

1 **Key factors in improving the synthesis and properties of visible-**
2 **light activated g-C₃N₄ for photocatalytic hydrogen production and**
3 **organic pollutant decomposition**

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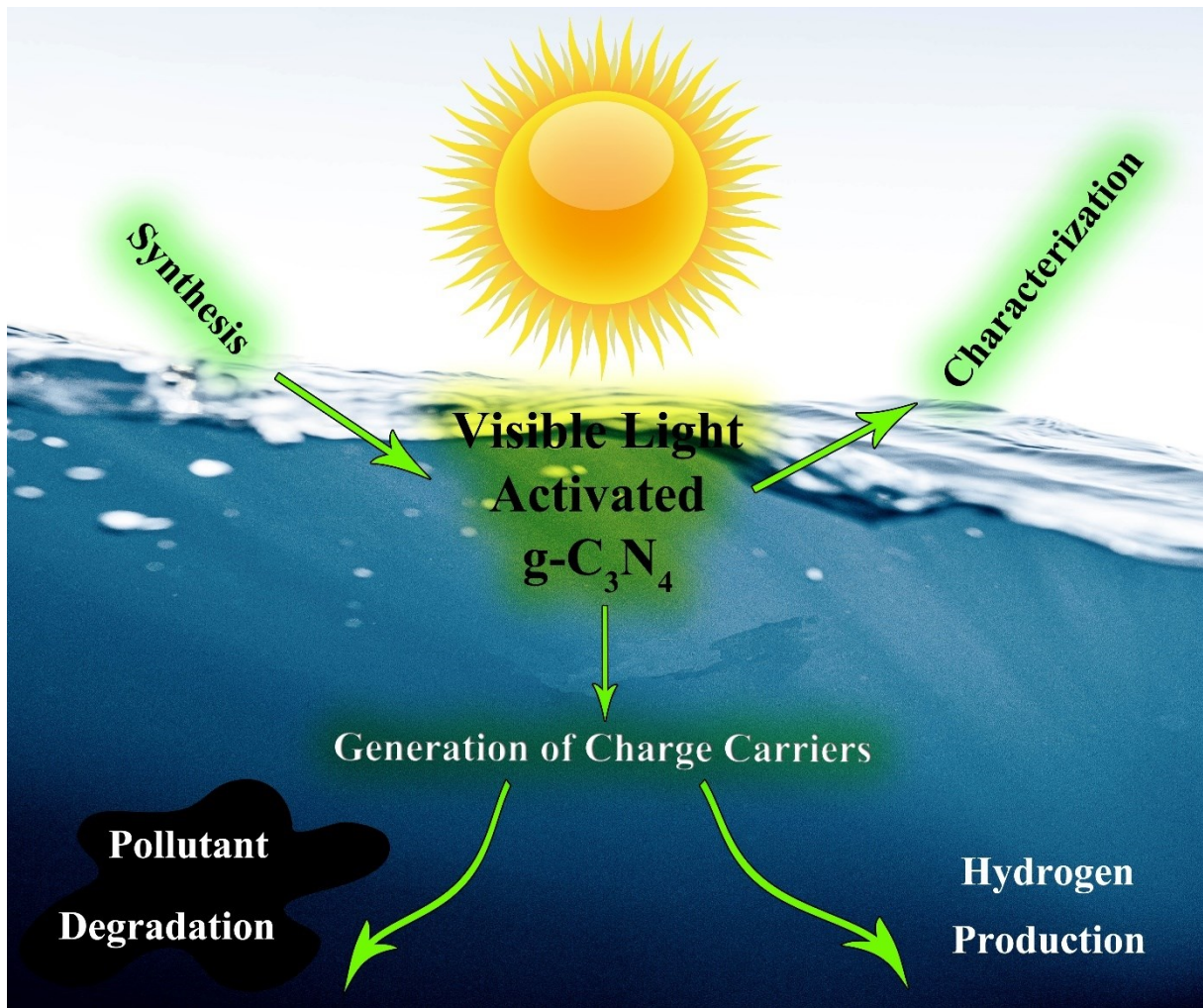
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1 Graphical abstract

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3

1 **Abstract**

2 Of current interest is visible-light activated g-C₃N₄, owing to its unique physicochemical
3 properties for the photocatalytic H₂ production and pollutant remediation. In this research, the
4 synthetic procedures, physicochemical properties, and major approaches to overcome the
5 intrinsic drawbacks of g-C₃N₄ are reviewed. Of different synthesis procedures for GCN,
6 thermal polymerization is recommended with advantages such as simplicity, economic, and
7 high yield. Element doping, as a facile method, can modify g-C₃N₄ structure and improve its
8 performance in the photocatalytic evolution of H₂ which is significantly increased from 208
9 μmol h⁻¹ g⁻¹ for bare g-C₃N₄ to 5128 μmol h⁻¹ g⁻¹ for P-doped g-C₃N₄. Ammonium salts can be
10 effectively used for the synthesis of g-C₃N₄ nanosheets and element doping simultaneously.
11 Vacancy defect plays an important role in the improvement of the photocatalytic activity.
12 Compared to custom photocatalysis and electrocatalysis, photoelectrocatalysis is highly
13 promising for pollutant decontamination due to the effective separation of charge carriers.

14

15 *Keywords:* Element doping; g-C₃N₄; Photocatalytic H₂ production; Photoelectrocatalysis;
16 Pollutant remediation

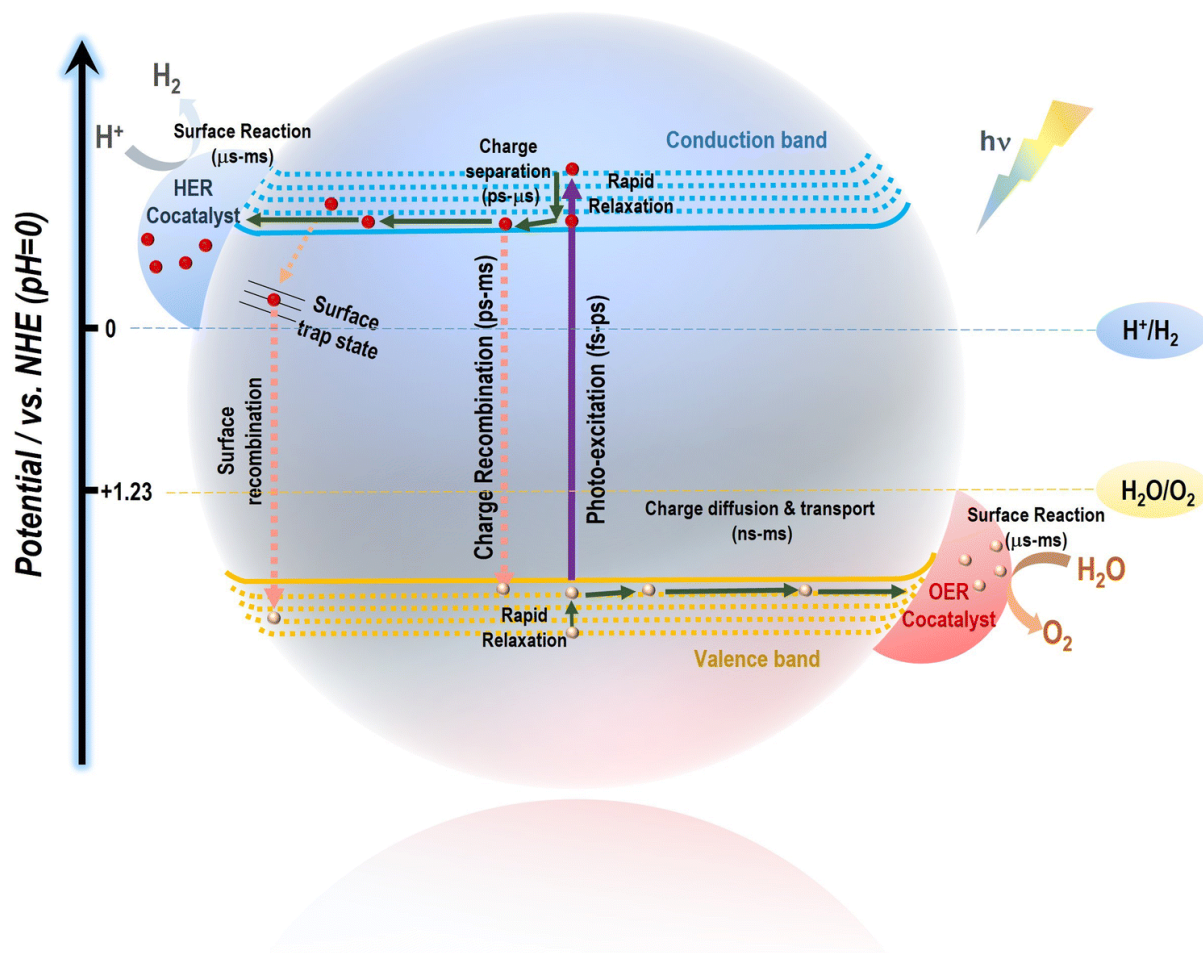
1 **1. Introduction**

2 The generation of renewable green energy such as hydrogen as a promising alternative source
3 of energy is of worldwide interest to support human sustainable development.¹ Photocatalytic
4 hydrogen production, with marginal adverse effects on the environment,² has gained extensive
5 attention due to its potential applications in future.³ Thermodynamically, the overall water
6 splitting into H₂ and O₂ is considered an uphill endothermic reaction ($\Delta G^\circ = 237 \text{ kJ mol}^{-1}$),
7 whereas a driving force for redox reactions is provided by sunlight irradiation in photocatalytic
8 reactions.⁴ Notably, the photogenerated holes and electrons on a semiconductor-based
9 photocatalyst can initiate oxygen evolution reaction (OER) and hydrogen evolution reaction
10 (HER), respectively, through the equations (1) and (2):⁵



13 The schematic illustration of photocatalytic water splitting is shown in Figure 1. To
14 evaluate the respective mechanisms of OER and HER, sacrificial agents are usually used.⁴
15 Sacrificial agents could highly affect the photocatalytic HER where methanol, sodium
16 sulfide/sodium sulfite, and triethanolamine (TEA or TEOA) are the most frequently used
17 sacrificial agents for oxide, sulfide, and carbon semiconductors, consecutively.² Notably, the
18 photocatalytic OER or HER half-reactions (equations (1) and (2)) could proceed much easier
19 than photocatalytic overall water splitting due to their considerably smaller ΔG than 273 kJ
20 mol⁻¹ (for overall water splitting).⁴ In addition to the energy issues, water availability and water
21 pollution are another serious problem as people living in arid regions may be forced to reduce
22 daily water intake and water use in agriculture, reducing their domestic food production.⁶
23 Wastewater treatment has been extensively used for the removal of organic pollutants, however,
24 persistent organic pollutants (POPs) and emerging organic pollutants (e.g. antibiotics,
25 endocrine disrupting chemicals) could remain in the treated effluent, due to the limited
26 efficiency of current treatment technologies.^{7,8} The adverse effects of environmental pollution

1 are challenging the entire world. For example, long-term exposure to antibiotic residues could
2 support the development of antibiotic resistance genes (ARGs) which present another
3 dimension to the pollution problem.⁹



4
5 Figure 1. Schematic illustration of photocatalytic water splitting using a semiconductor-
6 based material. Reproduced with permission from Ref. ¹⁰ Copyright 2022 The Royal Society
7 of Chemistry (<https://pubs.rsc.org/en/Content/ArticleLanding/2022/CS/D2CS90098J>).
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9 Different methods have been used for the removal of organic pollutants including
10 membrane processes (e.g. nanofiltration, forward osmosis, reverse osmosis), sonochemical
11 degradation, advanced oxidation processes (AOPs), adsorption, micellar-enhanced
12 ultrafiltration (MEUF), electrochemical oxidation (EO), precipitation, coagulation,
13 biodegradation, and ion exchange. Adsorption, coagulation, flocculation, and membrane
14 processes typically concentrate and transfer pollutants from one phase to another.¹¹ AOPs

1 include various techniques such as ozonation, persulfate-based oxidation, electrolysis,
2 photolysis, sonolysis Fenton-based processes, and photocatalysis.^{12, 13} Hydroxyl radicals ($\cdot\text{OH}$)
3 are extremely highly reactive radicals which can be generated by primary oxidants (such as
4 ozone or hydrogen peroxide), and are capable of oxidizing various organic pollutants.¹⁴
5 Notably, they could attack organic pollutants through four general mechanisms: transfer of
6 electrons, abstraction of hydrogens, radical combination, and radical addition.¹⁵ AOPs are
7 universal technologies which can be used for degradation of organic pollutants (even complete
8 mineralization).¹⁶ However, it is also of high importance to consider the toxicity of degradation
9 by-products or intermediates, some of which could be even more toxic than the original
10 pollutant.¹⁷ Overall, semiconductor photocatalysis relies on four steps as follows:¹⁸

- 11 a) Absorption of photons to create charge carriers,
- 12 b) Separation of charge carriers,
- 13 c) Transfer of charge carriers to the surface of photocatalyst, and
- 14 d) Redox reactions initiated by charge carriers.

15 Surface reactions could happen only if the oxidation potentials are more negative than the
16 potential of valence band (VB) or the reduction potentials are more positive than the potential
17 of conduction band (CB) in oxidation and reduction processes, respectively.¹⁹ Although
18 photocatalysis is well-known as an efficient process for water and wastewater treatment, several
19 key points should be considered to facilitate its practical applications. Importantly,
20 semiconductor particles cannot be easily recycled after wastewater treatment. Another key
21 point is that 3-6% of the solar irradiation is in the ultraviolet spectrum;²⁰ yet TiO_2 , the most
22 common semiconductor material, has negligible visible-light adsorption.¹⁸ TiO_2 ,^{13, 21} ZnO ,²²
23 SnO_2 ,²³ and Zn_2SnO_4 ^{24, 25} are among metal-based semiconductors used for photocatalytic
24 applications, but they all have wide band gap energies (>3.0 eV), restricting their visible light
25 harvest.

1 Taking advantages of their lower band gap energies than TiO_2 , ZnO , and SnO_2 , some other
2 metal-based semiconductors such as $\alpha\text{-Fe}_2\text{O}_3$,²⁶ Ag_3PO_4 ,²⁷ CdS ,²⁸ Ag_2S ,²⁹ BiVO_4 ,³⁰ MoS_2 ,³¹
3 ZnFe_2O_4 ^{32, 33} and Ag_3VO_4 ³⁴ have been used for photocatalytic applications under visible light
4 exposure. However, there are still some drawbacks for $\alpha\text{-Fe}_2\text{O}_3$ (low diffusion length of holes,
5 high recombination rate of charge carriers, and poor conductivity),³⁵ Ag_3PO_4 (low
6 photostability and fast recombination of charge carries)²⁷, MoS_2 (high recombination rate of
7 charge carriers, poor conductivity, and low stability),³¹ CdS (photocorrosion in pure water and
8 fast recombination of charge carries),³⁶ Ag_2S (fast recombination of charge carries),³⁷ ZnFe_2O_4
9 (leaching, agglomeration, fast recombination of charge carries, and high photocorrosion rate),³⁸
10 BiVO_4 (low photon efficiency and poor electron mobility),³⁹ and Ag_3VO_4 (instability and fast
11 recombination of charge carries).³⁴ Considering the energy conversion domain, using stable and
12 cost-effective semiconductors with suitable band gap energies for practical photocatalytic water
13 splitting is a major difficulty.^{40, 41} Metal-free photocatalysts are substantial alternatives to metal-
14 based semiconducting materials. Some metal-free photocatalysts have shown a superior
15 photocatalytic (PC) performance than metal-based counterparts.⁴² Owing to its optical,
16 structural, and electronic properties, graphene is one of the most important carbon-based
17 materials.⁴³ Graphitic carbon nitride, abbreviated as GCN or $\text{g-C}_3\text{N}_4$, is another burgeoning and
18 promising carbon-based metal-free material owing to its unique advantages including facile
19 synthesis using inexpensive precursors, high physicochemical stability, suitable band gap
20 energy, abundance, and eco-friendly nature.^{44, 45} As a polymeric material, it takes advantage of
21 its controllable morphology, size, pore structure, thickness, and easy synthetic procedure.⁴⁶

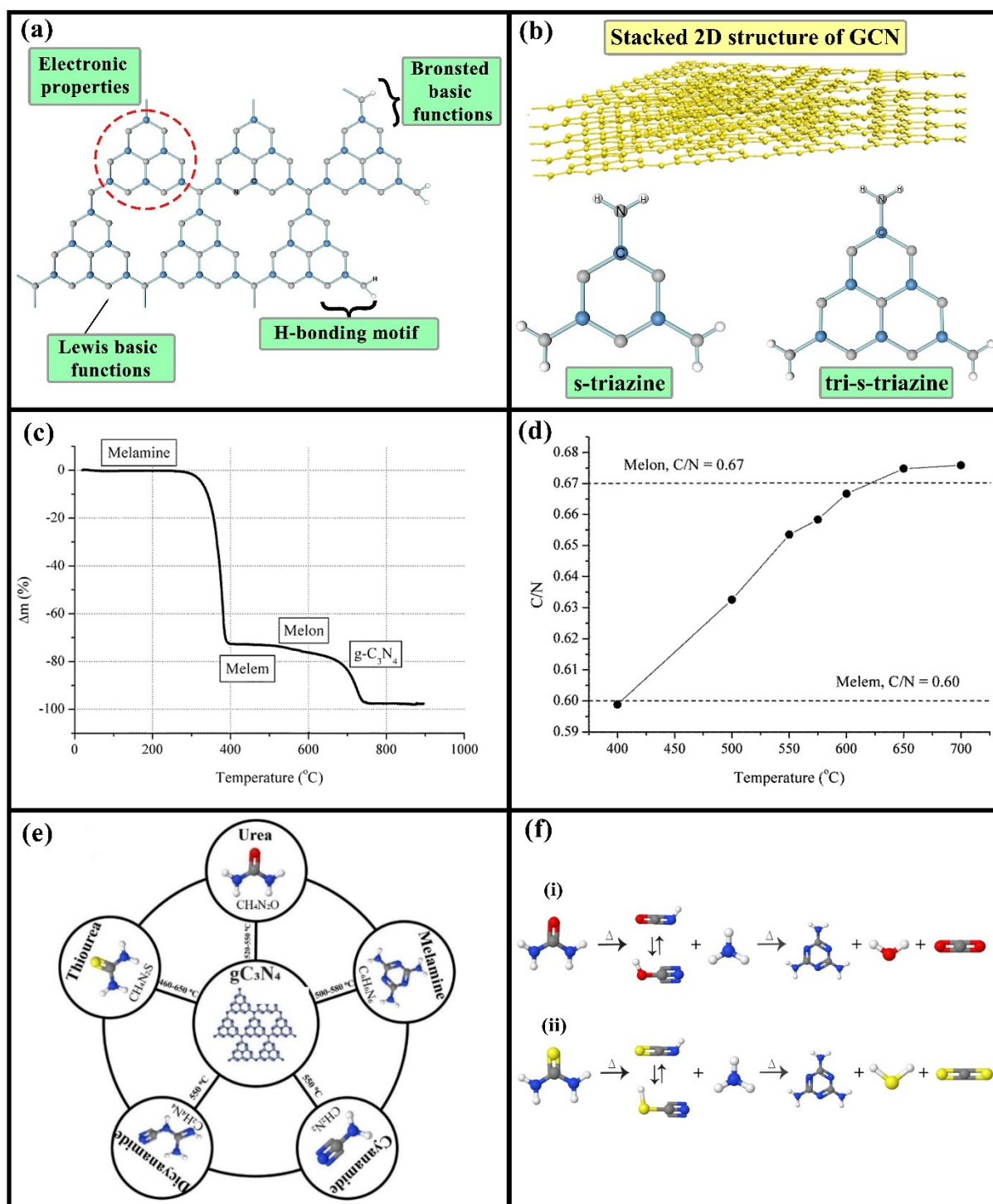
22 Compared with TiO_2 and ZnO , $\text{g-C}_3\text{N}_4$ possesses a more negative CB level that makes it a
23 suitable candidate for reduction reactions. Regarding the overall photocatalytic water splitting,
24 the bottom of the CB level and the top of the VB level of the used semiconductor must be
25 respectively negative and positive enough to facilitate reduction and oxidation reactions.⁵
26 Overall, with a low band gap energy of 2.7 eV, GCN is considered a promising photocatalytic

1 material for development of solar energy.⁴⁶ Although individual photocatalysts generally
2 provide weak photocatalytic activity for H₂ generation under visible light irradiation, a
3 combination of GCN with a metal co-catalyst and a sacrificial agent (hole scavenger) could
4 provide high photoactivity.⁴⁷ In addition to photocatalytic water splitting, GCN has been widely
5 used for photocatalytic CO₂ reduction.⁴⁸ Apart from its potential applications for solar fuel
6 production via photocatalytic water splitting and CO₂ reduction,⁴⁹ GCN-driven “gold rush” is
7 considered a novel and hotspot field which has received wide attention in environmental
8 remediation.⁵⁰ Up to now, GCN has been used for the degradation of various types of organic
9 pollutants from different industries such as textile, plastic, pharmaceutical, and agriculture
10 (pesticides and insecticides), as summarized in Table 1.

Table 1. Applications of g-C₃N₄ composites in the photocatalytic degradation of organic pollutants under visible light.

