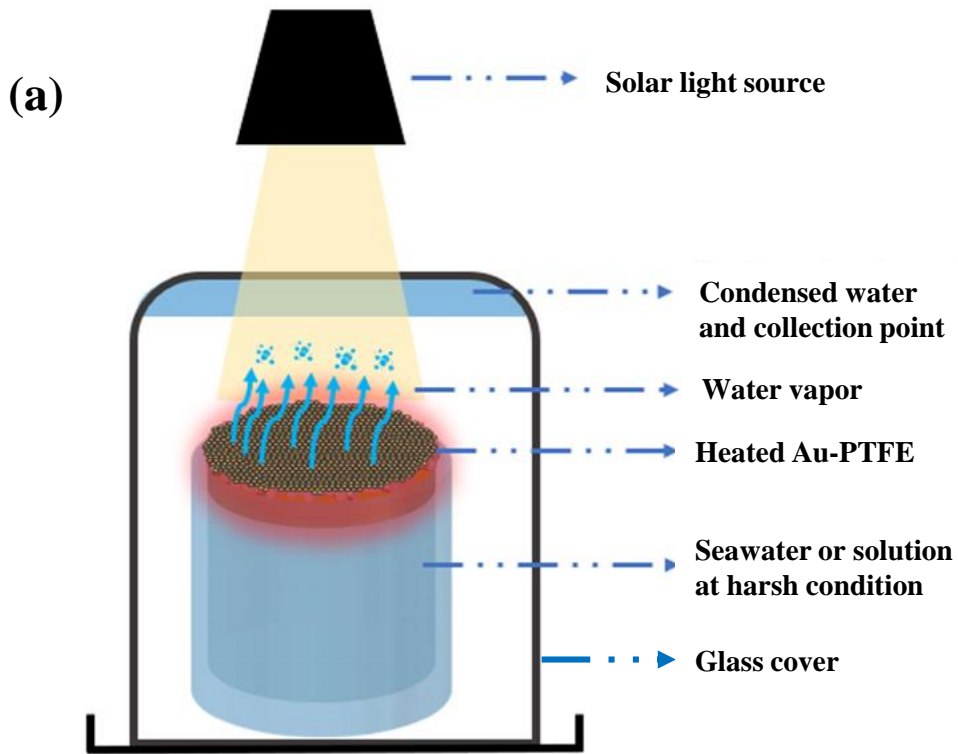
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421 **Fig. 5** (a) Schematic of the purpose-built device for collecting water vapour. (b) The

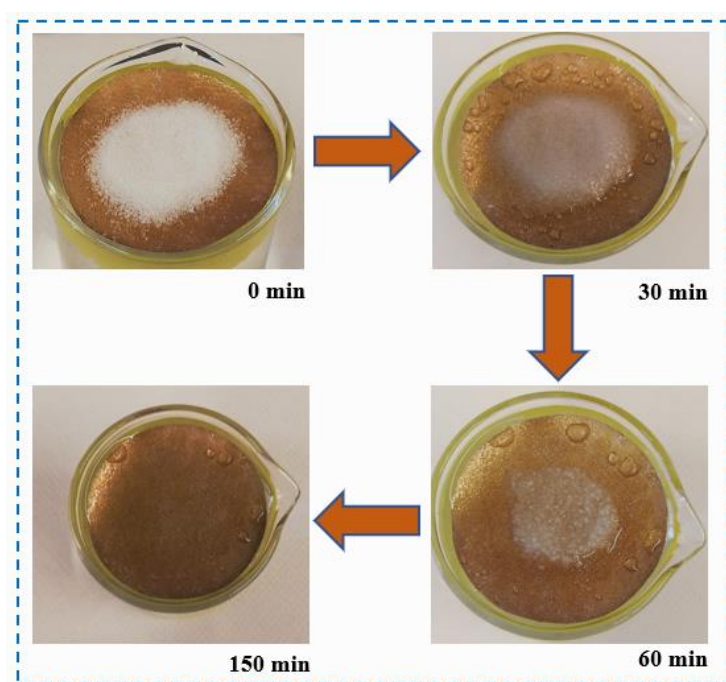
422 concentrations of ions (Na⁺, K⁺, Ca²⁺, and Mg²⁺) in seawater collected from Cronulla Beach

423 showing a high reduction in ion concentrations before and after the desalination process.

424

425 **3.6 Durability and stability test**

426 The durability and stability of the solar evaporator system are essential factors for practical
427 solar water evaporation applications. The salt formation on the solar absorbers will cause a
428 significant decrease in the light absorption of the photothermal materials, causing a lower
429 evaporation surface (Ibrahim et al., 2020; Ibrahim et al., 2021a). Therefore, it should be
430 addressed before implementing this technology for practical application. The self-cleaning
431 capability of the Au-PTFE@CS evaporator was investigated by placing 2.5 g of salt on the top
432 of the membrane (see **Fig. 6**). After sunlight irradiation at 1 sun intensity, the salt gradually
433 disappeared with continuous vapor flow through the holes between the coated fibre of Au-
434 PTFE membrane until no salt was observed after 2.5 hours (**Fig. 6**). The self-cleaning ability
435 of the designed evaporators can be ascribed to the spaces between the heated fibres in Au-
436 PTFE membranes which allows continuous vapor flow for chemical advection and diffusion
437 induced by salt concentration gradient, resulting in salt dissolution (Irshad et al., 2021; Zhou
438 et al., 2021). In addition, the hydrophilic open pores CS allows formation of thin layer water
439 underneath of the Au nanolayer membrane resulting in sufficient water supply and leading to
440 dissolution of the solid NaCl.