Application	Photocatalyst	Pollutant	Illumination source	Photocatalytic activity	Reference
Textile	g-C ₃ N ₄ /CuWO ₄ nanocomposite	Methylene blue (MB)	250 W High pressure mercury lamp	~ 97% (t = 3 h)	51
Textile	g-C ₃ N ₄ /Bi ₂ WO ₆ composite	Methyl orange (MO)	500 W Xenon lamp	~ 100% (t = 3 h)	52
Textile	g-C ₃ N ₄	Rhodamine B (RhB)	300 W Xenon lamp	~ 68% (t = 50 min)	53
Plastic	Pd/g-C ₃ N ₄	Bisphenol A	350 W Xenon lamp	~ 100% (t = 6 h)	54
Plastic	g-C ₃ N ₄ /BiVO ₄ heterostructure	Phenol	500 W Halogen lamp	~ 50.2% (t = 2 h)	55
Pharmaceutical	g-C ₃ N ₄ (1.5 wt%)/Hap composite	Tetracycline (TC)	300 W Xenon arc lamp	~ 100% (t = 15 min)	56
Pharmaceutical	Ag/TiO ₂ /g-C ₃ N ₄ nanocomposite	Amoxicillin	300 W Xenon lamp	k = 0.0614 min ⁻¹	57
Agriculture	Cu-doped ZnO/ g-C ₃ N ₄	Atrazine	Visible radiation (400-700 nm)	~ 90% (t = 3 h)	58
Agriculture	g-C ₃ N ₄ /N-doped CeO ₂ composite	Diuron	1500 W Xenon lamp	~ 46% (t = 2 h)	59

1 Carbon nitrides mainly include carbon and nitrogen and can be obtained from carbon
2 materials (through substitution of C atoms by N atoms). GCN is a polymeric metal-free n-type
3 semiconductor material that has received attention owing to its unique structural, electrical,
4 physicochemical, and optical properties. Notably, GCN is the most stable polymorph of carbon
5 nitrides in the ambient atmosphere.^{60, 61} Overall, there are seven allotropes of C_3N_4 including
6 alpha, beta, cubic, pseudocubic, g-h-triazine, g-o-triazine, and g-h-heptazine with space groups
7 of $P3_1c$ (159), $P3$ (143), $I-43d$ (220), $P-42m$ (111), $P-6m2$ (187), $P2mm$ (25), and $Cmc2_1$ (36),
8 respectively.⁶² GCN has a stacked two-dimensional (2D) layered structure where single layers
9 of nitrogen heteroatom-substituted nanosheets of graphite are bound only by weak van der
10 Waals forces. These single layers of nitrogen heteroatom-substituted nanosheets of graphite
11 are formed by sp^2 hybridization of nitrogen and carbon atoms.¹⁹ Real g- C_3N_4 possesses a tiny
12 amount of hydrogen which presents as primary/secondary amine groups on the terminating
13 edges.⁶¹ The probable multiple surface functionalities on g- C_3N_4 are illustrated in Figure 2(a).



1

2 Figure 2. (a) Probable multiple functionalities of g-C₃N₄. Redrawn from Ref. ⁶⁰ Copyright

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(<https://pubs.rsc.org/en/content/articlelanding/2008/jm/b800274f>). (b) The stacked 2D

5

structure of g-C₃N₄ and its primary building blocks. Redrawn from Ref. ¹⁹ Copyright 2017

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1 from Ref. ⁶³ Copyright 2017 Elsevier. (d) Molar ratios of C/N for condensation of melamine
2 products at different temperatures from 400 °C to 700 °C. Reprinted with permission from
3 Ref. ⁶³ Copyright 2017 Elsevier. (e) Common precursors and corresponding calcination
4 temperatures used to synthesize GCN (hydrogen, carbon, oxygen, nitrogen, and sulfur atoms
5 are represented by white, gray, red, blue, and yellow balls). Reprinted with permission from
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7 terms and conditions of the Creative Commons Attribution (CC BY) license
8 (<https://creativecommons.org/licenses/by/4.0/>). (f) The conversion mechanism of (i) urea and
9 (ii) thiourea to melamine at high temperatures (hydrogen, carbon, oxygen, nitrogen, and
10 sulfur atoms are represented by white, gray, red, blue, and yellow balls). Reprinted with
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14
15 The incomplete condensation of precursors could result in the existence of hydrogen and
16 creation of some surface defects including the Lewis-base character which can promote the
17 relocation of electrons on the surface, consequently improving the catalytic activity.⁶¹
18 GCN, which could take advantages of the availability of hydrogen and nitrogen, has unique
19 electron-rich properties, basic surface functionalities, good electronic band structures, and high
20 thermal/hydrothermal (physicochemical) stability.^{61, 65} Its particular surface properties,
21 induced by nitrogen, enables g-C₃N₄ as more promising than other carbon nanomaterials.⁶⁵
22 Generally, two different condensation states (i.e. building blocks) are considered as its primary
23 building blocks: triazine (ring of C₃N₃) with periodic arrays of single carbon vacancies and tri-
24 s-triazine rings (heptazine: triring of C₆N₇) with larger periodic vacancies in the lattice.
25 Heptazine rings are connected through planar tertiary amino groups.¹⁹ The stacked 2D structure

1 of g-C₃N₄ and its primary building blocks are shown in Figure 2(b). Tri-s-triazine is considered
2 the most accepted structural unit under ambient conditions^{66, 67} since tri-s-triazine-based g-
3 C₃N₄ is energetically more stable than triazine-based g-C₃N₄.¹⁹ The high condensation and tri-
4 s-triazine ring structure are responsible for the high stability of g-C₃N₄ concerning thermal and
5 chemical attacks.⁶⁸

6 This study aims to critically review the principles and advances in the synthesis of g-C₃N₄,
7 and to suggest potential approaches to improve its photocatalytic activity. In addition,
8 photoelectrocatalysis is reviewed as a more promising technology than photocatalysis for the
9 decontamination of organic pollutants over GCN-based materials.

10

11 **2. Methods of g-C₃N₄ synthesis**

12 Various methods including chemical vapor deposition (CVD), physical vapor deposition
13 (PVD), solid-state reaction, thermal annealing/condensation, sonochemical, and solvothermal
14 have been used for the synthesis of GCN. Thermal condensation suffers from its low yield
15 (approximately 6%) that originates from the higher polymerization temperature of g-C₃N₄ than
16 the sublimation point of reagents, resulting in their loss prior to the occurrence of reaction.⁴⁶
17 However, its simplicity, absence of impurities (during synthesis), low cost, and convenience⁶⁵
18 have introduced this method as the most efficient and widely-used for the synthesis of GCN.⁴⁶
19 ⁶³ Thus, this method is discussed in detail.

20

21 **2.1. Thermal condensation**

22 Reactive nitrogen-rich and oxygen-free compounds that possess pre-bonded C-N core
23 structures (such as heptazine and triazine derivatives) are the most common precursors in
24 chemical routes.⁶¹ g-C₃N₄ catalysts are usually produced by thermal condensation of organic
25 compounds (such as urea, thiourea, cyanamide, dicyanamide, and melamine) as the most

1 attractive method owing to the use of earth-abundant and cheap precursors and the simplicity
2 of synthetic process.⁶⁹ Thermal polymerization synthesis of GCN includes poly-addition and
3 poly-condensation.⁷⁰ Wang et al. synthesized GCN using cyanamide as the precursor,⁷¹
4 although it should be noted that cyanamide is explosive and expensive. Generally, the synthesis
5 of g-C₃N₄ using cyanamide and dicyanamide (as its derivative) is hard to operate and
6 expensive.⁶¹ It has been shown that using precursors containing other elements, in addition to
7 H, N and C, could facilitate the process of thermal condensation and results in the structural
8 alterations.⁷² To overcome these obstacles, different guanidine compounds such as guanidine
9 carbonate,⁷³ guanidine thiocyanate,⁷⁴ and guanidine hydrochloride (as an inexpensive, widely
10 available, and environmentally-friendly precursor)⁷² have been used for the synthesis of g-
11 C₃N₄. However, GCN synthesized by melamine showed a higher photocatalytic activity than
12 that synthesized by guanidine carbonate for the degradation of MO.⁷³ The presence of oxygen
13 atoms in urea could facilitate the condensation of GCN and boost its stability.⁶⁴ The
14 thermogravimetric curve of annealing of melamine and the molar ratio of C/N for condensation
15 of melamine products at different temperatures are shown in Figures 2(c) and 2(d),
16 respectively. As shown in Figure 2(c), the first and the highest reduction of mass,
17 approximately 70%, occurs between 280 and 400 °C. This remarkable decrease of mass was
18 attributed to the sublimation and condensation of melamine which is accompanied by the
19 elimination of ammonium and formation of melem (C₆N₁₀H₆) as a by-product. Noteworthy,
20 melam could be formed at temperatures above 300 °C, typically 400-500 °C. At temperatures
21 higher than 500 °C, melon (C₆N₉H₃) could be produced due to the polymerization of melem.
22 The polymerization of melon to the GCN planar structure could occur at temperatures higher
23 than 600 °C. However, GCN is unstable at high temperatures, resulting in the formation of
24 some volatile decomposition products.⁶³ Common precursors and corresponding calcination
25 temperatures used to synthesize GCN are shown in Figure 2(e). As evident, melamine could

1 be generated from either thiourea, urea, dicyanamide or cyanamide, where its conversion to
2 melem and melon could lead to the production of g-C₃N₄ by polymerization of tri-s-triazine
3 ring structure.⁶⁴ The conversion mechanism of thiourea and urea into melamine is shown
4 schematically in Figure 2(f).

5

6 **2.1.1. Effect of different precursors on photocatalysis**

7 Xin et al. synthesized g-C₃N₄ from the pyrolysis of cyanamide, melamine, and dicyandiamide
8 precursors for the photocatalytic degradation of MB under visible light irradiation. Using
9 different precursors did not significantly affect the morphology and texture of GCN in their
10 research. GCN synthesized by dicyandiamide showed the highest photocatalytic efficiency
11 owing to its high C/N ratio (0.685), good crystal structure, and moderate degree of
12 condensation. They concluded that increasing C/N ratio and degree of condensation could
13 improve the photocatalytic activity.⁷⁵ Zhang et al. synthesized GCN by the pyrolysis of
14 melamine, dicyandiamide, urea and thiourea precursors (550 °C, 3 h, and heating rate of 10
15 °C/min) for the photocatalytic removal of NO under visible-light irradiation. The band gap
16 energies of g-C₃N₄ were 2.51, 2.56, 2.58, and 2.88 eV by using thiourea, melamine,
17 dicyandiamide, and urea, respectively. The minimum and maximum photo-absorption values
18 in the visible-light region were obtained by g-C₃N₄ synthesized by urea and thiourea,
19 respectively.⁷⁶ The preparation method of g-C₃N₄ affects its photo-absorption ability. Thus, the
20 type of precursor and condensation temperature that result in the formation of different local
21 structures, defects, and packing (degree of condensation) are among factors which affect the
22 photo-absorption ability of g-C₃N₄.⁶⁸ Notably, due to its aromatic structure, melamine itself
23 has a strong absorption edge and could significantly absorb UV irradiation.⁶³ The g-C₃N₄
24 synthesized by thiourea, dicyandiamide, melamine and urea achieved BET surface areas of 23,
25 18, 14, and 153 m²/g; pore volumes of 0.14, 0.09, 0.06, and 0.40 cm³/g; and apparent rate

1 constants of 0.310, 0.079, 0.298, and 0.384 min⁻¹ for the photocatalytic degradation of NO
2 respectively. The results clarify the key role of oxygen and sulfur heteroatoms in enlarging the
3 pore volume and enhancing the specific surface area. Notably, they all showed very similar
4 thermal stabilities (<550 °C, <563 °C, <575 °C, and <530 °C, consecutively).⁷⁶ Although the
5 synthetic efficiency (yield) of GCN using different precursors is remarkably different, it has
6 rarely been reported. Alwin et al. have measured the yield of GCN synthesized by different
7 precursors at different annealing conditions. Notably, both annealing conditions and type of
8 precursors significantly affected the yield of GCN though at varying degrees. It should be noted
9 that increasing the temperature, time, and rate of annealing decreased the yield of GCN.
10 However, the effect of annealing rate was not considerable. Furthermore, the yields of GCN
11 using melamine, thiourea, and dicyanamide precursors were 26%, 11%, and 53%, respectively,
12 signifying the moderate production efficiency of GCN using melamine.⁷⁷ It should be noted
13 that the yield of GCN by using urea is much lower than that by using melamine⁷⁸,
14 dicyanamide,⁷⁹ and guanidine hydrochloride.⁷⁹

15 The high content of mesopores and micropores in GCN synthesized using urea and thiourea
16 was responsible for the formation of interconnected porous networks. Although the thermal
17 stability of g-C₃N₄ synthesized by different precursors did not vary significantly, the highest
18 and the lowest amount were attributed by using melamine and urea, respectively. Notably,
19 different thermal stability is attributed to different packing between the layers and/or different
20 degree of condensation during the polymerization process. Although using urea resulted in the
21 production of g-C₃N₄ with the lowest photo-absorption ability, it yielded the highest
22 photocatalytic activity attributed to the large pore volume and high specific surface area. It was
23 concluded that the photocatalytic activity of g-C₃N₄ strongly depends on its specific surface
24 area, degree of condensation, pore volume, and band gap energy.⁷⁶

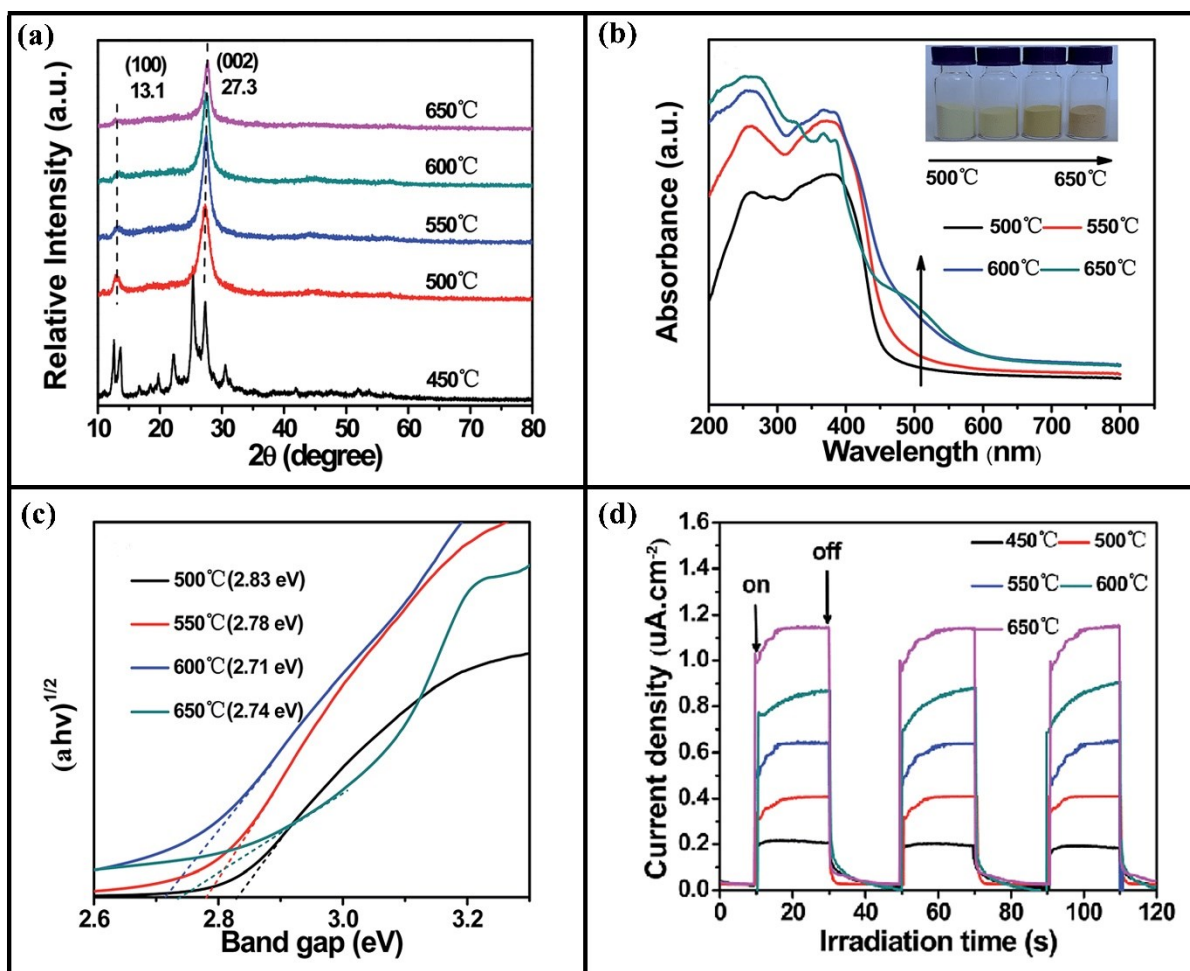
1 The higher the order of condensation, the stronger the conjugative effect is observed as a
2 result of the formation of defects in packing and local structures.⁸⁰ Notably, most of active
3 materials are slightly disordered, which means that grain boundaries, structural defects,
4 stacking defects, and surface termination are inherent features of catalysts.⁶⁸ Wu et al.
5 suggested that the formation of defect-free GCN is inevitable due to the various intermediates
6 evolved during synthesis (such as cyanuric acid, ammelide, biuret, and ammeline). Although
7 some intermediates could improve the separation/transfer of charge carriers, others may serve
8 as recombination centers of photo-induced e^-/h^+ pairs.⁸¹ It has been shown that urea could result
9 in the formation of thin crinkly paper-folded sheets of g-C₃N₄, while thiourea could result in
10 the formation of thick crinkly paper-folded sheets of g-C₃N₄.⁸⁰ Owing to their excessive oxygen
11 and sulfur contents that result in the formation of some gases (including H₂S, CO₂, NH₃, and
12 H₂O) during the polymerization process, favoring the formation of porous structures and
13 expansion of packing layers, both urea and thiourea could be effectively used to synthesize
14 GCN with high specific surface area and large pore volume.^{76, 82} Noteworthy, NH₃ and CO₂
15 gases could evolve during the thermal treatment process at low temperatures (< 200 °C) and
16 high temperatures, respectively.⁶⁵ The production of gas bubbles during calcination might be
17 considered as the main reason for the large surface area of g-C₃N₄ synthesized by thiourea.
18 Owing to this feature, thiourea has been used as the bubble template for the synthesis of
19 nanoporous g-C₃N₄ by dicyandiamide-thiourea mixtures.⁸³

20

21 **2.1.2. Effect of synthesis temperature**

22 Mo et al. synthesized g-C₃N₄ by thermal condensation process (heating rate of 2 °C min⁻¹, 2 h)
23 using melamine as the precursor and evaluated the effect of different temperatures (450, 500,
24 550, 600, and 650 °C) on the photocatalytic degradation of MB under visible light irradiation.
25 The XRD patterns of GCN synthesized at different temperatures are shown in Figure 3(a),

1 clarifying the formation of g-C₃N₄ at temperatures higher than 500 °C. Notably, g-C₃N₄ was
 2 completely decomposed at 700 °C. The additional peaks at 450 °C are attributed to the melem
 3 derivatives (as a result of the low polycondensation of intermediate melamine).⁸⁴



4
 5 Figure 3. (a) XRD pattern of GCN samples produced at 500, 550, 600, and 650 °C.

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7 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra19586a>).

8 (b) Photo-absorption ability of GCN samples produced at 500, 550, 600, and 650 °C. Reproduced with permission

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10 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra19586a>).

11 (c) Band gap energy of GCN samples produced at 500, 550, 600, and 650 °C. Reproduced with permission from Ref.

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1 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra19586a>). (d) Photocurrent
2 response of GCN samples produced at 500, 550, 600, and 650 °C. Reproduced with
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4 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra19586a>).
5

6 Although a lumpy morphology was observed for g-C₃N₄ synthesized at 500 °C, it became
7 slacking by increasing the temperature, gradually resulting in the increase of specific surface
8 area. Noteworthy, few pores were developed on the surface at 600 °C, and many pores were
9 fused by increasing the temperature up to 650 °C. BET results also confirmed the higher
10 specific surface area of g-C₃N₄ (6.6 times) synthesized at 650 °C than that synthesized at 500
11 °C. The photo-absorption ability of g-C₃N₄ synthesized at 500, 550, 600, and 650 °C is shown
12 in Figure 3(b). Increasing the temperature from 500 to 650 °C resulted in the enhancement of
13 absorption in the visible-light region. This redshift could be caused by the enhancement of π -
14 plane conjugation degree and/or polymerization degree, significantly affecting the visible-light
15 absorption.⁸⁴ The same trend was also reported by Praus et al. for the condensation of melamine
16 products at different temperatures from 400 °C to 700 °C,⁶³ and Yan et al. for the condensation
17 of melamine products at 500, 520, 550, and 580 °C.⁸⁵ The band gap energies of GCN
18 synthesized at different temperatures are shown in Figure 3(c). Increasing the calcination
19 temperature from 500 to 550 and 600 °C resulted in the reduction of band gap energy from
20 2.83 to 2.78, and 2.71 eV, respectively. Further increase of the calcination temperature up to
21 650 °C led to the increase of band gap energy dedicated to the quantum confinement effect due
22 to the formation of a fluffier structure than that obtained at 600 °C. Notably, the
23 photoluminescence (PL) intensity of GCN samples decreased by increasing the temperature
24 where the lowest recombination rate of photo-generated e^-/h^+ pairs was achieved at 650 °C.⁸⁴
25 The same trend was also reported by Praus et al. for the condensation of melamine products at

1 different temperatures from 400 °C to 700 °C.⁶³ The transient photocurrent response of g-C₃N₄
 2 samples synthesized at 500, 550, 600, and 650 °C is compared in Figure 3(d), where the
 3 photocurrent response increased by increasing the temperature which was in good agreement
 4 with the results obtained from PL spectra. In summary, increasing the calcination temperature
 5 could improve both separation and transfer of photo-generated e⁻/h⁺ pairs.⁸⁴

6 Yousefzadeh et al. concluded that increasing the calcination temperature could enhance
 7 C/N molar ratio,⁸⁶ which was also suggested by Yan et al. for the condensation of melamine
 8 products at 500, 520, 550, and 580 °C.⁸⁵ Although the stoichiometric ratio of C/N for g-C₃N₄
 9 is 0.75, the highest obtained ratio of C/N was 0.71 (at 650 °C). This mismatch accompanied by
 10 the presence of hydrogen indicates the incomplete condensation of melamine and existence of
 11 some amino groups in the formed melons (even at high temperatures).⁸⁶ Notably, elemental
 12 analysis confirmed the presence of hydrogen in all melamine condensation products at different
 13 temperatures from 400 °C to 700 °C (Table 2).

14

15 Table 2. Elemental analysis and empirical composition of g-C_xN_yH_z. Reprinted with
 16 permission from Ref. ⁶³ Copyright 2017 Elsevier.

Temperature (°C)	C (wt. %)	N (wt. %)	H (wt. %)	x	y	z
400	33.0	64.3	2.7	5.5	9.2	5.4
450	34.5	63.5	2.0	5.7	9.1	3.9
550	35.4	63.1	1.5	5.9	9.0	2.9
575	35.6	63.0	1.4	5.9	9.0	2.9
600	35.9	62.8	1.3	6.0	9.0	2.5
650	36.2	62.6	1.1	6.0	8.9	2.2
700	36.3	62.6	1.1	6.0	8.9	2.2

17

1 Melem could be formed at a temperature of 400 °C. Deamination could be responsible for
2 the decrease of hydrogen content at temperatures of 450 to 500 °C, resulting in the formation
3 of a mixture of melon and melem. Melon is the main component of the melamine condensation
4 in the range of 550 to 575 °C. Further decrease of hydrogen at higher temperatures could be
5 devoted to the formation of polymerized melon and deamination. The high content of hydrogen
6 signifies the formation of melon-like $C_xN_yH_z$ sheets, with $-NH_2$ and $-NH$ groups, instead of
7 C_3N_4 sheets. The molar ratios of C/N for condensation of melamine products at different
8 temperatures from 400 °C to 700 °C are shown in Figure 2(d). Notably, the theoretical molar
9 ratios of C/N for melon, melem, and melam are 0.67, 0.60, and 0.54, respectively. The
10 theoretical molar ratio of C/N for g- C_3N_4 is 0.75 that was much higher than the maximum value
11 of C/N molar ratio for condensation of melamine products at different temperatures from 400
12 °C to 700 °C (as shown in Figure 2(d)), clarifying the incomplete condensation of amino groups
13 of melon. Overall, the formation of g- $C_3N_{4.5}H$ or $C_6N_9H_2$ is more probable than the formation
14 of g- C_3N_4 (even at high temperatures).⁶³

15

16 **3. Different morphologies of g- C_3N_4**

17 **3.1. Bulk g- C_3N_4**

18 As previously discussed, bulk g- C_3N_4 can be easily produced by thermal condensation of
19 different precursors (e.g. melamine, urea, thiourea, dicyandiamide, and cyanamide) between
20 400 and 600 °C. Overall, urea is considered an excellent precursor for the synthesis of flaky
21 GCN, providing high porosity and specific surface area. In general, the less the heating rate,
22 the higher is the porosity of g- C_3N_4 . It has been shown that crystallinity and specific surface
23 area could enhance, marginally, by increasing the annealing temperature.⁶⁵

24

25 **3.2. Three-dimensional g- C_3N_4 nanostructures**

1 Three dimensional (3D) nanostructures are usually produced by nanoscale building blocks, and
2 have received remarkable attention due to their unique properties (a combination of
3 straightforward molecular transport and high surface area).⁶⁵ Recently, Yuan et al. synthesized
4 3D porous thermally exfoliated g-C₃N₄ nanosheets for the photocatalytic degradation of
5 organic dyes. To prepare 3D porous g-C₃N₄, the as-synthesized g-C₃N₄ has been mixed and
6 stirred with concentrated H₂SO₄ for its intercalation. After freeze drying and calcination, the
7 light-yellow 3D porous g-C₃N₄ has formed. The formation of such a nanostructure significantly
8 increased the specific surface area from 12.0 (for bulk GCN) to 54.3 m²/g (for 3D-porous GCN)
9 due to the porousification properties of H₂SO₄. The photoluminescence spectrum of 3D-porous
10 g-C₃N₄ showed a drastic reduction of the emission intensity compared with that of bulk g-C₃N₄
11 that clarifies the promising potential of 3D structures for photocatalytic purposes (as confirmed
12 by the photocatalytic degradation of RhB).⁸⁷

13

14 **3.3. Two-dimensional g-C₃N₄ nanostructures**

15 As stated previously, GCN has a graphitic layer structure (the same as graphene) with weak
16 van der Waals interactions between its layers.⁸⁸ In general, the host layers of the layered
17 materials, including g-C₃N₄, are composed of energetic ionic bonds and/or strong covalent
18 bonds stacked with each other by electrostatic force with interlayer oppositely charged ions
19 and/or weak van der Waals force, providing the possibility of using various chemical and
20 physical processes to ruin such weak interlayer interactions for fabrication of sole sheets with
21 a few or single host layers called nanosheets. An important tip to synthesize g-C₃N₄ nanosheets
22 is that significantly weakening the interlayer van der Waals interactions should coincide with
23 maintaining the hydrogen bonding host layers.^{89, 90} Overall, there are two major strategies to
24 produce 2D GCN nanosheets including top-down and bottom-up approaches.⁶⁵ Notably, liquid
25 exfoliation (using ultrasonication), high-shear mixing and ball milling, chemical exfoliation,

1 thermal oxidation exfoliation, and gas bubble templating are major methods used in the top-
2 down category.

3

4 **3.3.1. Liquid exfoliation**

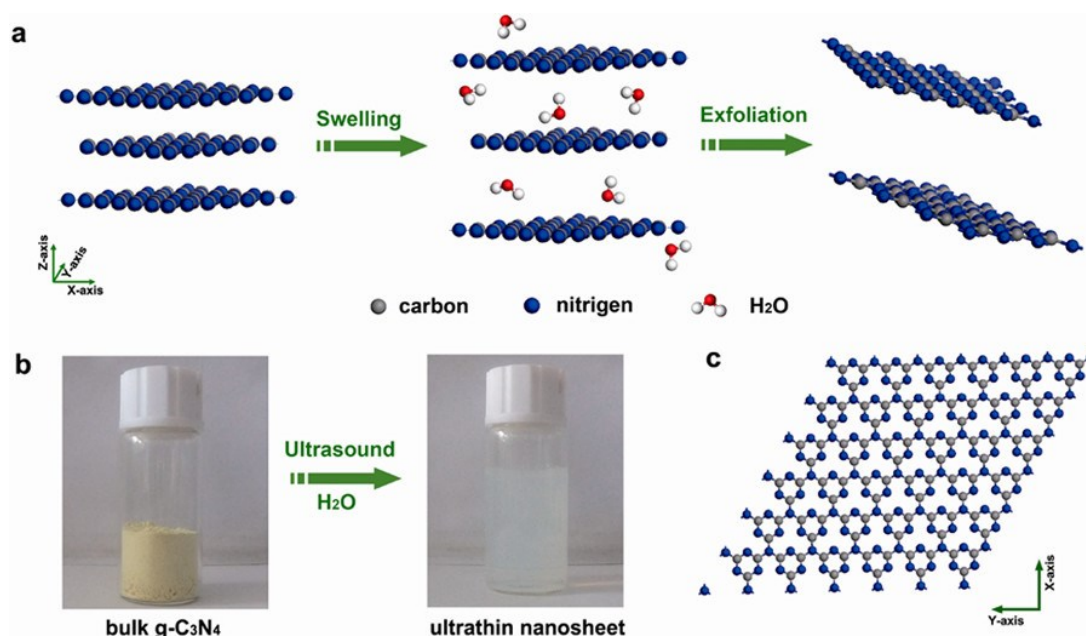
5 Liquid-Phase Exfoliation (LPE) is one of the most straightforward and effective methods to
6 weaken the van der Waals interactions.⁹¹ Two major methods have been used for exfoliation
7 of 2D materials including cavitation, which occurs in sonication, and shear forces employed in
8 both high-shear mixing and micro fluidization (under high shear rate).⁹² The ultrasound-
9 assisted method has been widely used for exfoliation of various layered materials such as
10 MoS₂,⁹³ MoSe₂,⁹⁴ BN,⁹⁵ WS₂,⁹⁶ graphene,⁹⁷ and g-C₃N₄.^{41, 90, 98} In this process, the sonication
11 power that generates cavitation bubbles induces chemical and/or physical changes in the
12 layered material. Rarefactions and compressions, originated from propagation of the ultrasonic
13 waves through the medium, exert low-pressure and high-pressure to pull and push molecules.
14 During rarefaction, formation and growing of microbubbles occurs. Their collapse, followed
15 by reaching to an unstable state, could generate strong shockwaves (with high temperature,
16 pressure, and speed) which produce normal and shear forces on 2D materials^{90, 92}. It has been
17 stated that minimizing the energy of exfoliation can maximize the concentration of dispersed
18 particles.⁹⁹ Thus, the solvent plays an important role in the ultrasonic exfoliation method.
19 Noteworthy, similar surface energy of the layered material and the solvent is more favorable.⁹⁰
20 Zhang et al. concluded that ultrathin g-C₃N₄ nanosheets can be easily produced by a green route
21 from liquid exfoliation of bulk g-C₃N₄ in water.⁸⁹ Owing to the presence of NH_x groups in the
22 structure of g-C₃N₄, originated from the polycondensation of precursors containing nitrogen
23 and carbon, the interlayer block of GCN could be easily swelled and/or intercalated by polar
24 solvents including ethanol, methanol, water, formamide, and DMF (*N,N*-dimethylformamide).
25 Notably, water exhibited the highest efficiency owing to its highest polarity (among these polar

1 solvents) and appropriate surface energy.⁹⁰ In another research, Zhang et al. dispersed ultrathin
2 negatively charged g-C₃N₄ nanosheets (with a zeta potential of around -30.3 mV) in water with
3 the concentration of around 0.15 mg mL⁻¹.⁸⁹ Hernandez et al. generated graphene using liquid
4 exfoliation of graphite. High exfoliation efficiency could be provided by minimizing the
5 enthalpy of mixing which depends on the balance of solvent and graphene surface energies.
6 The enthalpy of mixing (per unit volume) could be calculated by equation (3):¹⁰⁰

$$7 \frac{\Delta H_{mix}}{V_{mix}} \approx \frac{2}{T_{flake}} (\delta_G - \delta_{sol})^2 \phi \quad (3)$$

8 where δ_i is the square root of surface energy, ϕ is the volume fraction of graphene, ΔH_{mix} is
9 the enthalpy of mixing, and T_{flake} is the thickness of a graphene sheet. The energy (per unit
10 area) spent on peeling two flakes apart by overwhelming the van der Waals forces is considered
11 the surface energy of graphite. Hence, the minimal enthalpy of mixing could be achieved by
12 using solvents whose surface energies match graphene. It has been proposed that efficient
13 solvents could be characterized by surface tensions in the range of 40-50 mJ m⁻² (while the
14 surface energy of graphite is ~70-80 mJ m⁻²).¹⁰⁰ Notably, equation (3) has also been used by
15 Dong et al.⁹⁰ and Zhang et al.⁸⁹, suggesting water as an appropriate solvent for the liquid
16 exfoliation of GCN. Yang et al. dispersed GCN nanosheets in different solvents including
17 isopropanol (IPA), N-methyl-pyrrolidone (NMP), water, ethanol, and acetone. Notably, IPA
18 and NMP exhibited excellent potentials (even better than water) to disperse g-C₃N₄ nanosheets
19 and to exfoliate bulk GCN due to their appropriate surface energy of around 40 mJ m⁻². The
20 results showed that increasing the sonication time could lead to the formation of thinner
21 nanosheets. Compared with NMP, IPA could take advantages of its lower raw material cost
22 and boiling point.¹⁰¹⁻¹⁰³ It is notable that since slow evaporation of the solvent could enhance
23 the aggregation possibility of exfoliated GCN nanosheets, IPA could be more preferable than
24 NMP (owing to its easier removal).¹⁰¹ Schematic illustration of the fabrication process of GCN
25 nanosheets from bulk GCN by liquid exfoliation process (using IPA and NMP), photographs

1 of bulk GCN and suspension of GCN nanosheets, and perfect crystal structure of the GCN
2 nanosheets (along the z axis) are shown in Figure 4. It is worth mentioning that heteroatom-
3 mediated and template-assisted methods are among common bottom-up approaches.⁶⁵



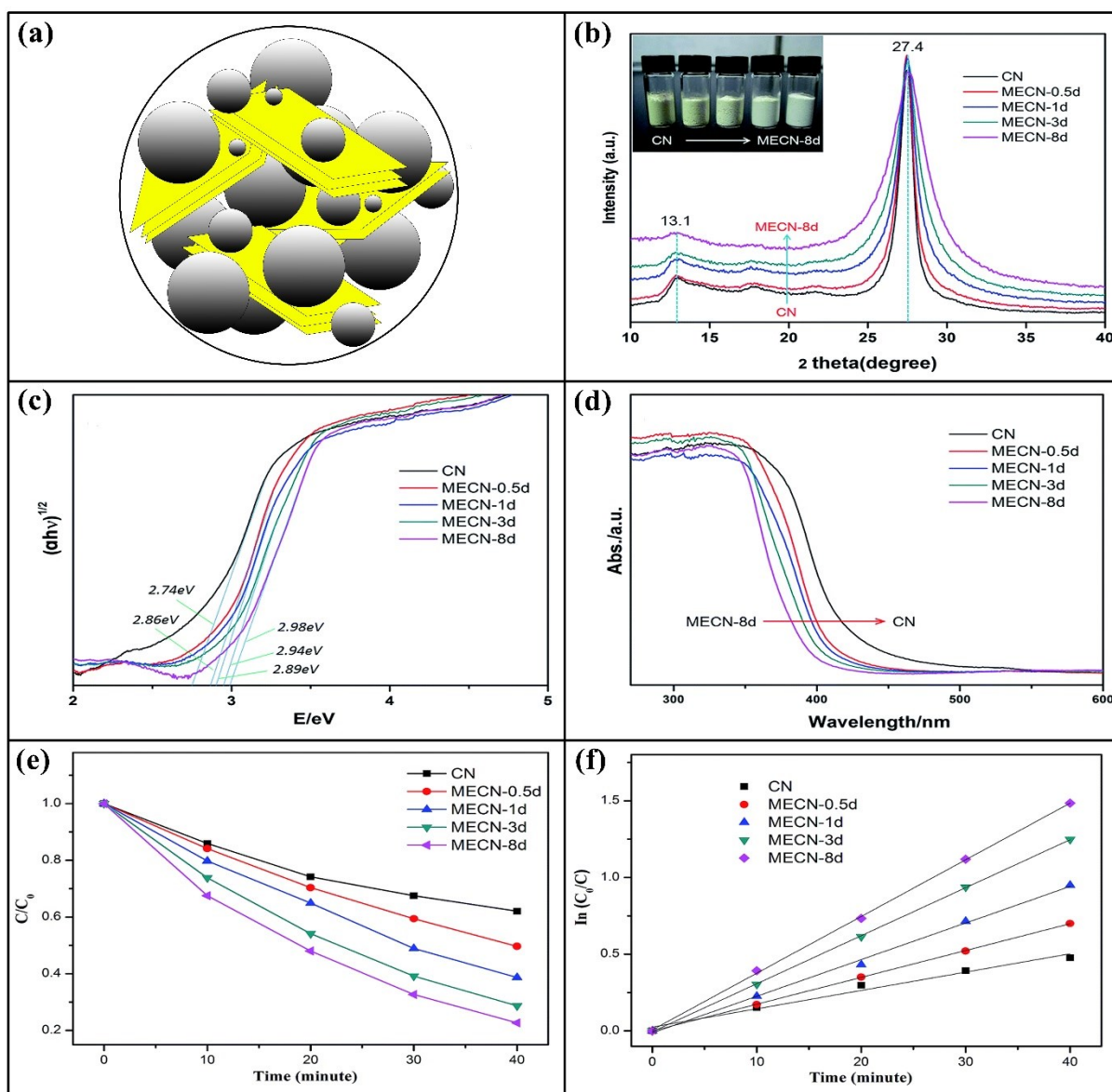
4
5 Figure 4. (a) Schematic illustration of the fabrication process of g-C₃N₄ nanosheets from bulk
6 g-C₃N₄ powder using liquid exfoliation. Reprinted with permission from Ref.⁸⁹ Copyright
7 2013 American Chemical Society. (b) Photograph of suspension of g-C₃N₄ nanosheets and
8 bulk g-C₃N₄. Reprinted with permission from Ref.⁸⁹ Copyright 2013 American Chemical
9 Society. (c) Perfect crystal structure of g-C₃N₄ nanosheets along the z-axis. Reprinted with
10 permission from Ref.⁸⁹ Copyright 2013 American Chemical Society.

11

12 3.3.2. High-shear mixing and ball milling

13 Although sonication of bulk GCN in appropriate solvents could result in the formation of
14 defect-free GCN nanosheets, the existence of planar hydrogen bonding between C-N layers of
15 GCN and the low energy of ultrasonication might limit its yield and scalability.¹⁰⁴ Liquid
16 exfoliation techniques tend to yield either no-defect (or low-defect) contents or high production
17 rates.¹⁰⁵ Besides, some methods could break the chemical structure, consequently, providing

1 low quality nanosheets.¹⁰⁴ Thus, shear mixing has been extensively used for the dispersion of
2 nanoparticles in liquids. Overall, high-shear mixing could be considered as a substantial
3 alternative to sonication for exfoliation of layered materials.¹⁰⁵ Yan et al. used high-shear
4 mixing for exfoliation of bulk g-C₃N₄ for the first time (water was used as the dispersion
5 agent).¹⁰⁴ Although the thickness of g-C₃N₄ nanosheets provided by sonication has been higher
6 than 2 nm in some cases,^{41, 106, 107} high-shear mixing can lead to the formation of nanosheets
7 with the uniform thickness of ~0.7 nm.¹⁰⁴ Since the theoretical thickness of one C-N layer is
8 ~0.325 nm,⁹⁰ it could be concluded that high-shear mixing is capable of producing GCN
9 nanosheets (with approximately one or two C-N layers), providing much more reactive sites
10 and higher specific surface area for photocatalytic purposes compared with general methods
11 including sonication. Due to the possibility of fabrication of defect-free and extremely thin
12 nanosheets, high-shear mixing could be considered a promising method for exfoliation of
13 GCN. Furthermore, it can be easily scaled-up to industrial levels.¹⁰⁴ Owing to its low cost and
14 facile operation, ball milling (mechanical milling) has also been employed as a simple method
15 for the synthesis of large quantities of some 2D materials (e.g. graphene and boron nitride).¹⁰⁸
16 Overall, it is considered an efficient strategy to produce ultrathin GCN with plentiful porosity
17 (due to the mechanical forces).¹⁰⁹ Ball milling process could be performed at dry or wet
18 condition. However, the properties of the final products could be different. Ma et al. used wet
19 ball milling to exfoliate GCN for the photocatalytic decomposition of organic pollutants,¹¹⁰
20 while Zhu et al. used dry ball milling for the scalable production of g-C₃N₄ nanosheets.¹⁰⁸ In
21 milling process, the friction and shear force of grinding balls gradually weakens the van der
22 Waals interactions,¹⁰⁸⁻¹¹⁰ whereas the strong sp² bonded in-plane structure could remain
23 stable.¹⁰⁸ The schematic illustration of the mechanical exfoliation of bulk g-C₃N₄ using ball
24 milling process is shown in Figure 5(a).



1
2 Figure 5. (a) Schematic illustration of the mechanical exfoliation of bulk g-C₃N₄ using ball
3 milling process. Redrawn from Ref. ¹⁰⁸ Copyright 2015 The Royal Society of Chemistry
4 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra09040g>). (b) XRD patterns of
5 GCN nanosheets after milling for 0.5, 1, 3 and 8 days. Reproduced with permission from Ref.
6 ¹⁰⁸ Copyright 2015 The Royal Society of Chemistry
7 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra09040g>). (c) Photo-adsorption
8 ability of GCN nanosheets after milling for 0.5, 1, 3 and 8 days. Reproduced with permission
9 from Ref. ¹⁰⁸ Copyright 2015 The Royal Society of Chemistry
10 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra09040g>). (d) Band gap energies of

1 GCN nanosheets after milling for 0.5, 1, 3 and 8 days. Reproduced with permission from Ref.

2 ¹⁰⁸ Copyright 2015 The Royal Society of Chemistry

3 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra09040g>). Effect of milling time on

4 the photocatalytic activity of GCN in terms of (e) C/C_0 and (f) $\ln(C_0/C)$ vs t (min).

5 Reproduced with permission from Ref. ¹⁰⁸ Copyright 2015 The Royal Society of Chemistry

6 (<https://pubs.rsc.org/en/content/articlelanding/2015/ra/c5ra09040g>).