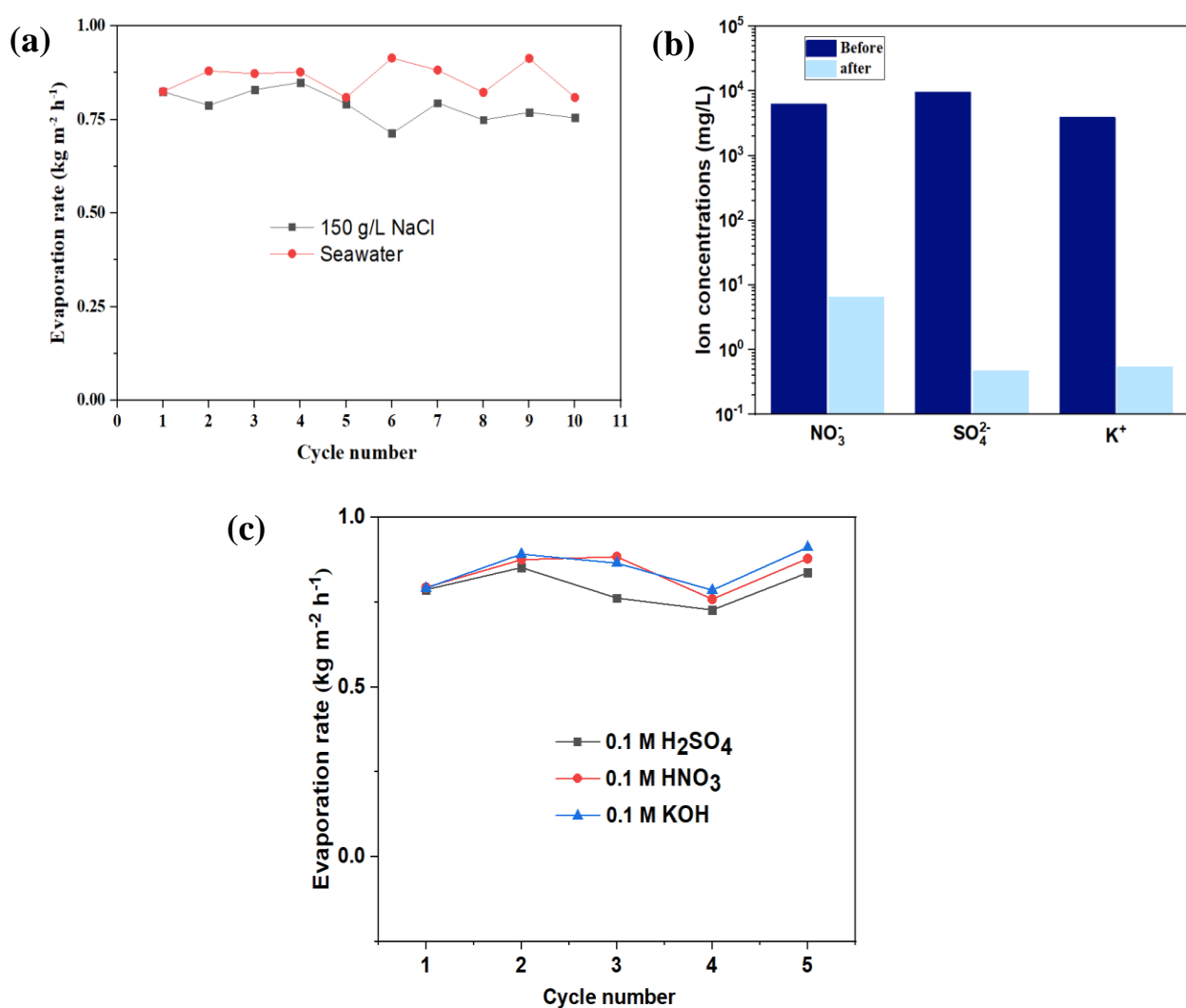
441

442 **Fig 6.** Photo-image of salt-dissolution of nanolayer Au-PTFE based solar evaporator after
443 adding 2.5 g NaCl, demonstrating a self-cleaning property