7

8 The XRD patterns and color of g-C₃N₄ nanosheets after milling for 0.5, 1, 3, and 8 days
9 are shown in Figure 5(b). Increasing the milling time led to change of the color of g-C₃N₄ from
10 yellow to white, signifying change of the photo-absorption ability. Besides, increasing the
11 milling time has broaden widths of the diffraction peaks which can be attributed to either the
12 induced-strain or change of the crystallite size. The band gap energy and photo-absorption
13 ability of g-C₃N₄ at different milling times are shown in Figures 5(c) and 5(d), respectively. As
14 evident, the photo-absorption ability decreased by increasing the milling time. On the other
15 hand, the band gap energy increased which could be attributed to the quantum confinement
16 effect originated from the shift of conduction band and valence band edges in different
17 directions (due to change in the density of electronic states).¹⁰⁸ Yang et al. also observed the
18 same trend for g-C₃N₄ nanosheets synthesized by ultrasonication.¹⁰¹ The effect of milling time
19 on the photocatalytic activity of GCN is shown in Figures 5(e) and 5(f). The longer the milling
20 time, the higher photocatalytic activity is observed. Although ball milling led to a decrease in
21 the photo-absorption ability, the higher specific surface area of GCN nanosheets yielded a
22 higher photocatalytic activity than bulk GCN, clarifying the crucial importance of the surface
23 area in photocatalytic reactions.¹⁰⁸

24 Taking advantage of the layered structure of both MoS₂ and g-C₃N₄, Ansari et al. used dry
25 mechanical milling for the synthesis of large scale 2D MoS₂-GCN heterostructures for the

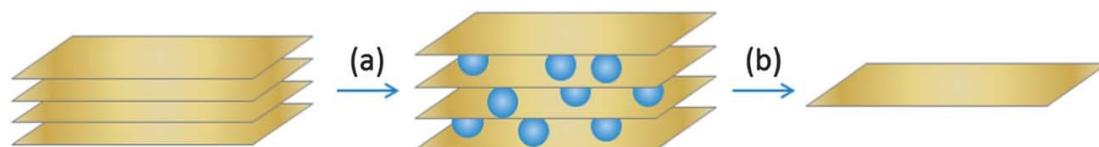
1 photocatalytic degradation of RhB under visible-light irradiation.¹¹⁶ The 2D MoS₂(10%)-GCN
2 heterostructure showed much higher photo-absorption ability than that of bare bulk g-C₃N₄.
3 Although the PL intensity of MoS₂(10%)-g-C₃N₄ heterostructure was lower than that of bulk
4 g-C₃N₄, the emission band of MoS₂(10%)-g-C₃N₄ heterostructure remarkably reduced
5 (compared with bare bulk g-C₃N₄). Overall, MoS₂(10%)-g-C₃N₄ heterostructure showed much
6 higher photocatalytic activity than both bare bulk g-C₃N₄ and bare g-C₃N₄ nanosheets.¹¹¹

7

8 **3.3.3. Chemical exfoliation**

9 Although ultrasonication has been widely used for the exfoliation of GCN, it suffers from its
10 low energy input and unsatisfactory efficiency, consequently. Overall, liquid phase exfoliation
11 using ultrasonication could not be considered as a potential method for mass production of
12 nanosheets.¹¹² Besides, it usually results in the formation of GCN nanosheets with larger
13 thicknesses than 2 μm.^{41, 106, 107} Compared with liquid phase exfoliation using ultrasonication,
14 chemical exfoliation has considerable advantages including the incorporation of guests
15 between the layers, and the formation of thinner nanosheets (unilamellar nanosheets). Besides,
16 it can take advantage of high efficiency of exfoliation and its cost-effectiveness.⁹⁰ H₂SO₄ is
17 considered a potential candidate for such a purpose owing to its mild oxidation effect (unlike
18 KMnO₄ as a strong oxidant).¹¹³⁻¹¹⁵ Although KMnO₄ has been used for exfoliation of graphite
19 in the Hummers method, generally, it cannot be used for exfoliation of GCN due to the weak
20 stability of GCN against strong oxidation effect of KMnO₄.¹¹³ Niu et al. showed that Hummers
21 method can only be used for the production of g-C₃N₄ with thickness of several hundred
22 nanometers (rather than nanosheets).¹¹⁶ Both diluted and concentrated sulfuric acid have been
23 used for chemical exfoliation of GCN. Although diluted H₂SO₄ was introduced as an “efficient
24 knife” to create large-aspect-ratio nanosheets, it provided the final yield of ~60%. Using
25 concentrated H₂SO₄ led to the formation of g-C₃N₄ nanosheets in a short time (< 30 min) with

1 the final yield of ~70%. Notably, this method is scalable and can be used for the fabrication of
2 uniform g-C₃N₄ nanosheets. Schematic illustration of the synthetic strategy of production of
3 GCN nanosheets using concentrated H₂SO₄ is shown in Figure 6.¹¹⁵ Although chemical
4 oxidation could be effectively used for exfoliation of g-C₃N₄, the thickness of GCN nanosheets
5 could be significantly affected by the procedure. For example, using H₂SO₄ for chemical
6 exfoliation of g-C₃N₄, as reported by Tong et al., led to the production of g-C₃N₄ nanosheets
7 with a thickness of ~80 nm.¹¹⁵ While Xu et al. produced monolayer GCN nanosheets by
8 chemical oxidation using H₂SO₄.¹¹³ Although chemical exfoliation is a rapid and facile method
9 for the production of GCN nanosheets, even after a short time around 1 min, it could suffer
10 from its low yield in some cases like that reported by Zou et al (i.e. ~30%).¹¹⁴ In addition, the
11 acid treatment involved could be cumbersome.¹¹⁵ On the other hand, ultrasonic exfoliation in
12 polar solvents takes a long time¹¹⁵ and its dispersion efficiency is low.¹¹⁷



13
14 Figure 6. Synthetic strategy of production of g-C₃N₄ nanosheets using concentrated H₂SO₄:
15 (a) Intercalation of H₂SO₄ (98 wt%) in the interplanar space of g-C₃N₄ and (b) Exfoliation of
16 intercalated g-C₃N₄ into a single layer. Reproduced with permission from Ref. ¹¹³ Copyright

17 2013 The Royal Society of Chemistry

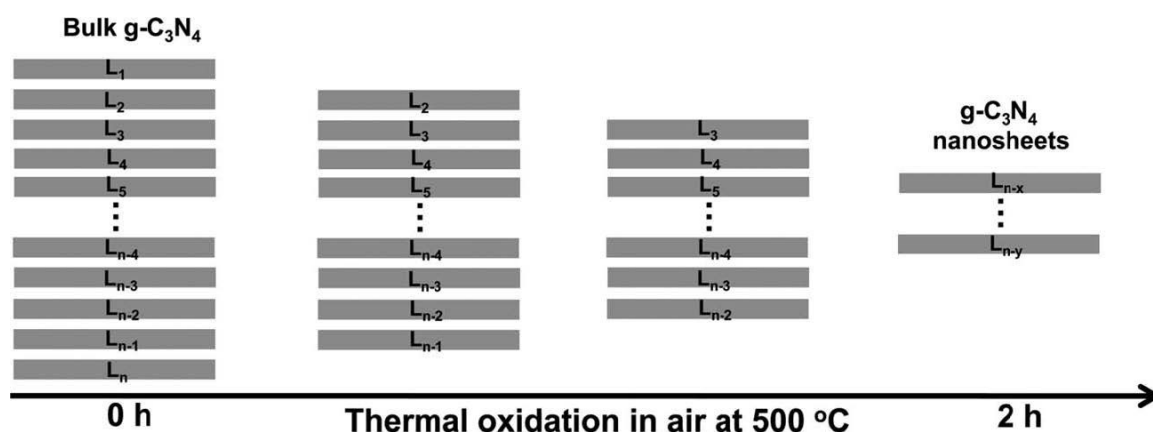
18 (<https://pubs.rsc.org/en/content/articlelanding/2013/ta/c3ta13188b>).

19

20 3.3.4. Thermal oxidation exfoliation

21 It has been shown that thermal oxidation of bulk GCN in air is another method for the
22 fabrication of g-C₃N₄ nanosheets. Firstly, Niu et al. developed a simple method called “direct
23 thermal oxidation etching” to create g-C₃N₄ nanosheets. Since strands of polymeric melon units
24 are connected to each other by hydrogen bond, they are not stable enough against oxidation in

1 air. Thus, they could be oxidized away gradually, from pristine $g\text{-C}_3\text{N}_4$ so that its thickness
 2 decreases to the nanometer scales. This is a low-cost method which can be easily scaled up.
 3 Besides, it takes advantage of its environmental friendliness (unlike chemical exfoliation).
 4 Notably, GCN nanosheets with a thickness of ~ 2 nm could be fabricated using this method.¹¹⁶
 5 The schematic illustration of the synthetic process of GCN nanosheets using thermal oxidation
 6 of pristine GCN in air is shown in Figure 7. Zhang et al. exfoliated $g\text{-C}_3\text{N}_4$ using various
 7 techniques. The yields of thermal, chemical, and ultrasonic exfoliation methods were 47.5%,
 8 58.1%, and 14.7%, respectively, whereas ultrasonic exfoliation in polar solvents using a
 9 mixture of ethanol and water (25:75, v/v) provided the lowest yield of all.¹¹⁸



10
 11 Figure 7. Schematic illustration of the synthetic process of GCN nanosheets using thermal
 12 oxidation of pristine GCN in air at 500 °C. Reproduced with permission from Ref. ¹¹⁶

13 Copyright 2012 John Wiley and Sons.

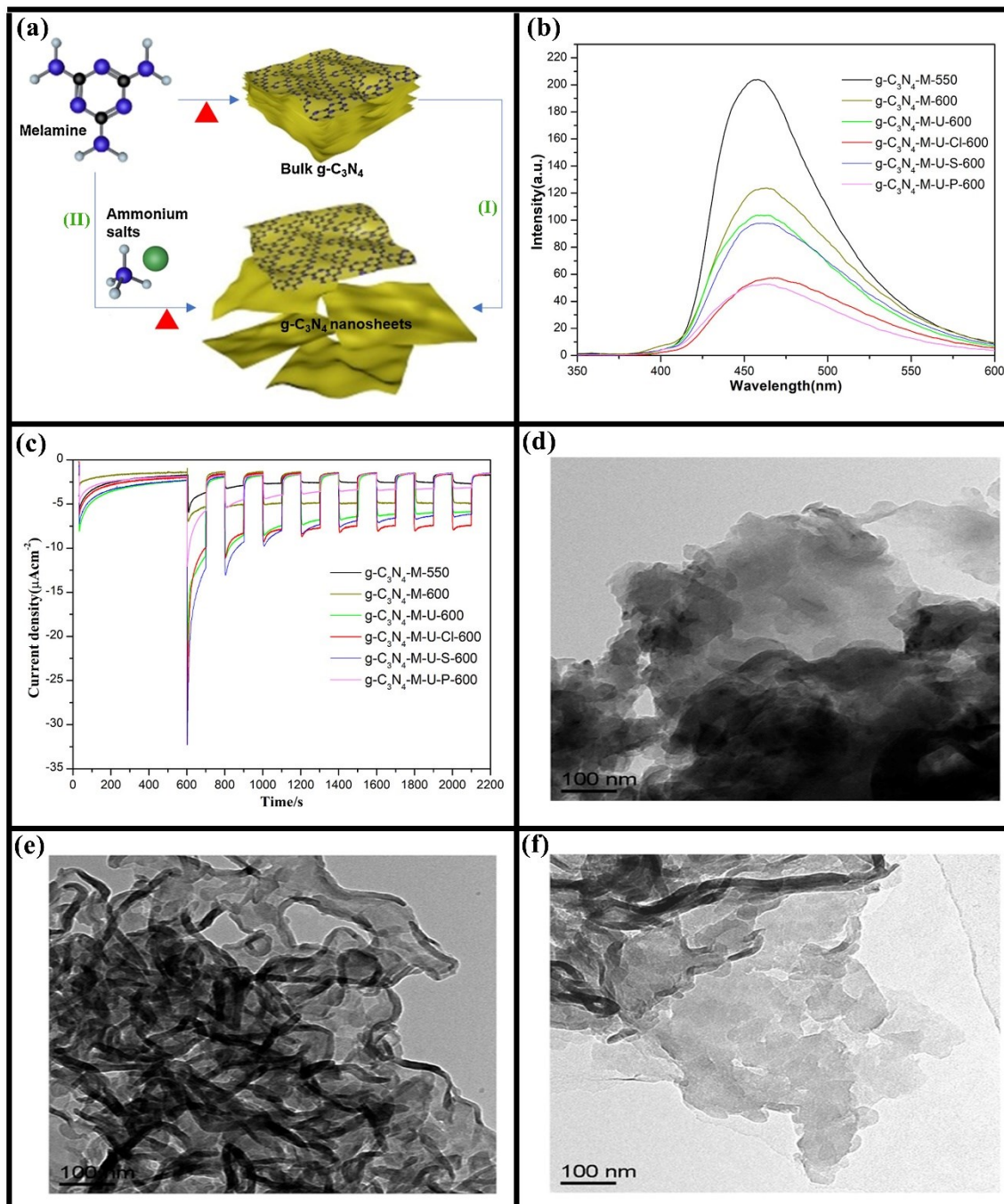
14 3.3.5. Gas bubble templating

15 Generally, a post-treatment such as ultrasonication, thermal oxidation, or chemical exfoliation,
 16 is required to produce GCN nanosheets from bulk GCN (Figure 8(a), pathway (I)). Besides,
 17 most of the current methods suffer from their prolonged duration, energy consumption, and
 18 low yield.¹¹⁹ As shown in Figure 8(a) pathway (II), the simple one-step exfoliation of the $g\text{-C}_3\text{N}_4$
 19 has been recently applied using different ammonium salts such as ammonium bromide,
 20

1 ammonium chloride, ammonium sulfate, ammonium phosphate, or ammonium fluoride for
2 large-scale production of GCN nanosheets.^{50, 53, 119-121} Noteworthy, the polymerization process
3 of melamine could be modified by generation of bubbles evolved from decomposition of the
4 additive ammonium salts, at various stages of polymerization.⁵³ Lu et al. used the mixture of
5 dicyandiamide (as the GCN precursor) and ammonium chloride (as the gas bubble template)
6 to synthesize crinkly 2D nanosheets. Notably, 2D-GCN nanosheets not only exhibited higher
7 BET surface area than bulk GCN, but also increased the electron transport ability and improved
8 the lifetime of charge carriers, resulting in the higher efficiency of GCN nanosheets than bulk
9 GCN for photocatalytic hydrogen production.¹²⁰ Depending on the type of ammonium salts,
10 their addition to the precursor of g-C₃N₄ can not only produce GCN nanosheets, but also
11 provide Br-, Cl-, P-, and F-doped GCN simultaneously.^{53, 119, 121} Notably, both Br- and Cl-
12 doped GCN nanosheets exhibited higher photo-absorption ability in the visible-light region
13 (wavelengths of 450-800 nm). Besides, they both yielded lower recombination rates of charge
14 carriers than bulk GCN, resulting in their higher photocatalytic activity for the degradation of
15 oxytetracycline antibiotic.¹¹⁹ In addition, F-doped GCN nanosheets showed lower photo-
16 absorption ability than bulk g-C₃N₄ in the visible region, signifying the partial destruction of
17 the conjugated system of GCN by F doping. However, due to the much stronger photocurrent
18 response of F-doped GCN nanosheets than bulk GCN, the photocatalytic production of H₂ gas
19 over F-doped GCN nanosheets was ~4 times higher than that over bulk GCN.¹²¹ Zhang et al.
20 designed a multi-stage polymerization to produce GCN nanosheets using the mixture of
21 melamine, urea, and volatile ammonium salts (ammonium phosphate, ammonium chloride, and
22 ammonium sulfate). Compared with the bulk g-C₃N₄ synthesized by melamine (denoted as g-
23 C₃N₄-M-600) with the BET surface area of 22.2 m² g⁻¹ and HER rate of 432.0 μmol h⁻¹ g⁻¹, the
24 GCN nanosheets obtained by the mixture of melamine and urea with the addition of ammonium
25 phosphate, ammonium chloride, and ammonium sulfate (denoted as g-C₃N₄-M-U-P-600, g-

1 C₃N₄-M-U-Cl-600, and g-C₃N₄-M-U-S-600 with large BET surface areas of 47.4, 103.3, and
2 53.5 m² g⁻¹, respectively) showed significantly higher photocatalytic activities for the
3 production of H₂ gas (i.e. 1193.7, 1853.8, and 1167.8 μmol h⁻¹ g⁻¹, respectively). Their
4 improved surface structure could be evidenced by PL and photocurrent spectra as shown in
5 Figures 8(b) and 8(c), consecutively. As evident, g-C₃N₄-M-U-P-600 and g-C₃N₄-M-U-Cl-600
6 catalysts exhibited the lowest PL intensity of all. As per g-C₃N₄-M-U-Cl-600, the creation of
7 thin nanosheet structure has been accounted for the improved separation of e⁻/h⁺ pairs, but as
8 per g-C₃N₄-M-U-P-600, the introduction of cyano groups to the structure has been considered
9 the main reason for the reduced recombination rate of charge carriers. Notably, the surface
10 charge distribution of the catalyst could be affected by the existence of cyano groups acting as
11 electron-withdrawing groups.⁵³ TEM images of GCN synthesized by melamine, the mixture of
12 melamine/urea (denoted as g-C₃N₄-M-U-600), and the mixture of melamine/urea/ammonium
13 chloride are compared in Figures 8(d), 8(e) and 8(f), respectively. Unlike bulk GCN with its
14 thick and dense layers, the addition of urea and ammonium chloride, acting as gas bubble-
15 assisted GCN, resulted in a lower thickness and a flaky structure with irregular or bubble-like
16 morphology. Although a mixture of melamine and urea, and a more complex mixture
17 (melamine, urea, ammonium chloride) yielded a crinkly structure, a typical ultrathin nanosheet-
18 like architecture was obtained by the latter mixture (melamine, urea, ammonium chloride),
19 clarifying the key role of volatile ammonium salts in modifying the morphology of GCN
20 obtained by thermal polymerization process. The surface charge of the g-C₃N₄-M-600, g-C₃N₄-
21 M-U-600, g-C₃N₄-M-U-P-600, g-C₃N₄-M-U-S-600, and g-C₃N₄-M-U-Cl-600 catalysts were -
22 31, -29, -38, -23, and -22 mv, respectively. Notably, the most negative surface charge was
23 recorded for g-C₃N₄-M-U-P-600, indicating the high impact of the incorporation of P atom on
24 the electronic properties of GCN. Once P is incorporated to the GCN framework, some
25 covalent bonds are formed between N neighbors and P atom (using four of its total five

1 electrons). The remained sole e^- of P atom could delocalize into the π -conjugated triazine ring,
2 generating an electron-rich state of the P-doped GCN. Considering the negative values of zeta
3 potentials, the repulsion interaction or electrostatic attraction could decrease the adsorption of
4 anionic organic dyes (like MO) and improve the adsorption of cationic organic dyes (like MB)
5 on the surface of catalysts as confirmed by evaluation of the organic dyes removal by
6 adsorption, where the removal efficiency of MB was significantly higher than that of MO by
7 using all catalysts. Based on their findings, the synergy of surface charge state and BET surface
8 area played an important role in the promoted decomposition performance (for MB, MO, and
9 RhB) and absorption of the synthesized GCN composites, i.e. g-C₃N₄-M-U-P-600, g-C₃N₄-M-
10 U-S-600, and g-C₃N₄-M-U-Cl-600, than g-C₃N₄-M-600.⁵³



1
 2 Figure 8. (a) Schematic illustration of the synthesis of GCN nanosheets by (I) post-treatment
 3 processes such as ultrasonication, thermal oxidation, or acid treatment of bulk GCN, and (II)
 4 single-step exfoliation by the addition of ammonium salts. Reprinted with permission from
 5 Ref. ¹¹⁹ Copyright 2019 Elsevier. (b) PL of GCN samples synthesized by melamine, the
 6 mixture of melamine/urea, and the mixture of melamine/urea/ammonium salts. Reprinted
 7 with permission from Ref. ⁵³ Copyright 2019 American Chemical Society

1 (<https://pubs.acs.org/doi/10.1021/acsomega.9b01510>). (c) Photocurrent spectra of GCN
2 samples synthesized by melamine, the mixture of melamine/urea, and the mixture of
3 melamine/urea/ammonium salts. Reprinted with permission from Ref. ⁵³ Copyright 2019
4 American Chemical Society (<https://pubs.acs.org/doi/10.1021/acsomega.9b01510>). TEM
5 images of GCN samples synthesized by (d) melamine, (e) the mixture of melamine/urea, and
6 (f) the mixture of melamine/urea/ammonium salts. Reprinted with permission from Ref. ⁵³

7 Copyright 2019 American Chemical Society

8 (<https://pubs.acs.org/doi/10.1021/acsomega.9b01510>).

10 **3.4. One-dimensional g-C₃N₄ nanostructures**

11 One dimensional (1D) GCN nanostructures refer to GCN nanotubes, GCN nanowires, GCN
12 nanobelts, and GCN nanorods.^{65, 122} Due to their high light harvesting, appropriate mass
13 transfer properties, and high specific surface area, these nanostructures are promising for
14 electrochemical and electronic applications.⁶⁵ Various methods have been used to fabricate
15 these nanostructures. Although template assisted methods are among common methods
16 employed for the fabrication of g-C₃N₄ nanorods (due to their ease), they need strong
17 alkali/acid to remove the templates.¹²³ Bai et al. used a simple reflux method to synthesize g-
18 C₃N₄ nanorods (without template). Notably, g-C₃N₄ nanorods showed remarkably higher
19 photocatalytic activity (~1.5 times) than that of g-C₃N₄ nanosheets, due to the elimination of
20 surface defects and enhancement of active lattice face.¹²⁴ Chemical vapor deposition (CVD) is
21 among efficient methods used for synthesis of g-C₃N₄ nanotubes. Gao et al. used TiO₂
22 nanotubes, as appropriate templates, for synthesis of g-C₃N₄-TiO₂ nanotube arrays using
23 chemical vapor deposition.¹²⁵ In general, CVD suffers from the toxicity of some by-products
24 evolved during the process, and high expenses (due to the use of high temperatures).¹²⁶ Overall,
25 using non-template methods is favored especially when removal of the template is needed.

1 Recently, Jiang et al. synthesized g-C₃N₄ nanotubes using thermal polymerizing reaction
2 (without any templates). It has been concluded that tubular morphology could accelerate
3 transfer of electrons along the axial direction which can result in the superior photocatalytic
4 activity of GCN nanotubes than that of GCN nanosheets.¹²⁷ Apart from nanorods, template
5 assisted method has also been used for the fabrication of GCN nanowires. Although template
6 assisted methods are the most frequently used methods employed for production of 1D GCN,
7 physical and chemical stimulants created by thermal treatment, ultrasonication, mechanical
8 force, and high pressure can also be used for exfoliation and re-assembling bulk g-C₃N₄ into
9 1D nanostructures.¹²³ 1D g-C₃N₄ nanostructures could yield high photocatalytic activity, but
10 they can suffer from the complex synthesis. Furthermore, it is worth mentioning that the
11 compactness of films obtained by 2D structures could be higher than that achieved by 1D
12 structures, as reported by Mohammadi et al. in the case of ZnO nanosheets and ZnO nanorods
13 deposited by electrophoretic deposition.¹²⁸ Thus, 2D GCN nanostructures could be more
14 promising for immobilization purposes.

15

16 **3.5. Zero-dimensional g-C₃N₄ nanostructures**

17 Zero-dimensional (0D) g-C₃N₄ nanoparticles could shorten the length of charge transfer.
18 Besides, they exhibit more rich surface active sites for catalytic purposes.¹²⁹ 0D GCN refers to
19 GCN quantum dots (GCN QDs) where photogenerated electron-hole pairs are not able to
20 transfer freely.¹²³ GCN nanostructures with the size of below 10 nm exhibit remarkable
21 quantum confinement effects. Moreover, they usually provide unique properties including non-
22 toxicity, bright fluorescence, and water solubility. Both top-down and bottom-up methods have
23 been used to synthesize g-C₃N₄ QDs.⁶⁵ In top-down methods, the synthesis of g-C₃N₄ QDs
24 starts with macroscopic g-C₃N₄ structures which is followed by some processes including
25 evaporation-condensation, sonication, hydrothermal treatment, and hydrolysis methods to

1 fabricate 2D/1D GCN nanostructures and finally GCN QDs. In bottom-up methods, carbon
2 and nitrogen rich precursors are used in some processes such as microwave-assisted methods,
3 solid phase reactions, and quasi-chemical vapor deposition to synthesize GCN QDs.¹³⁰
4 Although g-C₃N₄ nanomaterials indicated great potential in the field of photocatalysis, general
5 synthetic methods for 0D g-C₃N₄ nanoparticles suffer from tedious cleaning processes and
6 complex oxidation-reduction reactions, usually, resulting in their low yields and prolonged
7 synthesis. Noteworthy, 0D g-C₃N₄ nanoparticles generally have wider band gap energy than
8 2D/3D g-C₃N₄ materials, originated from the quantum size effect, limiting their visible-light
9 harvest. Besides, the complex structure of 0D g-C₃N₄ nanoparticles has challenged systemic
10 evaluation of the photocatalytic mechanism.¹²⁹

11

12 **4. Strategies to overcome the drawbacks of pristine g-C₃N₄**

13 Despite its advantages, the photocatalytic activity of bare g-C₃N₄ is not satisfactory due to the
14 three major reasons as follows:^{131, 132}

- 15 i. Fast recombination of the photogenerated electron-hole pairs that results in low
16 quantum efficiency,
- 17 ii. Inefficient absorption of solar light, and
- 18 iii. Low surface area (by using some precursors such as melamine) and non-textured pores
19 due to the high degree of condensation of monomers.

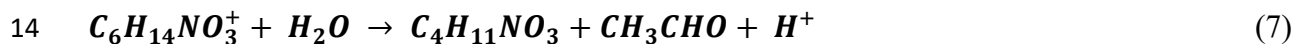
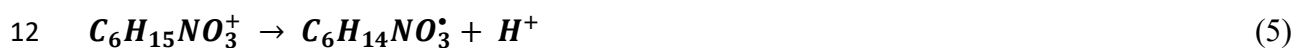
20 The recombination rate of photogenerated e⁻/h⁺ pairs should be minimized to achieve high
21 photocatalytic activity for decontamination of organic pollutants. For such a purpose, several
22 strategies could be considered including using the mixed precursors of GCN, heterojunction
23 structures, applying a bias potential, and element doping.

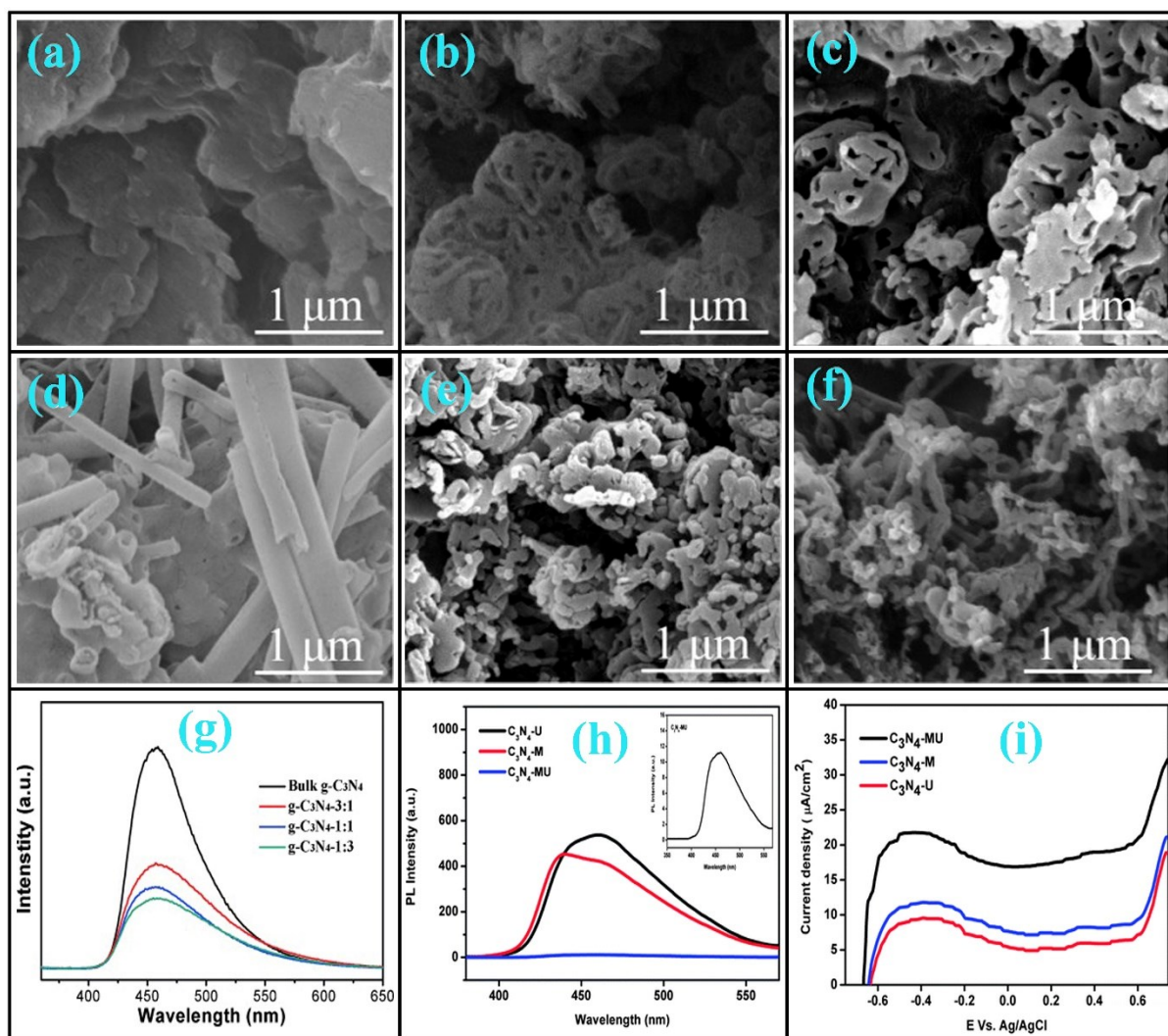
24

25 **4.1. Multi-component precursors of GCN for generation of supramolecular aggregates**

1 The formation of supramolecular aggregates is a potential approach to overcome the
2 shortcomings of GCN.¹³³⁻¹³⁶ Liao et al. tailored to the morphology of g-C₃N₄ by self-assembly
3 and calcination using the mixtures of melamine and urea with different molar ratios of 1:3, 1:2,
4 1:1, 2:1, and 3:1, and the fixed amount of cyanuric acid. The SEM images of bulk GCN
5 synthesized by melamine and GCN-x catalysts synthesized using a mixture of melamine and
6 urea (x denotes the molar ratio of melamine to urea) are compared in Figure 9(a-f). As evident,
7 the morphology of bulk GCN is thoroughly different from those obtained by using the mixture
8 of melamine and urea. Notably, GCN-3:1 provided a hollow structure with BET surface area
9 of 71.6 m² g⁻¹, while GCN-1:1 and GCN-1:3 yielded tube-embedded and wormlike
10 morphologies with BET surface areas of 49.6 and 97.4 m² g⁻¹, consecutively. Overall, the
11 morphology of GCN significantly varied from hollow to wormlike structure by tuning the
12 molar ratio of melamine to urea. Noteworthy, due to the shift of the valence and conduction
13 band edges in opposite directions, originated from the quantum confinement effect, GCN-x
14 catalysts exhibited marginally higher band gap energies than that of bulk GCN. The
15 photocatalytic degradation of RhB over all GNC-x catalysts was higher than that over bulk
16 GCN with the highest apparent rate constant of 0.1746 min⁻¹ for GCN-1:3, owing to its high
17 BET surface area and low recombination rate of charge carries evidenced by PL spectra in
18 Figure 9(g).¹³⁷ Martha et al. have also evaluated the effect of mixing urea and melamine on the
19 photocatalytic production of hydrogen in an aqueous solution with 10 vol% of TEA solution
20 as sacrificial electron donor. The bandgap energies of GCN synthesized by urea, melamine,
21 and the unimolar mixture of urea/melamine were 2.72, 2.70, and 2.71 eV; and the BET surface
22 areas were 50, 30, and 10 m² g⁻¹, respectively. Compared with GCN synthesized by urea and
23 melamine with generation of 25 and 30 μmol H₂ gas, that synthesized by the unimolar mixture
24 of urea and melamine exhibited the highest efficiency (48 μmol), which could be highly related
25 to the significantly reduced recombination rate of charge carriers as confirmed by PL (Figure

1 9(h)) and photocurrent spectra (Figure 9(i)). Further improvement was carried out by loading
 2 Pt (1 wt%), as a co-catalyst, on the surface of catalysts where the evolution rate of H₂ increased
 3 from 48 μmol to 135 μmol over the Pt-loaded GCN catalyst synthesized by the unimolar
 4 mixture of melamine and urea. It should be noted that the layers of GCN sheets were well
 5 separated using the mixture of urea and melamine. Hence, TEA solution could easily penetrate
 6 between the GCN layers, facilitating the reduction reaction. Upon light irradiation, the
 7 photogenerated h⁺ in the VB of GCN could oxidize and decompose TEA, while the
 8 photogenerated e⁻ in the CB of GCN could reduce H⁺ to H₂ through Pt loaded on the surface of
 9 catalyst.⁸² The proposed reactions of TEA, acting as sacrificial electron donor, in H₂ generation
 10 systems as provided in equations (4)-(7):^{2, 138}





1
2 Figure 9. SEM images of (a) bulk GCN, (b) GCN-3:1, (c) GCN-2:1, (d) GCN-1:1, (e) GCN-
3 1:2, and (f) GCN-1:3 samples. Reproduced with permission from Ref. ¹³⁷ Copyright 2014
4 John Wiley and Sons. (g) Photoluminescence spectra of bulk GCN, GCN-3:1, GCN-1:1, and
5 GCN-1:3 samples. Reproduced with permission from Ref. ¹³⁷ Copyright 2014 John Wiley
6 and Sons. (h) Photoluminescence spectra of GCN synthesized by urea, melamine, and their
7 mixture. Reproduced with permission from Ref. ⁸² Copyright 2013 The Royal Society of
8 Chemistry (<https://pubs.rsc.org/en/content/articlelanding/2013/TA/c3ta10851a>). (i)
9 Photocurrent spectra of GCN synthesized by urea, melamine, and their mixture. Reproduced
10 with permission from Ref. ⁸² Copyright 2013 The Royal Society of Chemistry
11 (<https://pubs.rsc.org/en/content/articlelanding/2013/TA/c3ta10851a>).

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The beneficial effect of using the combined precursors of g-C₃N₄ on the photocatalytic activity was also reported by Zhang et al. by mixing urea and thiourea where the optimal rate of HER over the GCN synthesized by the mixture (2.44% mass fraction of thiourea) was 2.60 and 6.17 times higher than those synthesized by urea and thiourea, respectively. Notably, both enlarged BET surface area and reduced recombination rate of charge carriers were responsible for the superior photocatalytic activity of GCN synthesized by the mixture of urea and thiourea.¹³⁹ It is obvious the effect of combined precursors of g-C₃N₄ on BET surface area of the final product highly depends on the type of precursors and their weight/molar ratio. In another research, the mixture of melamine and cyanuric acid (5 and 5.1 g, respectively) was used to synthesize GCN where BET surface area remarkably increased from 7 to 70 m² g⁻¹ for the g-C₃N₄ synthesized by melamine and the mixture of melamine and cyanuric acid, consecutively, which is related to the formation of nanoporous GCN layers in the presence of cyanuric acid.¹⁴⁰ Recently, Zhang et al. used a mixture of urea (15 g) and melamine (6 g) to synthesize GCN for the photocatalytic production of H₂ gas. The photocatalytic production rate of hydrogen gas by using GCN synthesized from the mixture of urea and melamine (1395.4 μmol h⁻¹ g⁻¹) was remarkably higher than that synthesized by melamine (432.0 μmol h⁻¹ g⁻¹). For further improvement, 4g ammonium chloride was added to the mixture of urea and melamine. Furthermore, the addition of ammonium chloride significantly increased the photocatalytic production rate of H₂ gas (1853.8 μmol h⁻¹ g⁻¹) which is related to the improved BET surface area (originated from the production of GCN nanosheets), surface charge state, and surface structure of the synthesized GCN nanosheets.⁵³

24 **4.2. Element doping and defects engineering**

1 By tuning the bandgap structure, doping heteroatoms could extend the light harvest and fasten
2 the mobility of charge transfer.¹²¹ Therefore, to improve the photocatalytic efficiency of GCN,
3 various metal and non-metal elements (cations and anions, respectively) have been
4 incorporated into its lattice. However, the mechanism of tuning the bandgap structure highly
5 depends on the type of dopant. For instance, B doping shifts the valence band towards more
6 positive values,¹⁴¹ whereas P doping shifts the valence band towards more negative values.¹⁴²
7 Notably, the intrinsic Lewis base sites (in the form of imine or amine groups) together with the
8 formation of Lewis acid sites originated from the generation of P⁺ centers by doping
9 phosphorus, as illustrated in Figure 10(a), could result in delocalization of lone electrons,
10 favoring the fast separation of charge carriers.¹⁴³ Simultaneously, creation of the defective
11 bands within the band gap of GCN by phosphorus doping could increase the visible-light
12 harvest and narrow its band gap energy,¹⁴⁴ resulting in the remarkable increase of the
13 photocatalytic activity of P-doped GCN compared to that of bare g-C₃N₄.^{142, 145, 146} In addition,
14 phosphorus doping could significantly enhance the low yield of GCN by using urea which is
15 of the major drawbacks associated with using this precursor (despite of its several advantages).
16 For instance, the yield of GCN remarkably increased from 3% (for bare GCN) to 25% (for P-
17 doped GCN) by the addition of 10wt% hexachlorocyclotriphosphazene (HCCP) as the
18 phosphorus source.⁷⁹ Thus, it is considered a superior and ideal doping element to control the
19 surface properties (texture) and electronic structure of GCN.^{144, 147, 148} The photocatalytic
20 production rates of H₂ gas over bare GCN and P-doped GCN are compared in Table 3.

Table 3. Comparison of the photocatalytic hydrogen production between bare g-C₃N₄ and element-doped g-C₃N₄.

Element doped	Photocatalytic H ₂ evolution ($\mu\text{mol h}^{-1} \text{g}^{-1}$)				Light source	Apparent quantum yield (AQY) for element-doped GCN (%)	Reference
	Bare GCN	Element-doped GCN	Amount of Pt as co-catalyst (wt%)	Type and amount of sacrificial agent (vol%)			
P	60	570	1	Methanol (20)	300 W Xenon lamp	-	146
P	107	318	1	TEA (10)	300 W Xenon lamp	-	147
P	~174	506	3	TEA (10)	300 W Xenon lamp	-	143
P	208	5128	3	TEA (10)	300 W Xenon lamp	-	142
P	430	2020	3	TEA (10)	300 W Xenon lamp	4.32 ($\lambda = 420 \text{ nm}$)	145
B	330	910	1	TEA (10)	150 W Xenon lamp	-	149
S	330	615	1	TEA (10)	150 W Xenon lamp	-	149
P	330	575	1	TEA (10)	150 W Xenon lamp	-	149

S	449.86	1186.38	3	TEA (10)	300 W Xenon lamp	10.2 ($\lambda = 420$ nm)	150
S	68	572	3	TEA (10)	300 W Xenon lamp	0.39* ($\lambda = 420$ nm)	151
S	20.3	121.6	1	Methanol (25)	300 W Xenon lamp	2.6 ($\lambda = 420$ nm)	152
Ni	98.67	155.71	~3	TEA (10)	300 W Xenon lamp	-	153
Zn	27.5	297.5	0.5	Methanol (~18.5)	200 W Xenon lamp	3.2 ($\lambda = 420$ nm)	154
Co	190	560	3	TEA (10)	300 W Xenon lamp	-	155
Na	96	~144	3	TEA (10)	300 W Xenon lamp	-	156
K	96	~606	3	TEA (10)	300 W Xenon lamp	-	156
Mo	0	0	0	Methanol (10)	300 W Xenon lamp	-	157
S	0	0	0	Methanol (10)	300 W Xenon lamp	-	157
Mo and S	0	294	0	Methanol (10)	300 W Xenon lamp	0.24 ($\lambda = 420$ nm)	157
P and O	86	159	0**	TEA (~14.3)	300 W Xenon lamp	-	158

1 *: Extracted from the publication by Plot Digitizer

2 **: 5 mg of Erythrosin B was added to the initial aquatic solution as sensitizer

3

1 As evident, phosphorus doping has significantly improved the photocatalytic evolution rate
2 of hydrogen gas in all cases. Guo et al. fabricated P-doped tubular GCN for photocatalytic
3 production of hydrogen using several phosphorus sources including sodium pyrophosphate,
4 ammonium phosphate, sodium hypophosphite, and sodium phosphite. Notably, all synthesized
5 P-doped GCN catalysts by these phosphates (CN-SP, CN-AP, CN-SH, CN-SPh, respectively)
6 showed higher photocatalytic activity than that of bare GCN (Figure 10(b)). Among these
7 phosphates, the highest efficiency was obtained by sodium pyrophosphate which has been
8 attributed to the high amount of phosphorus doped. Furthermore, the separation of charge
9 carriers increased and the visible-light harvest improved by using sodium pyrophosphate as
10 confirmed by UV-vis DRS spectra and Electrochemical Impedance Spectroscopy (EIS)
11 Nyquist plots in Figures 10(c) and 10(d), respectively. Considering computational results,
12 projected density of states (PDOSs) changed a lot by P-doping, and the CB shifted to a position
13 with a lower energy, resulting to the reduction of the band gap energy confirmed by UV-vis
14 DRS spectra. Based on their calculations, both introduction of phosphorus to the CB and the
15 formation of surface carbon defects could be responsible for the reduced band gap energy by
16 P-doping. The bulk C/N mass ratios of bulk GCN and CN-SP (obtained by the Element
17 Analyzer) were both 0.68, while the corresponding surface C/N atomic ratios (obtained by
18 XPS) were 0.74 and 0.69, clarifying the formation of carbon defects only on the surface of
19 GCN. The co-introduction of surface carbon defects and P-doping with enlarged BET surface
20 area ($28.2 \text{ m}^2 \text{ g}^{-1}$ against $8.6 \text{ m}^2 \text{ g}^{-1}$) have been accounted for the considerably higher
21 photocatalytic activity of CN-SP than bulk GCN.¹⁴⁶ Recently, Yang et al. synthesized 3D
22 flower-like P-doped GCN (Figure 10(e)) with a high content of porosity and surface area using
23 the mixture of cyanuric acid and melamine as the supramolecular precursor of g-C₃N₄ and
24 phosphoric acid as the phosphorus source. The beneficial effects of combined precursors for
25 synthesis of GCN were explained in the previous section. BET specific surface area of bulk