444

445 To examine the durability of Au nanolayer-PTFE membrane for solar desalination, we
446 conducted the evaporation test using natural seawater and high salinity brine (150 g/L NaCl)
447 as model brine solution. The test was carried out for 10 continuous cycles (1 h/cycle) under
448 one sun intensity. Interestingly, excellent reusability with stable performance over 10 cycles
449 was demonstrated, with only a slight decrease in the evaporation rate of $0.85 \text{ kg m}^{-2} \text{ h}^{-1}$ for
450 seawater, to $0.82 \text{ kg m}^{-2} \text{ h}^{-1}$ for the brine solution (150 g/L NaCl) (**Fig. 7a and Fig. S10**). The
451 slight reduction in the evaporation rate for brine solution is attributed to the decrease in vapor
452 pressure of the brine solution (Zhang et al., 2015; Xu et al., 2019; Luo et al., 2021). The above-
453 mentioned results indicate excellent salt resistance due to the continuous water supply through
454 CS, which also provides a thin layer of water. In addition to that, due to the surface
455 hydrophobicity of the Au nanolayer-PTFE membrane, no noticeable salt was formed. This
456 result implies that MS technique is a reliable technology to fabricate a stable nanolayer
457 photothermal membrane for seawater desalination, especially towards high-salinity solution
458 with anti-fouling properties. We further extend the durability study to harsh chemical
459 conditions, under acidic (0.1M H_2SO_4 , pH = 1), basic (0.1M KOH, pH = 13), and oxidizing
460 conditions (0.1M HNO_3), corresponding to ionic strengths (I) of 0.3 M, 0.1 M, and 0.1 M,
461 respectively. The ion concentrations of SO_4^{2-} , K^+ and NO_3^- were reduced to 0.4649, 6.41121,
462 and 0.5434, mg L^{-1} , respectively, revealing remarkable efficiency in obtaining clean water even
463 under harsh conditions (**Fig. 7b**). Besides, the evaporation rate indicates that the fabricated Au
464 nanolayer membrane possess an excellent evaporation rate of $0.80 \text{ kg m}^{-2} \text{ h}^{-1}$ (under 0.1 M
465 H_2SO_4), $0.85 \text{ kg m}^{-2} \text{ h}^{-1}$ (under 0.1 M KOH), and $0.85 \text{ kg m}^{-2} \text{ h}^{-1}$ (under 0.1 M HNO_3), which
466 are close to that of the corresponding DI water evaporation rate ($0.88 \text{ kg m}^{-2} \text{ h}^{-1}$), suggesting

467 its stable evaporation performances (**Fig. S11**). The slight reduction in evaporation rate of
 468 H_2SO_4 is attributed to the higher I value compared to HNO_3 and KOH . Additionally, the cycling
 469 stability was carried out for 5 cycles (1 h test per 1 cycle). Ultimately, the results showed
 470 excellent reusability with stable performance (**Fig. 7c**). This good performance are attributed
 471 to three main reasons: (i) the chemical resistivity of the gold (Hammer and Norskov, 1995;
 472 Badawy et al., 2010; Kiriarachchi et al., 2018), (ii) the superior mechanical strength of the Au
 473 nanolayer-PTFE membrane, and (iii) the strong adhesiveness and attachment of the thin layer
 474 Au on the PTFE membrane. Therefore, it can be concluded that the fabricated gold membrane
 475 allowed efficient steam diffusion, minimized salt accumulation, and maintained membrane
 476 cycle stability under harsh environments.



492 **Figure 7. (a)** Cycling performance of the Au nanolayer membrane using seawater, revealing
493 stable evaporation rate for a period of 10 cycles. **(b)** Ion concentrations (SO_4^{2-} , NO_3^- , and K^+)
494 before and after the evaporation showing the capability of Au nanolayer membrane for
495 excellent ions rejection under harsh environment. **(c)** The cycling stability under acidic, basic
496 and oxidizing conditions for a period of 5 cycles.

497

498 **Conclusion**

499 This study demonstrates the feasibility of applying a magnetron sputtering method to fabricate
500 a novel Au nanolayer membrane with robust mechanical strength and chemical stability for
501 SWE applications. Detailed morphological, structural and material analysis techniques
502 indicated that gold nanolayer were formed and homogeneously coated on the fiber of PTFE
503 membrane surface. The resultant Au nanolayer membranes achieved a water evaporation rate
504 of $0.88 \text{ kg m}^{-2} \text{ h}^{-1}$, which corresponds to an efficiency of 49.5%, under 1-sun irradiation. Also,
505 this system revealed a high rejection rate of salt ions from seawater with a removal efficiency
506 of 99.9%, meeting the standard required for drinking water. Furthermore, the adhesiveness and
507 the strong attachment of the thin film Au nanolayer on the membrane resulted in excellent
508 durability and stability even under harsh conditions. Our proposed novel strategy may open up
509 a new pathway for designing a highly stable nanolayer thin-film PTM for solar water
510 purification.

511

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523

524 **CrediT authorship contribution statement**

525 **Idris Ibrahim:** Conceptualization, methodology, writing-original draft, review & editing
526 Visualization, data curation, Investigation. **Dong Han Seo:** Conceptualization, supervision,
527 visualization, data curation, review & editing. **Alexander Angeloski:** Methodology, review &
528 editing. **MJ Park:** Methodology. **Andrew McDonagh:** Supervision, data curation, review &
529 editing. **Avi Bendavid:** Conceptualization, methodology, investigation. **Hokyong Shon:**
530 Supervision, resources, review & editing. **Leonard Tijing:** Supervision, resources,
531 investigation, planning, data curation, visualization, review & editing.

532

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