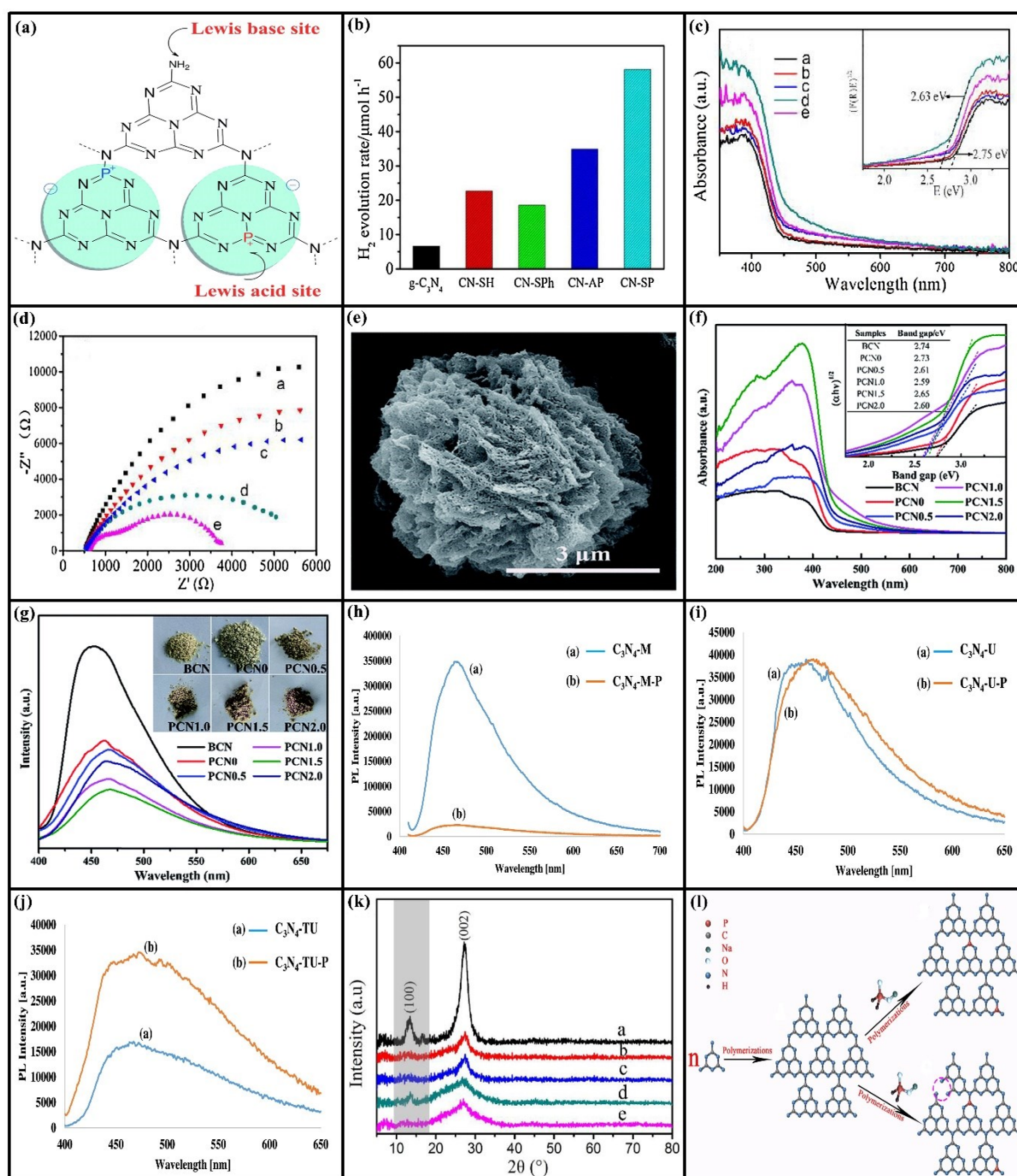
1 GCN (synthesized by melamine) and P-doped GCN catalysts (synthesized by the mixture of
2 cyanuric acid and melamine and different amounts of phosphoric acid from 0 to 0.5, 1.0, 1.5,
3 and 2.0 g) were 9.87, 96.12, 126.25, 235.85, 222.12, and 152.00 m² g⁻¹, consecutively. Notably,
4 using the combined precursor significantly increased BET surface area from 9.87 (for GCN
5 synthesized by melamine) to 96.12 m² g⁻¹ (for GCN synthesized by the mixture of melamine
6 and cyanuric acid). Furthermore, phosphorus doping using phosphoric acid remarkably
7 increased BET surface area compared to bare GCN. The unique structure associated with
8 phosphorus doping resulted in the improved visible-light harvest (Figure 10(f)) and the
9 increased separation efficiency of charge carriers (Figure 10(g)), remarkably promoting the H₂
10 evolution rate over the optimal P-doped g-C₃N₄ catalyst, fabricated by the addition of 1.5 g
11 phosphoric acid, compared with bare g-C₃N₄ (almost 24-fold). Notably, Pt nanoparticles (3
12 wt%) were deposited on the surface of the as-prepared GCN samples (as co-catalyst). The
13 remarkably higher surface reactivity and surface area provided by the distinctive hierarchical
14 structure containing ultrathin porous nanosheets resulted in the much higher photocatalytic
15 activity of the optimal P-doped g-C₃N₄ than bare GCN.¹⁴² It should be noted that the effect of
16 phosphorus doping on BET surface area largely depends on the type of precursors used for
17 synthesis of GCN and the phosphorus source. Bellardita et al. used ammonium dihydrogen
18 phosphate to synthesize P-doped GCN using three different precursors of urea, melamine, and
19 thiourea. The BET surface areas of GCN synthesized by urea, melamine, and thiourea were 28,
20 7 and 5 m² g⁻¹, which were changed to 16, 4 and 28 m² g⁻¹ respectively with phosphorus doping.
21 Notably, depending on the type of precursor, doping phosphorus could affect the textural
22 properties of the obtained GCN. It has been proposed that the effects of evolution of H₂S and
23 CS₂ gases during condensation of thiourea is not similar to the effect of evolution of CO₂ during
24 condensation of urea which is responsible for the superior BET surface area of g-C₃N₄
25 synthesized by urea compared to that synthesized by thiourea. Thus, the probable reaction of

1 H₂S and CS₂ with oxygen of the phosphorus source at high temperatures might generate
2 volatile gases, initializing production of pores during synthesis of P-doped GCN using thiourea.
3 In addition to BET surface area, the recombination rates of charge carriers are highly dependent
4 on the type of precursor of g-C₃N₄. As evidenced by PL spectra in Figure 10(h-j), among P-
5 doped GCN catalysts obtained by urea, melamine, and thiourea, only that synthesized by
6 melamine significantly quenched the recombination rate of charge carries.¹⁴⁰

7 More recently, Lin et al. synthesized P-doped g-C₃N₄ nanosheets using sodium
8 hypophosphite (as the phosphorus source) for the photocatalytic hydrogen production. Pt
9 nanoparticles (1 wt%) were deposited on the surface of the as-prepared samples (as co-
10 catalyst). As expected, all P-doped GCN catalysts with different phosphorus contents exhibited
11 higher efficiencies than bare GCN. Furthermore, it has been concluded that phosphorus doping
12 could result in the formation of carbon defects by cleavage of the N-C bond of C=N-C,
13 suppressing the recombination rate of photogenerated e⁻/h⁺ pairs and improving the separation
14 of conjugated π electrons in the structure of g-C₃N₄.¹⁴⁷ Creation of surface carbon defects and
15 their beneficial effects on the photocatalytic activity of P-doped g-C₃N₄ has also been reported
16 by other researchers.¹⁴⁶ The reduced crystallinity of P-doped GCN compared to bare GCN is
17 evidenced by XRD patterns shown in Figure 10(k). The significantly improved photocatalytic
18 H₂ production of the optimal P-doped GCN compared to bare g-C₃N₄ (318 against 107 μmol
19 $\text{h}^{-1} \text{g}^{-1}$, respectively) was attributed to the beneficial effects of both phosphorus doping and
20 carbon defects. It has been proposed that P atoms replaced C atoms in C-N bond, though
21 partially, during P doping (considering XPS analysis).¹⁴⁷ Schematic structures of GCN and P-
22 doped GCN with and without the existence of carbon defects are compared in Figure 10(l).

23 In addition to the photocatalytic evolution of H₂ gas, P-doped g-C₃N₄ has been used for the
24 photocatalytic degradation of organic pollutants (e.g. RhB) where the photocatalytic
25 degradation efficiency of P-doped g-C₃N₄ was nearly 3 times higher than that of bare GCN. It

1 is thought that the sole e^- of P could delocalize to the π -conjugated triazine ring once C is
 2 replaced by P in the GCN framework during P doping. Finally, P^+ centers could be formed as
 3 the result of the sole e^- delocalization and the built-in P-N bond polarization. Hence, the fast
 4 separation of charge carriers could be achieved by the existence of P^+ centers acting as Lewis
 5 acid sites (in addition to the inherent Lewis base sites in GCN as imine or amine groups),
 6 improving the photocatalytic performance in either HER or RhB decomposition.¹⁴³



7

1 Figure 10. (a) Effects of phosphorus doping on the chemical and electronic properties of
2 GCN. Reproduced with permission from Ref. ¹⁴³ Copyright 2015 The Royal Society of
3 Chemistry (<https://pubs.rsc.org/en/content/articlelanding/2015/ta/c4ta05292g>). (b)
4 Photocatalytic activity of bulk GCN and P-doped GCN synthesized by various phosphorus
5 sources. Reprinted with permission from Ref. ¹⁴⁶ Copyright 2017 Elsevier. (c) UV-vis DRS
6 spectra of bulk GCN and P-doped GCN synthesized by various phosphorus sources (a: GCN
7 bulk, b: CN-SH, c: CN-SPH, d: CN-AP, and e: CN-SP). Reprinted with permission from Ref.
8 ¹⁴⁶ Copyright 2017 Elsevier. (d) EIS Nyquist plots of bulk GCN and P-doped GCN
9 synthesized by various phosphorus sources (a: GCN bulk, b: CN-SH, c: CN-SPH, d: CN-AP,
10 and e: CN-SP). Reprinted with permission from Ref. ¹⁴⁶ Copyright 2017 Elsevier. (e) SEM
11 image of 3D flower-like P-doped GCN. Reproduced with permission from Ref. ¹⁴² Copyright
12 2018 The Royal Society of Chemistry
13 (<https://pubs.rsc.org/en/content/articlelanding/2018/ta/c8ta05723k>). Comparison of (f) UV-
14 vis DRS and (g) PL spectra of bulk GCN (BCN) and P-doped GCN (PCN_x where x clarifies
15 the amount of phosphoric acid). Reproduced with permission from Ref. ¹⁴² Copyright 2018
16 The Royal Society of Chemistry
17 (<https://pubs.rsc.org/en/content/articlelanding/2018/ta/c8ta05723k>). PL spectra of GCN and
18 P-doped GCN synthesized by (h) melamine, (i) urea, and (j) thiourea. Reprinted with
19 permission from Ref. ¹⁴⁰ Copyright 2018 Elsevier. (k) XRD patterns of bulk GCN and P-
20 doped GCN synthesized by various phosphorus sources (a: GCN bulk, b: CN-SH, c: CN-
21 SPH, d: CN-AP, and e: CN-SP). Reprinted with permission from Ref. ¹⁴⁶ Copyright 2017
22 Elsevier. (l) Schematic structures of GCN and P-doped GCN with and without the existence
23 of carbon defects. Reprinted with permission from Ref. ¹⁴⁷ Copyright 2020 Elsevier.

24

1 Overall, the introduction of non-metals to GCN could break its symmetry, resulting in the
2 faster separation rate of charge carriers.¹⁵⁹ Recently, halogen doping (F, Cl, Br, and I) has been
3 effectively used to improve the photocatalytic activity of GCN. It is notable that all halogen-
4 doped GCN exhibited superior efficiencies than bare GCN for atrazine removal, which can
5 distribute electrons nonuniformly and form hydrophobic sites on GCN (in addition to the
6 reduction of band gap energy and the increase of electrons migration space). Moreover, halogen
7 doping could either downshift VB (F and Cl) or upshift CB (I) to improve the oxidation
8 function of holes or hydroxyl radicals.¹⁶⁰ Mishra et al. synthesized B-, S-, and P-doped GCN
9 nanosheets by solid-state calcination of melamine followed by chemical exfoliation using HCl
10 for photocatalytic HER. The band gap energies of bare GCN, S-doped GCN, B-doped GCN,
11 and P-doped GCN were 2.70, 2.76, 2.69, and 2.77 eV, respectively. Not only B-doped GCN
12 exhibited the minimal band gap energy, but also boron doping improved light absorption.
13 Notably, B, S, and P doping considerably reduced the recombination rate of charge carriers
14 (compared to bare GCN) where the minimal PL intensity was observed for B-doped GCN.
15 Besides, B doping could reduce the charge transfer resistance (compared to bare GCN) as
16 confirmed by photo-electrochemical measurements. They were all responsible for the
17 significantly higher evolution rate of H₂ over B-doped GCN than bare GCN (18.2 μmol h⁻¹
18 against 6.6 μmol h⁻¹).

19 It is worth mentioning that S-doped GCN also showed higher photocatalytic H₂ generation
20 (12.3 μmol h⁻¹) than bare GCN.¹⁴⁹ However, the effect of S doping could be significantly
21 improved by using mesoporous GCN structures with nitrogen defects (vacancies).¹⁶¹ In
22 general, several types of defects could exist in GCN.¹⁶² Notably, defect engineering by the
23 creation of vacancy defect is considered a promising strategy to boost the photocatalytic
24 activity of GCN through different mechanisms.¹⁶³⁻¹⁶⁶ Vacancy defect could include nitrogen
25 vacancies, carbon vacancies, or dual vacancies.¹⁶⁴ It is notable that the electronic structure

1 could be effectively tuned by vacancy defect which could act as the special capture centers for
2 electrons (in parallel).¹⁶³ In addition, the lattice disorders initiated by the creation of vacancy
3 defect could adjust VB and CB towards each other, simultaneously, reducing the band gap
4 energy.¹⁶⁶ Numerous methods including multi-step calcination in a reducing environment
5 (H_2),^{164, 167} pretreatment in concentrated HNO_3 ,¹⁶⁸ and post treatment with salts (e.g. eutectic
6 mixture KCl and LiCl)¹⁶⁹ have been used for the generation of nitrogen vacancies in GCN. Liu
7 et al. synthesized S-doped and N defect co-modified mesoporous GCN for the photocatalytic
8 H_2 generation. A high specific surface area ($153.8 \text{ m}^2 \text{ g}^{-1}$) was achieved by the mesoporous
9 structure. In addition, S doping and N defect optimized the band structure, and reduced the
10 band gap energy. A surplus CB energy level was also created, improving migration and
11 separation of charge carriers. They all resulted in the extremely higher hydrogen generation
12 rate over S-doped and N defect co-modified mesoporous GCN than S-doped GCN ($4441 \mu\text{mol}$
13 $\text{g}^{-1} \text{ h}^{-1}$ against $91 \mu\text{mol g}^{-1} \text{ h}^{-1}$), clarifying the crucial role of nitrogen vacancy in improvement
14 of the photocatalytic activity of GCN. At the same time, it exhibited a superior activity for the
15 photocatalytic degradation of several organic pollutants including tetracycline hydrochloride,
16 crystal violet, MB, and RhB.¹⁶¹ He et al. synthesized C-doped and defects co-modified GCN
17 using one-pot polycondensation of urea with the addition of ethylenediaminetetraacetic acid
18 disodium salt (EDTA-2Na) as modifier for the photocatalytic degradation of bisphenol A
19 (BPA) under visible light irradiation. It has been concluded that co-modification of C doping
20 and defects could reduce the band gap energy and increase the mobility of charge carriers,
21 consequently, resulting in the remarkably higher kinetic constant of BPA photocatalytic
22 removal over C-doped and defects co-modified GCN (with 0.02 g EDTA-2Na) than pristine
23 GCN (0.0269 min^{-1} against 0.0031 min^{-1}).¹⁷⁰

24 Cation doping or metal doping is another interesting modification method of GCN to
25 promote its photocatalytic performance.^{19, 171, 172} Notably, due to the strong attraction between

1 negatively charged nitrogen atoms and metal cations, GCN is easily capable of capturing metal
2 cations. Moreover, cation doping could improve the photocatalytic activity of GCN.¹⁵⁹ Hence,
3 in addition to non-metal doping, cation-doped g-C₃N₄ photocatalysts have received extensive
4 attention due to their promising potential for environmental and energy-related applications.
5 For example, Oh et al. reported that Co- and Fe-doped GCN can significantly improve the
6 selective sulfathiazole degradation compared to pristine GCN.¹⁷³ Deng et al. found that Ni-
7 doped GCN can increase the photocatalytic hydrogen generation up to 155.71 $\mu\text{mol g}^{-1} \text{h}^{-1}$
8 (approximately 1.6 times higher than bare GCN). In addition, 10-fold increase in the constant
9 rate was obtained for the photodegradation of MO. Based on their findings, higher visible light
10 harvest, lower band gap energy, and suppressed recombination rate of charge carriers could
11 account for the superior efficiency of Ni-doped GCN than bare GCN.¹⁵³ Other kinds of metal
12 elements like Zn,¹⁷⁴ K,¹⁵⁶ Na,¹⁵⁶ Li,¹⁷⁵ V,¹⁷⁶ Cu,¹⁷⁶ and Ag¹⁷⁷ have been effectively used to
13 enhance the photocatalytic activity of bare GCN. It has been suggested that the cation size
14 effect could play an important role in the variation of structure and electronic/optical properties
15 ion-doped g-C₃N₄, and consequently the improved photocatalytic H₂ generation.¹⁵⁶ It should
16 be noted that delocalizability of unpaired electrons and their high concentration (initiated by
17 cation doping) could provide high conductivity, favoring the transport and separation of charge
18 carriers.¹⁷⁸ Lee et al. synthesized Mg-doped GCN, by the addition of Mg precursor
19 (Mg.Cl₂.6H₂O) during the thermal condensation of urea. Then, Mg-doped GCN was suspended
20 in ethylene glycol, with the addition of NaOH, while stirred for 2 h and heated at 160 °C for
21 12 h in an autoclave to synthesize nitrogen vacancy-rich GCN. The calculated band gap
22 energies of Mg-doped GCN, pristine GCN, and nitrogen vacancy-rich GCN were 2.89, 2.79,
23 and 2.73 eV, respectively. Unlike Mg-doped GCN, nitrogen vacancy-rich GCN reduced the
24 band gap energy, though slightly, clarifying that N defects could improve the photo absorption
25 ability of GCN. Moreover, nitrogen vacancy-rich GCN provided the lowest recombination rate

1 of charge carriers and the highest mobility of charge carriers. Due to its superior properties,
2 nitrogen vacancy-rich GCN exhibited the highest photocatalytic activity for the degradation of
3 oxytetracycline with the efficiency of ~92% compared to that of ~46% and ~81% (after 135
4 min) for bare GCN and Mg-doped GCN, respectively.¹⁶⁵

5 Despite of its several advantages, cation doping is generally accompanied with the
6 generation of some new energy levels in the electronic structure of GCN. It changes the
7 transition state and charge density of the initial structure. On the upside, it could increase the
8 visible light absorption and reduce the recombination speed of electron-hole pairs. However,
9 on the downside, the new defect energy levels could serve as recombination centers for charge
10 carriers, leading to the decrease of quantum efficiency.¹⁶²

11 Noteworthy, the metal-free character of GCN could be preserved by non-metal doping. In
12 addition, non-metal elements are easily capable of the formation of covalent bonds with other
13 compounds (because of high electronegativity and ionization energies) by receiving electrons
14 during processes.¹⁵⁹ Actually, nitrogen triangles with six lone-pair electrons are present in the
15 GCN framework, enabling its efficient use for nonmetal doping.¹⁶²

16 In summary, both cation and anion doping can be effectively used to modify the properties
17 of GCN while element doping followed by vacancy defect could be more promising. Notably,
18 further detailed research is needed to fully understand the effects of cation and anion doping,
19 and to determine the optimal approach for specific applications.

20

21 **4.3. Construction of heterostructures**

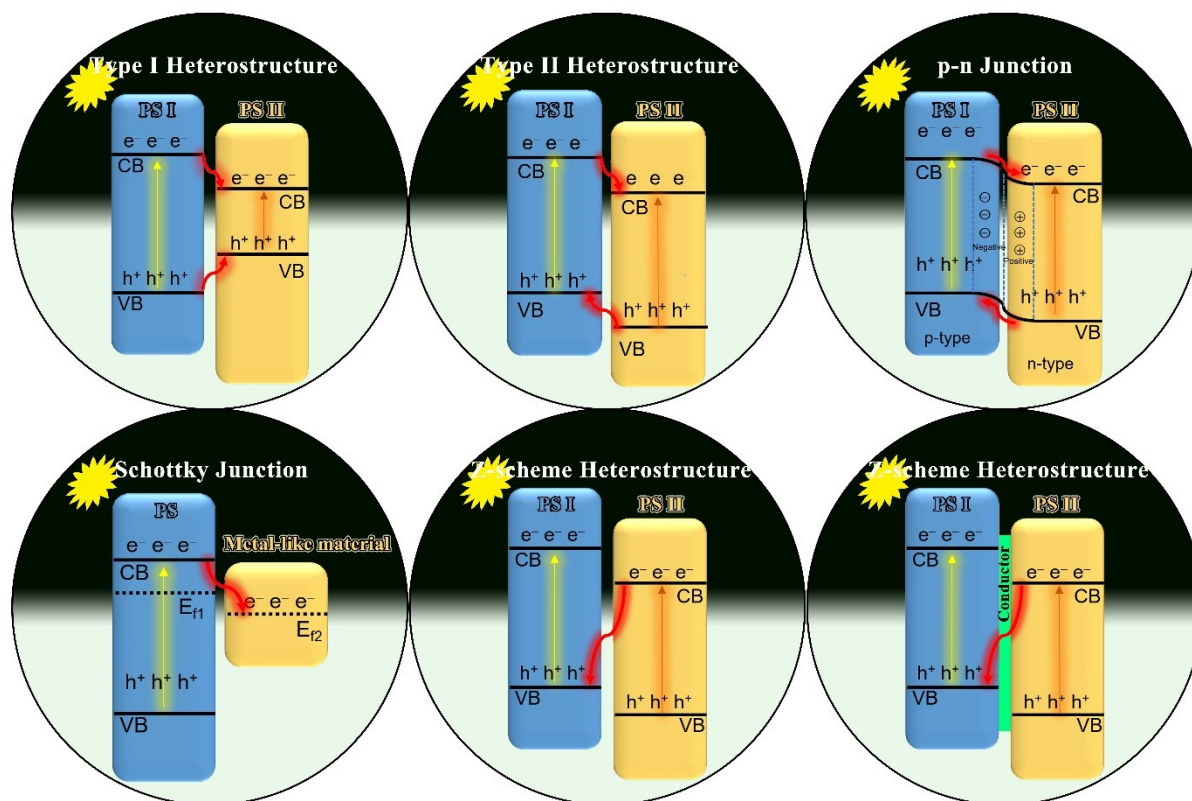
22 In general, coupling g-C₃N₄ with other semiconductors could improve the separation of charge
23 carriers and could increase its photocatalytic efficiency, consequently. Overall, based on the
24 relative energy band location between the two semiconductors, heterojunction structures are
25 classified into three different groups.¹⁷⁹ In Type-A heterostructures, the position of the CB of

1 photosystem I (PS I) is more negative than that of photosystem II (PS II). Besides, the position
2 of the VB of PS II is more positive than that of PS I. Under these circumstances, the CB of PS
3 II is positioned between the CB and the VB of PS I. By lowering the CB edge potential, the
4 photoexcited electrons of PS I migrate to the CB of PS II (due to its lower energy). In Type-B
5 heterostructures, the CB of PS II is more negative than that of PS I. Besides, the VB of PS I is
6 more positive than that of PS II. By raising the VB edge potential, the VB of PS II is located
7 between the VB and the CB of PS I (unlike Type-A). Thus, the photo-generated holes of PS I
8 migrate to the VB of PS II.^{179, 180} In Type-C heterostructures, the position of the CB of PS II is
9 more negative than that of PS I. In addition, the position of the VB of PS II is more positive
10 than that of PS I. Under these circumstances, although photo-generated electrons of PS I could
11 migrate for its VB to CB, neither holes nor electrons are transferred from PS I to PS II (due to
12 their higher energies). Noteworthy, most of the heterojunctions are categorized in Type-A
13 system.¹⁷⁹

14 Typically, Type-A and Type-B systems could be considered Type-II heterostructures
15 (staggered lineup).¹⁸¹ Type-II heterojunctions are the most desirable systems for photocatalytic
16 applications.^{180, 181} Type-C system is considered Type-I heterostructure (straddling
17 alignment).¹⁸¹ In order to drive a visible-light activated Type-II heterostructure, two
18 prerequisite conditions should be considered. Firstly, the coupled semiconductor should
19 possess strong visible light absorption. Secondly, the energy band alignment between the
20 semiconductors should satisfy the conditions of Type-II heterojunctions.¹⁸²

21 Recently, direct Z-scheme heterostructures, known as 3rd generation Z-scheme
22 heterostructures, have received much attention due to their potential impact on photocatalytic
23 performance compared with Type-II heterojunction photocatalysts.^{180, 183-186} Schematic
24 illustration of the typical separation routes of charge carriers are compared in Figure 11 (E_f
25 stands for Fermi level). It should be noted that the Schottky junction is constructed by a metal-

1 like material and a semiconductor. Due to the difference in work function, the photoexcited
 2 electrons can quickly transfer from the semiconductor to the metal-like material at their
 3 interface, resulting in alignment of the Fermi energy levels and reducing the recombination rate
 4 of charge carriers, remarkably.^{187, 188}



5
 6 Figure 11. Schematic representation of the **typical separation routes** of charge carrier.

7 Redrawn from Ref. ¹⁸⁷ Copyright 2019 Elsevier.

8
 9 In order to construct direct Z-scheme heterostructures, several key factors should be
 10 considered. Firstly, one of the semiconductors should possess a higher Fermi level, smaller
 11 work function, and higher CB and VB positions than the other one. Secondly, these
 12 heterojunctions are generally composed of a reduction photocatalyst and an oxidation
 13 photocatalyst.¹⁸⁵ Upon irradiation, in a typical Type-II heterojunction system, photo-generated
 14 holes transfer from the VB of PS II to that of PS I, while the photo-generated electrons move
 15 from the CB of PS I to that of PS II (Figure 11). When it comes to the direct Z-scheme

1 heterostructures, the photo-induced electrons in PS II combine with the photo-induced holes in
2 the VB of PS I. Thus, holes and electrons with higher oxidation and reduction potentials are
3 accumulated in the VB of PS II and CB of PS I, respectively (Figure 11). This spatial separation
4 of photo-generated e^-/h^+ pairs could improve redox capabilities.¹⁸⁰ Precipitation/deposition
5 processes and immobilization of one component on the surface of another one, are effective
6 methods to construct direct Z-scheme heterostructures. The effects of the formation of
7 heterostructures are briefly explained in the following section by providing some examples.

8

9 **4.3.1. Type I heterojunction**

10 Jing et al. synthesized SnS₂-GCN composites using different morphologies of SnS₂ (3D flower-
11 like, nanoparticles, and nanosheets) for photocatalytic H₂ generation under visible light
12 irradiation. Among those, SnS₂-GCN nanocomposites produced by SnS₂ nanoparticles with no
13 noble metal serving as co-catalyst showed the highest evolution rate of H₂, i.e. 6305.18 μmol
14 $\text{h}^{-1} \text{g}^{-1}$, while the photocatalytic generation rate of H₂ over pure GCN (containing 3 wt% Pt)
15 and SnS₂ nanoparticles (without Pt) was 106.82 and 371.27 $\mu\text{mol h}^{-1} \text{g}^{-1}$, respectively. The
16 interaction between GCN and SnS₂ has been accounted for the low recombination rate and
17 efficient charge transfer of photogenerated carriers in SnS₂-GCN nanocomposites, resulting in
18 their promising photocatalytic activity. Noteworthy, The VB and CB positions of GCN could
19 straddle those of SnS₂ nanoparticles when exposed to visible light, generating the Type I
20 heterojunction.¹⁸⁹ Dong et al. synthesized GCN-GCN isotype heterostructures by using a
21 composite precursor produced by mixing 6 g each of melamine and urea in water (30 mL)
22 followed by heating at 550 °C for 3 h. Finally, a Type I GCN-GCN metal-free heterojunction
23 was obtained where the photogenerated electrons could transport from the CB of the GCN
24 synthesized by urea to that synthesized by melamine, but the photogenerated holes could not
25 be transferred from one side to the other. It has been accounted for the improved separation of

1 charge carriers, playing a direct role in the higher photocatalytic NO removal in air over GCN-
2 GCN heterojunction than that over GCN synthesized by either melamine or urea (despite of
3 the remarkably lower specific surface area of GCN-GCN heterojunction, $32 \text{ m}^2 \text{ g}^{-1}$, than GCN
4 synthesized by urea ($91 \text{ m}^2 \text{ g}^{-1}$)).¹⁹⁰

5 In fact, charge carriers are usually gathered on one semiconductor in the Type I
6 heterostructures (upon light irradiation), whereas the spatial separation of charge carriers could
7 be effectively achieved by the Type II heterostructures, providing considerable improvement
8 of the photocatalytic activity.¹⁸⁷

9

10 **4.3.2. Type II heterojunction**

11 The distinctive staggered band structures between the two semiconductors in Type II
12 heterostructures may greatly improve the spatial separation of charge carriers. The internal
13 field could finally encourage the charge carrier's separation and migration. Notably,
14 appropriate band positions to separate charge carriers could be efficiently provided by Type II
15 alignment. Due to the potential advantages of Type II heterostructures in
16 photocatalysis, they are the most widely studied in the literature.¹⁸⁷ For instance, Rapit et al.
17 synthesized GCN-MoS₂ heterostructures for the photocatalytic removal of phenol and Cr(VI).
18 In detail, MoS₂-decorated GCN heterostructures (containing 0.5, 1, 3, and 10 wt% MoS₂) were
19 synthesized by mixing colloidal solutions of bulk GCN (suspended in isopropanol) and MoS₂
20 nanoparticles (suspended in a water/isopropanol solution) followed by sonication, centrifuge,
21 and calcination. Notably, MoS₂-decorated GCN heterojunctions showed higher photo
22 absorption ability in the visible light region than bare GCN. The optimal nanocomposite
23 (containing 1 wt% MoS₂ nanoparticles) exhibited superior efficiency than both MoS₂ and
24 pristine C₃N₄ in photocatalytic reduction/oxidation processes under simulated solar light
25 (through a Type II mechanism considering the scavenging experiments).¹⁹¹ Kalidasan et al.

1 synthesized Nd-doped $\text{WO}_3@\text{GCN}$ heterojunction (Type II) using a calcination method where
2 GCN (0.86 g) was ground with Nd-doped WO_3 (0.144 g) followed by calcination at 450 °C for
3 2 h. The synthesized heterojunction showed lower recombination rate of charge carriers than
4 both bare GCN and Nd-doped WO_3 , and considerably higher photocatalytic activity (84.51%)
5 for MB degradation than both pristine GCN and Nd-doped WO_3 (57.52% and 60.32%,
6 respectively) under visible light irradiation after 2 h. The results were attributed to the tight
7 contact at the assembly of the n-n heterostructure, facilitating the interfacial transfer of charge
8 carriers between GCN and Nd-doped WO_3 , and the appropriate light harvest.¹⁹²

9 Overall, Type II heterostructures could provide a pathway for the effective spatial
10 separation of charge carriers. Notably, the high apparent quantum efficiency of GCN- CeO_2
11 Type II heterojunction with rich oxygen vacancies, i.e. 17.1%, has been accounted for its higher
12 photocatalytic CO_2 reduction than pristine GCN and CeO_2 .¹⁹³ However, further improvement
13 of the photocatalytic activity could be obtained using p-n heterostructures, combined by n-type
14 and p-type semiconductors, providing an extra electrical field to facilitate the transfer of charge
15 carries and accelerate the separation of charge carriers compared to that obtained by Type II
16 heterostructures.¹⁸⁷

17

18 4.3.3. p-n heterojunction

19 Generally, the location of Fermi level of n-type and p-type semiconductors is close to their CB
20 and VB, respectively. It has been proposed that the holes could move from the p-type to the n-
21 type semiconductor prior to light irradiation. Notably, the diffusion of charge carriers persists
22 until an equilibrium condition is achieved in the Fermi level system. Finally, the effective
23 separation of charge carriers could occur as the result of the formation of an electrical field at
24 the interface of n-type and p-type semiconductors.¹⁸⁷ Ng et al. used Cu_2O and nitrogen-
25 deficient GCN (as p-type and n-type semiconductors, respectively) to synthesize $\text{Cu}_2\text{O}@g-$

1 C₃N_{4-x} nanocomposites for the photocatalytic generation of H₂. Based on their estimation, the
2 p-n junction could accelerate the directional transfer of charge carriers, due to the formation of
3 electric field, and yield the accommodation of holes and electrons at separated positions of
4 Cu₂O and nitrogen-deficient GCN, consecutively, improving the separation of charge carriers
5 and increasing their lifetime where the optimal nanocomposite (containing 3wt% Cu₂O)
6 exhibited 1.8 and 2.7 times higher photocatalytic H₂ generation rate than g-C₃N_{4-x} and Cu₂O,
7 respectively. Noteworthy, g-C₃N_{4-x} showed much higher efficiency than bare g-C₃N₄ (227.3
8 against 129.7 μmol g⁻¹ h⁻¹),¹⁹⁴ demonstrating the vital role of engineering defects in
9 improvement of the photocatalytic activity of GCN as discussed previously. Liu et al. also used
10 Cu₂O to synthesize Cu₂O-GCN heterostructures. GCN immobilized on the surface of Cu₂O
11 formed a tight heterojunction. The built-in electric field generated at the interface of
12 heterostructure has been accounted for the migration and effective separation of charge carriers.
13 As the result, construction of the p-n heterostructure has been accounted for the considerably
14 higher (3-7 times) quantum yield of Cu₂O-GCN than that obtained by the single
15 counterparts.¹⁹⁵ However, pure GCN could suffer from its low quantum yield, resulting in
16 lower photocatalytic activity compared to heterojunctions.¹⁹³ Overall, the 90.7% photocatalytic
17 degradation of tetracycline over the optimal Cu₂O-GCN p-n heterostructure was significantly
18 higher than that over single Cu₂O and GCN (39.5% and 47.8%, respectively).¹⁹⁵

19

20 **4.3.4. Schottky junction**

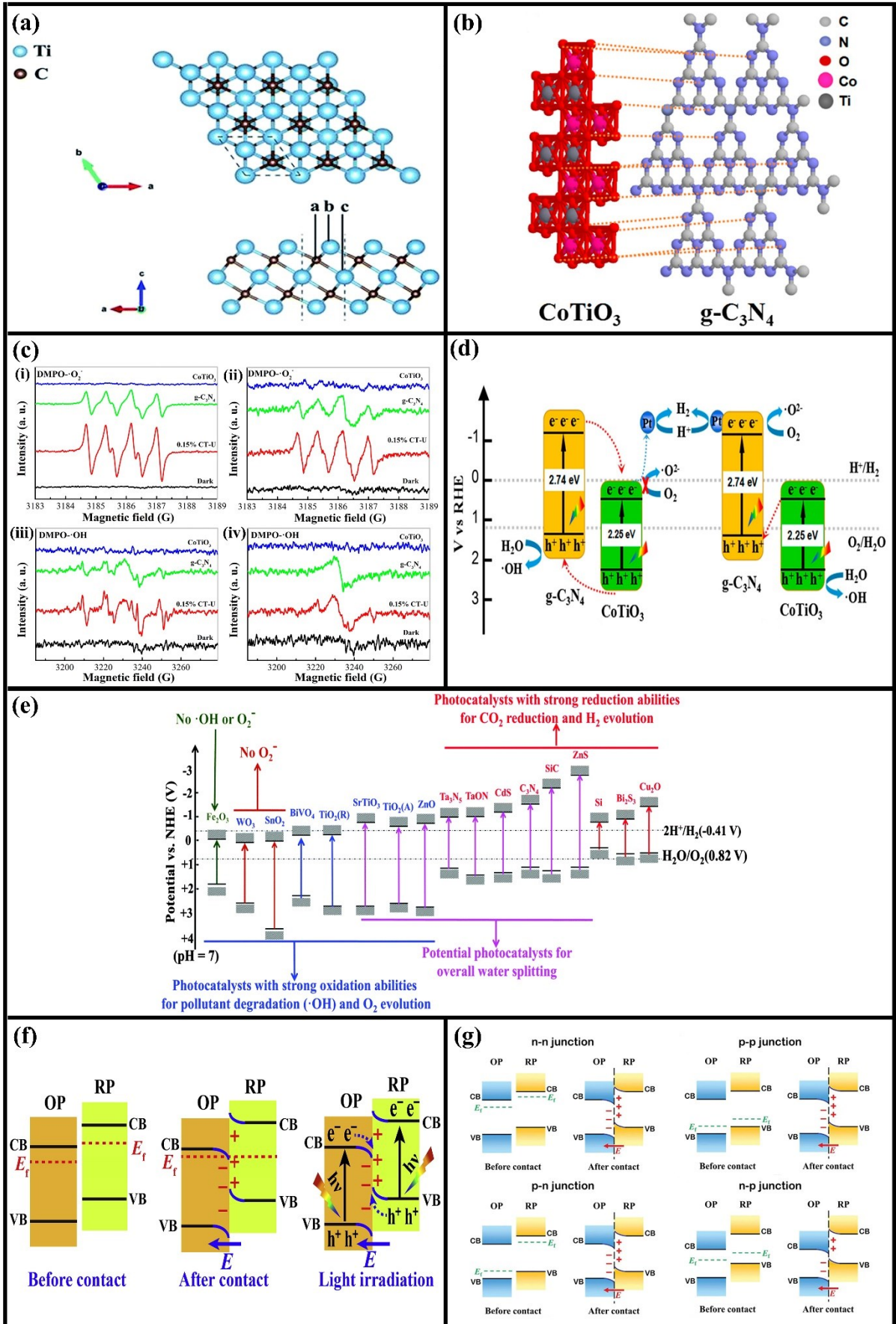
21 The Schottky junction, generally composed of a semiconducting material and a metal-like
22 material serving as co-catalyst, could be effectively used to construct a space-charge separation
23 area. The formation of such a junction could initiate the fast movement of electrons from one
24 component to the other at the interface of counterparts. Hence, the Fermi energy levels could
25 be aligned and the separation of charge carriers is greatly improved, increasing photocatalytic

1 activity.¹⁸⁷ Ni₂S,¹⁹⁶ MoB,¹⁹⁷ Co₂P,¹⁹⁸ and Ti₃C₂T_x (MXenes)¹⁹⁹⁻²⁰² are among numerous metal-
2 like materials used for the construction of Schottky junctions. Notably, MXenes are new 2D
3 nanostructure materials with the general formula of M_{n+1}X_nT_x (n=1-3) where M, X, and T_x are
4 early transition metals (Ta, Ti, Mo, Nb or V), nitrogen and/or carbon, and surface termination
5 groups (including -F, -O, and -OH), respectively.²⁰² It should be noted that the terminal
6 surface groups of MXene is capable of tuning its work function, enhancing the density of
7 electrons on the surface.¹⁹⁹ Among MXene materials, Ti₃C₂ (titanium carbide) is considered a
8 typical MXene has been extensively used as co-catalyst to promote photocatalytic activity due
9 to its promising properties including acceptable structural stability, excellent metal-like
10 conductivity, and high visible light harvest.¹⁹⁹⁻²⁰¹ Interestingly, Ti₃C₂ has been effectively used
11 as a co-catalyst to achieve the high AQY of 40.1% at 420 nm and an extremely high HER rate
12 of 14342 μmol h⁻¹ g⁻¹ by using CdS as the photocatalyst.²⁰³ Noteworthy, the AQY (which is
13 used to evaluate the photon conversion efficiency)²⁰² could be calculated using equation (8) in
14 a photocatalytic HER:²⁰²⁻²⁰⁵

$$15 \text{ AQY (\%)} = \frac{\text{Number of reacted electrons}}{\text{Number of incident photons}} \times 100 = \frac{\text{Number of evolved H}_2 \text{ molecules} \times 2}{\text{Number of incident photons}} \times 100 \quad (8)$$

16 In a case study, Sun et al. used Ti₃C₂T_x to synthesize GCN-Ti₃C₂T_x composites for efficient
17 photocatalytic hydrogen production under visible light irradiation. To prepare the composites,
18 bare GCN and black Ti₃C₂T_x powders were physically mixed at different mass ratios using a
19 mortar and pestle. Then, the composites were annealed at 200 °C in air or under nitrogen
20 protection to modify the surface termination groups of Ti₃C₂. Notably, the presence of Ti₃C₂T_x
21 in the composites with 30% Ti₃C₂ (mass ratio of Ti₃C₂T_x to GCN) increased the photo-
22 absorption ability and reduced the recombination rate of charge carriers compared to bare
23 GCN. Among different samples, the greatest photocatalytic H₂ generation obtained by the
24 composite containing 190% Ti₃C₂ (mass ratio of Ti₃C₂ to GCN) annealed in air with the H₂
25 production rate of 88 μmol h⁻¹ g⁻¹. To clarify the crucial effect of Ti₃C₂T_x, different amounts

1 of Pt, as another co-catalyst, from 1% to 5% (H_2PtCl_6 , mass ratio to GCN) were loaded on
2 GCN where the highest photocatalytic activity was achieved at 3% Pt with the H_2 production
3 rate of $56.2 \mu\text{mol h}^{-1} \text{g}^{-1}$ which was lower than that obtained by the addition of $\text{Ti}_3\text{C}_2\text{T}_x$.
4 Notably, the AQY reached to 1.27% for the optimal GCN- $\text{Ti}_3\text{C}_2\text{T}_x$ composite. Overall, the
5 addition of MXenes to GCN could increase HER rate by improving the separation of charge
6 carriers, due to the negative shift of Fermi level, resulted by the accumulation of
7 photogenerated electrons from GCN on Ti_3C_2 through the ohmic contact between metallic
8 MXenes and GCN semiconductor. The dependency of various surface termination groups on
9 the photocatalytic HER activity was estimated by density functional theory (DFT) calculations.
10 The top and side views of the Ti_3C_2 structure are illustrated in Figure 12(a). The termination
11 group sites on the Ti_3C_2 surface are represented by a (on top of the C atoms), b (over the surface
12 of Ti atoms), and c (over the mid-layer Ti atoms). The $-\text{O}$ and $-\text{F}$ termination groups occupy
13 the C sites, whereas the $-\text{OH}$ termination groups are ignored since they are only indirectly
14 related to the major step in the HER process. The $-\text{O}$ and $-\text{F}$ termination groups fill the C sites,
15 whereas the $-\text{OH}$ termination groups are ignored since they are not directly related to the
16 primary step in the photocatalytic HER. DFT calculations exhibited that Ti_3C_2 with oxygen
17 surface termination groups covered by H atoms (25%) provided the minimal free energy at
18 0.011 eV. Considering the experimental results, Ti_3C_2 with $-\text{O}$ surface termination groups (as
19 co-catalyst) provided the best photocatalytic activity confirmed by DFT calculations.²⁰²



1 Figure 12. (a) The top and side views of the Ti_3C_2 structure. Reproduced with permission
2 from Ref. ²⁰² Copyright 2018 The Royal Society of Chemistry
3 (<https://pubs.rsc.org/en/content/articlelanding/2018/TA/C8TA02706D>). (b) Schematic
4 interaction of tri-s-triazine GCN and CoTiO_3 via the generation of chemical bonds. Reprinted
5 with permission from Ref. ²⁰⁶ Copyright 2016 American Chemical Society. (c) ESR signals
6 of $\text{DMPO}\cdot\text{O}_2^-$ in methanol dispersion under (i) UV and (ii) visible light, and signals of
7 $\text{DMPO}\cdot\text{OH}$ in aqueous dispersion under (iii) UV and (iv) visible light. Reprinted with
8 permission from Ref. ²⁰⁶ Copyright 2016 American Chemical Society. (d) Comparing Type-II
9 and Z-scheme charge carrier transfer routes for $\text{CoTiO}_3/\text{GCN}$ hybrid photocatalysts.
10 Reprinted with permission from Ref. ²⁰⁶ Copyright 2016 American Chemical Society. (e)
11 Potential applications and band positions of some particular semiconductors (pH = 7 in
12 aqueous media). Reproduced with permission from Ref. ²⁰⁷ Copyright 2016 The Royal
13 Society of Chemistry (<https://pubs.rsc.org/en/content/articlelanding/2016/cs/c5cs00838g>). (f)
14 Schematic illustration of charge-transfer processes in S-scheme heterojunctions. Reprinted
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17 [licenses/elsevier-user-license](https://www.elsevier.com/about/policies/open-access-licenses/elsevier-user-license)). (g) Four probable types of S-scheme heterojunctions.
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19

20 **4.3.5. Z-scheme heterojunction**

21 As mentioned previously, construction of Type I, Type II, and p-n heterostructures could all be
22 considered effective approaches to reduce the recombination rate of charge carries. However,
23 on the downside, since reduction and oxidation reactions usually occur on the semiconducting
24 material with lower reduction and oxidation potentials, respectively, the redox sites of above
25 heterostructures are less active than their single counterparts, encouraging the application of Z-

1 scheme heterojunctions with their unique properties in photocatalytic systems where both
2 separation of charge carriers and high redox potentials could be achieved simultaneously.^{187, 210}

3 It is of high importance to carefully nominate PS I, PS II, and a redox electron mediator
4 for the construction of artificial photosynthesis systems (traditional liquid-phase Z-scheme
5 photocatalytic system as first generation Z-scheme heterostructures in which there is no direct
6 contact between semiconductors) by imitating natural photosynthesis. Noteworthy, the electron
7 acceptor-donor pair (A/D pair) could react with the photogenerated e^- and h^+ (in the CB of PS
8 I and in the VB of PS II), respectively. Unfortunately, artificial Z-scheme systems suffer from
9 several disadvantages including backward reactions, light absorption by the mediator that could
10 adversely affect the amount of light absorbed by the semiconductors, and being used only for
11 liquid phase reactions.^{208, 210}

12 Hence, all-solid-state Z-scheme photocatalytic systems (or indirect Z-scheme
13 photocatalytic systems) as second generation Z-scheme heterostructures (without the liquid
14 A/D pair and by using a solid conductive medium connecting semiconductors) were
15 proposed.^{208, 210} Various materials including noble metal particles (such as Au, Pt, and Ag)²¹¹⁻
16 ²¹³ and carbon spheres²¹⁴ have been used as electron mediators in such Z-scheme systems.
17 Although replacement of the shuttle ion pairs (used in traditional liquid-phase Z-scheme
18 photocatalytic systems) by a solid conductive medium (used in all-solid-state Z-scheme
19 photocatalytic systems) could broaden the application scope of Z-scheme systems, there are
20 still several drawbacks.²⁰⁹ Firstly, as shown in Figure 11, the reduction potential of electrons
21 in PS I is higher than that in PS II, meanwhile, the oxidation potential of holes in PS II is higher
22 than that in PS I. From thermodynamic viewpoint, it could facilitate the transfer of electrons
23 from the CB of Semiconductor 1 to the VB of Semiconductor 2 (rather than what is proposed
24 to be transferred from the CB of Semiconductor 2 to the VB of Semiconductor 1).^{208, 209}
25 Secondly, the proposed charge carrier route in all-solid-state Z-scheme systems could not be

1 easily achieved by taking the CdS-Au-TiO₂ junction as an example where the photogenerated
2 electrons in the CB of CdS and TiO₂ could migrate to Au (since it has the lowest Fermi level
3 in this system), while simultaneously, a Schottky barrier could be formed due to the band
4 bending at the interface of CdS/Au and Au/TiO₂, suppressing the electron flow from TiO₂ and
5 CdS to Au.^{208, 209} Thirdly, considering materials preparation, placing the conductor exactly in
6 the interface of semiconductors is not easy.^{208, 209} Fourthly, the conductor could compete with
7 semiconductors to absorb light, leading to low light harvest.^{208, 209}

8 Mediator-free Z-scheme systems, typically called direct Z-scheme photocatalytic systems,
9 could initiate the direct transfer of charge carriers between the closely contacted
10 semiconductors without a redox medium.²¹⁵ Hence, a higher photocatalytic activity is expected
11 due to the shortened transmission distance. Overall, direct Z-scheme heterostructures could
12 take unique advantages of the quick separation of charge carriers, cost-effectiveness (no need
13 to expensive redox mediators), and good redox potential, resulting in their wide applications in
14 either liquid or gas phase reactions (unlike the traditional liquid-phase Z-scheme photocatalytic
15 systems which could be only used for the liquid phase reactions).²¹⁰ Ye et al. synthesized
16 CoTiO₃-GCN hybrid photocatalysts using an in situ growth approach for the photocatalytic
17 HER from water splitting. They mixed urea (10 g) and CoTiO₃ powders which were ground in
18 an agate mortar where the weight ratios of CoTiO₃ to urea were 0.1, 0.15, 0.2, 0.3, 0.5, and 5
19 wt%. The composites, made of 2D GCN nanosheets and 1D CoTiO₃, were successfully
20 produced by heating the resultant mixture at 500 °C for 2 h and 520 °C for another 2 h, followed
21 by rinsing with water/ethanol and drying at 60 °C (in vacuum). Notably, before photocatalytic
22 experiments, the as-synthesized samples were loaded by Pt (3 wt%) as co-catalyst. The optimal
23 composite (weight ratio of CoTiO₃ to urea: 0.15%) yielded considerably higher photocatalytic
24 H₂ generation rate (858 μmol h⁻¹ g⁻¹) than pure GCN (422 μmol h⁻¹ g⁻¹) under artificial sunlight
25 irradiation. It also provided superior H₂ generation rate than pure GCN under visible light

1 irradiation ($22.2 \mu\text{mol h}^{-1} \text{g}^{-1}$ against $16.6 \mu\text{mol h}^{-1} \text{g}^{-1}$) which could be related to the close
2 interfacial contact and heterojunction interaction between CoTiO_3 and GCN. Figure 12(b)
3 shows the schematic interaction of tri-s-triazine GCN and CoTiO_3 via the generation of
4 chemical bonds. These interactions, which involve chemical bonds like Co-O-N or Ti-O-N ,
5 have been integrated using XPS data. Notably, the optimal hybrid photocatalyst yielded the
6 apparent quantum efficiencies of 3.23% and 38.4% at $420 \pm 20 \text{ nm}$ and 365 nm , respectively,
7 which were remarkably higher than that obtained by pure GCN (0.83% and 16.0% at 420 ± 20
8 nm at 365 nm , respectively). The generation of $\cdot\text{O}_2^-$ and $\cdot\text{OH}$ radicals over the optimal
9 nanocomposite and pure GCN irradiated by both UV and visible light were compared using
10 electron spin resonance (ESR) analysis. Overall, as shown in Figure 12(c), the intensity of $\cdot\text{O}_2^-$
11 and $\cdot\text{OH}$ signals for the hybrid photocatalyst was clearly higher than that for pure GCN,
12 clarifying that using the hybrid photocatalyst could improve the generation of $\cdot\text{O}_2^-$ and $\cdot\text{OH}$
13 radicals. The probable mechanisms for the separation of charge carriers over CoTiO_3 -GCN
14 hybrid photocatalyst are compared in Figure 12(d). By considering Type II heterojunction as
15 the mechanism of charge separation, the formation of $\cdot\text{O}_2^-$ and H_2 could be hindered (without
16 Pt co-catalyst). Furthermore, the oxidizability of holes will decrease by their transfer to the VB
17 of GCN. Hence, Type II heterojunction was unfavorable for the generation of active species
18 over CoTiO_3 -GCN hybrid photocatalyst. Overall, considering the results of photocatalytic
19 activity and ESR analysis, direct Z-scheme heterojunction has been considered the probable
20 charge carriers transfer route. Notably, compared with pure GCN, the recombination rate of
21 charge carriers increased by using the optimal composite as confirmed by the PL spectra.
22 Hence, the interfacial recombination of holes from the VB of GCN and electrons from the CB
23 of CoTiO_3 has been facilitated, boosting the effective separation of charge carriers and
24 increasing the redox potentials responsible for the improved H_2 generation.²⁰⁶ However, it
25 should be noted that the mechanism for the electron transfer in direct Z-scheme heterojunctions

1 (as the third generation of Z-scheme systems) is still unclear since the initial generations of Z-
2 scheme systems (i.e. traditional liquid-phase and all-solid-state) could be considered
3 problematic and not mature (direct Z-scheme heterojunctions could be considered from all-
4 solid-state Z-scheme systems).²⁰⁸

5

6 **4.3.6. S-scheme heterojunction**

7 Type-II and Z-scheme heterojunctions have been nominated as the probable charge carriers
8 transfer routes in hundreds of publications, while they both could suffer from several
9 shortcomings. For instance, thermodynamically, the transfer mechanism of charge carriers in
10 Type-II heterostructures reduces the redox potential, which is not favored in photocatalysis. In
11 addition, from dynamics point of view, the proposed pathway could be hindered by the
12 repulsion of similar charge carries (i.e. h^+/h^+ and e^-/e^-). Moreover, considering the energy
13 usage, some energy (VB or CB energy difference in each photocatalyst counterpart) is lost
14 without participating in photocatalytic reactions (Figure 11). Overall, the proposed charge
15 transfer route in Type-II heterostructures needs further clarification.²⁰⁹

16 To overcome the drawbacks of Type-II heterojunctions, while combining extensive
17 research in Z-scheme photocatalytic systems, S-scheme heterojunctions have been proposed
18 by Fu et al. as a new concept to clarify the probable route of charge carriers in
19 heterostructures.^{208,216} Considering the band positions, semiconductors are divided to reductive
20 and oxidative photocatalysts (RP and OP) with more negative CB positions and more positive
21 VB positions, respectively,²⁰⁷ as illustrated in Figure 12(e). S-scheme heterojunctions are
22 usually made of oxidative and reductive semiconductors with staggered band structures which
23 are comparable to Type-II heterojunctions, but the charge transfer pathways are totally
24 different. It has been proposed that not all the photogenerated charge carriers contribute to the
25 photocatalytic reactions. Notably, in S-scheme heterojunctions, idle photogenerated e^- and h^+

1 are recombined whereas the powerful ones are held in the CB of RP and the VB of OP,
2 respectively.²⁰⁸ The “S” figure-like transfer pathway of the photogenerated e^-/h^+ pairs at the
3 interface of S-scheme heterojunctions could provide the highest redox potential.²¹⁷ Schematic
4 illustration of charge transfer pathways in S-scheme heterojunctions is shown in Figure 12(f).
5 Such a charge transfer route could be driven by three major factors as follows:²⁰⁸

6 i. Internal electric field:

7 After close contact of semiconductors, due to the higher Fermi level of reductive
8 photocatalyst than oxidative photocatalyst (as shown in Figure 12(f)), electrons in the
9 RP are inclined to diffuse to the OP spontaneously, resulting in the formation of layers
10 depleted from electrons in the RP and accumulated by electrons in the OP.
11 Consequently, an internal electric field is formed at the junction, directed from the RP
12 to OP, accelerating the transfer of photogenerated e^- from the OP to RP.²⁰⁸

13 ii. Band bending:

14 After close contact of semiconductors to align Fermi energy, Fermi levels of the RP and
15 OP shift oppositely (downward and upward, respectively). Band bending encourages
16 the recombination of photogenerated e^- (in the CB of OP) and h^+ (in the VB of RP) at
17 the junction.²⁰⁸

18 iii. Coulombic attraction:

19 After close contact of semiconductors, the recombination of photogenerated e^- (in the
20 CB of OP) and h^+ (in the VB of RP) at the junction is likely due to the Coulombic
21 attraction between e^- and h^+ .²⁰⁸

22 Overall, charge transfer pathways in S-scheme photocatalytic systems are basically
23 different from those for Type-II and Z-scheme systems, as evidenced both theoretically and
24 experimentally.²⁰⁸ Similar to other types of heterojunctions, S-scheme systems also have some
25 limitations. Firstly, they are mainly for powder photocatalysts, while not applying to

1 photoelectrochemical processes using an external circuit. Secondly, the photocatalyst
2 counterparts are mainly n-type semiconductors and should have considerable Fermi level
3 differences and appropriate band structures.²⁰⁸ However, from a more general viewpoint, S-
4 scheme heterostructures could be constructed by a p-type or an n-type semiconductor once a
5 comprehensive requirement is satisfied. The specific requirement is that the Fermi level and
6 CB of OP should be lower than those of RP at the same time (OP or RP could be either p-type
7 or n-type semiconductors).²⁰⁹ All such probabilities for the construction of S-scheme
8 heterostructures are shown in Figure 12(g). Recently, ex/in situ irradiated XPS, EPR, and AFM
9 (with potential mode) characterization techniques have been effectively used to confirm the
10 formation of S-scheme photocatalytic systems.²⁰⁸

11 Kisch et al. suggested that only quantum yields (not rate constants) could be compared in
12 either heterogeneous or homogeneous photochemical reactions.²¹⁸ The adjustment of structure
13 and/or morphology is considered an effective approach to improve the quantum efficiency.¹⁴⁵
14 The effects of wavelength and modification of GCN structure (by either element doping or
15 production of heterostructures) on AQY of the photocatalytic H₂ production or pollutant
16 degradation are shown in Table 4. For instance, S doping increased the AQY from 4.06% for
17 bare GCN to 10.2% for S-doped GCN. Meanwhile, the heterojunction composed of B-doped
18 GCN nanodots and S-doped GCN nanosheets increased the AQY to 22.7%,¹⁵⁰ clarifying the
19 crucial role of both element doping and fabrication of heterojunctions in improving AQY. It is
20 also obvious that the AQY of H₂ production/pollutant removal is highly dependent on
21 wavelength, due to the relationship between photo-absorption ability and wavelength.¹⁴⁵

22
23
24

Table 4. Effects of wavelength and modification of GCN structure on AQY.

Modification method of g-C ₃ N ₄	AQY (%)		Wavelength (nm)	Light intensity	Reference
	Bare g-C ₃ N ₄	Modified g-C ₃ N ₄			
S doping	4.06	10.2	420	1.6 mW cm ⁻²	150
Heterojunction (B-doped GCN/S-doped GCN)	4.06	22.7	420	1.6 mW cm ⁻²	150
P doping	-	4.32	420	4.1 mW cm ⁻²	145
P doping	-	3.58	450	3.5 mW cm ⁻²	145
P doping	-	1.28	500	4.3 mW cm ⁻²	145
Heterojunction (Ag/g-C ₃ N ₄)	10.7×10^{-4}	14.3×10^{-4}	-	-	219
S doping	~0.03*	~0.39*	420	-	151
Heterojunction (V ₂ O ₅ /N-deficient g-C ₃ N ₄)	-	6.5	420	100 mW cm ⁻²	220
Heterojunction (V ₂ O ₅ /N-deficient g-C ₃ N ₄)	-	5.1	460	100 mW cm ⁻²	220
Heterojunction (V ₂ O ₅ /N-deficient g-C ₃ N ₄)	-	0.5	500	100 mW cm ⁻²	220
Heterojunction (g-C ₃ N ₄ /SnO ₂)	4.44×10^{-4}	5.73×10^{-4}	-	-	221
Heterojunction (g-C ₃ N ₄ /Ti ₃ C ₂)	0.141	3.654	-	-	222

Heterojunction (Co ₃ O ₄ /g-C ₃ N ₄)	0	0.053	420	-	223
P doping	0.66	1.78	420	-	224
Heterojunction (ReS ₂ /g-C ₃ N ₄)	-	4.10	365	31.81 mW cm ⁻²	225
Heterojunction (ReS ₂ /g-C ₃ N ₄)	-	2.82	420	22.30 mW cm ⁻²	225
Heterojunction (ReS ₂ /g-C ₃ N ₄)	-	1.17	450	19.10 mW cm ⁻²	225
Heterojunction (ReS ₂ /g-C ₃ N ₄)	-	0.29	485	15.77 mW cm ⁻²	225
Heterojunction (ReS ₂ /g-C ₃ N ₄)	-	0	535	6.92 mW cm ⁻²	225

1 * : Extracted from the publication by Plot Digitizer

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1 4.4. Applying a bias potential for pollutant degradation

2 The recombination of charge carriers is a vital factor affecting the efficiency of semiconducting
3 materials in PC applications. It is well understood that the modification of PC systems by the
4 addition of electrochemical systems can significantly increase the efficiency of PC systems by
5 controlling the redox reactions occurring separately at the anode and cathode.^{226, 227}
6 Noteworthy, applying a bias potential could provide a driving force to accelerate the transfer
7 of photogenerated electrons from the photoanode to the cathode through the external circuit,
8 promoting the separation efficiency of the photogenerated electron-hole pairs in
9 photoelectrocatalytic (PEC) processes. Therefore, the aggregated effects of electrocatalytic
10 (EC) and PC processes could result in the higher efficiency of PEC process than both
11 conventional EC and PC processes.²²⁸ Furthermore, PEC process can not only increase the
12 efficiency, but also enhance the decomposition capacity of recalcitrant pollutants.²²⁹ Overall,
13 PEC system is known as an efficient, promising, sustainable, and environmentally friendly
14 process (without secondary pollution) for decomposition of organic pollutants in aqueous
15 solution.^{226, 230} Hence, PEC process has found applications in water splitting²³¹ and solar
16 cells²³² (in addition to photocatalysis). The photocatalytic and PEC activity of some composites
17 synthesized by the addition of GCN for decomposition of various organic pollutants is
18 compared in Table 5. The results demonstrate that the PEC process yielded a higher efficiency
19 than PC process in all cases.

20 In a typical PEC system, the semiconducting material is immobilized on a conductive
21 substrate which serves as the photoanode.²²⁹ Various photoanodes have been produced using
22 TiO_2 ,²³³ ZnFe_2O_4 ,²³⁴ ZnO ,²³⁵ BiVO_4 ,²³⁶ MoSe_2 ,²³⁷ PbO_2 ,²³⁸ and graphite²³⁹ for H_2 generation
23 or the degradation of organic pollutants. Considering the synergetic effect of PC and EC, using
24 photoanodes with appropriate band gap energies and suitable charge-carrier conductivities is
25 desired to achieve high PEC performances.^{240, 241} Besides, using visible-light activated

- 1 photoanodes with high stability is desired.²⁴² Owing to its unique properties, g-C₃N₄ has
- 2 recently gained wide attention for the production of the photoanodes to promote the
- 3 degradation efficiency of organic pollutants.

1 Table 5. Comparison of photocatalytic and photoelectrocatalytic activity of the GCN composites for the degradation of organic pollutants.

Photoelectrode	Pollutant (concentration)	Source of light	Photoelectrocatalytic activity	Photocatalytic activity	Reference
PbO ₂ /GCN	Reactive brilliant blue KN-R (60 mg/L)	175 W Xe lamp	~91% (120 min)	~18% (120 min)	243
CQDs/GCN	Methylene blue (5 mg/L)	500 W Xe lamp	97.2% (180 min)	57.02% (180 min)	228
GCN/TiO ₂	Phenol (5 mg/L)	500 W Xe lamp	~99% (150 min)	~26% (150 min)	244
GCN/TiO ₂	Tetracycline (10 mg/L)	300 W Xe lamp	95% (60 min)	65% (60 min)	229
Fe ₂ O ₃ /GCN	Clofibric acid (30 mg/L)	NA	99.7% (300 min)	40.6% (300 min)	230
GCN/BiVO ₄	Diclofenac sodium (10 mg/L)	300 W Xe lamp	31.2% (120 min)	7.0% (120 min)	240
POM/GCN	Rhodamine B (NA)	NA	88.2% (120 min)	23.2% (120 min)	241
TiO ₂ /GCN/CQDs	1,4-dioxane (10 mg/L)	LED lamps	78.0% (360 min)	33.2% (360 min)	245
TiO ₂ /GCN	1,4-dioxane (10 mg/L)	LED lamps	~49% (360 min)	~23% (360 min)	245
Co ₃ O ₄ /GCN	Clofibric acid (30 mg/L)	300 W Xe lamp	95.3% (360 min)	~6% (360 min)	242

2 Note: POM (polyoxometalate)

3

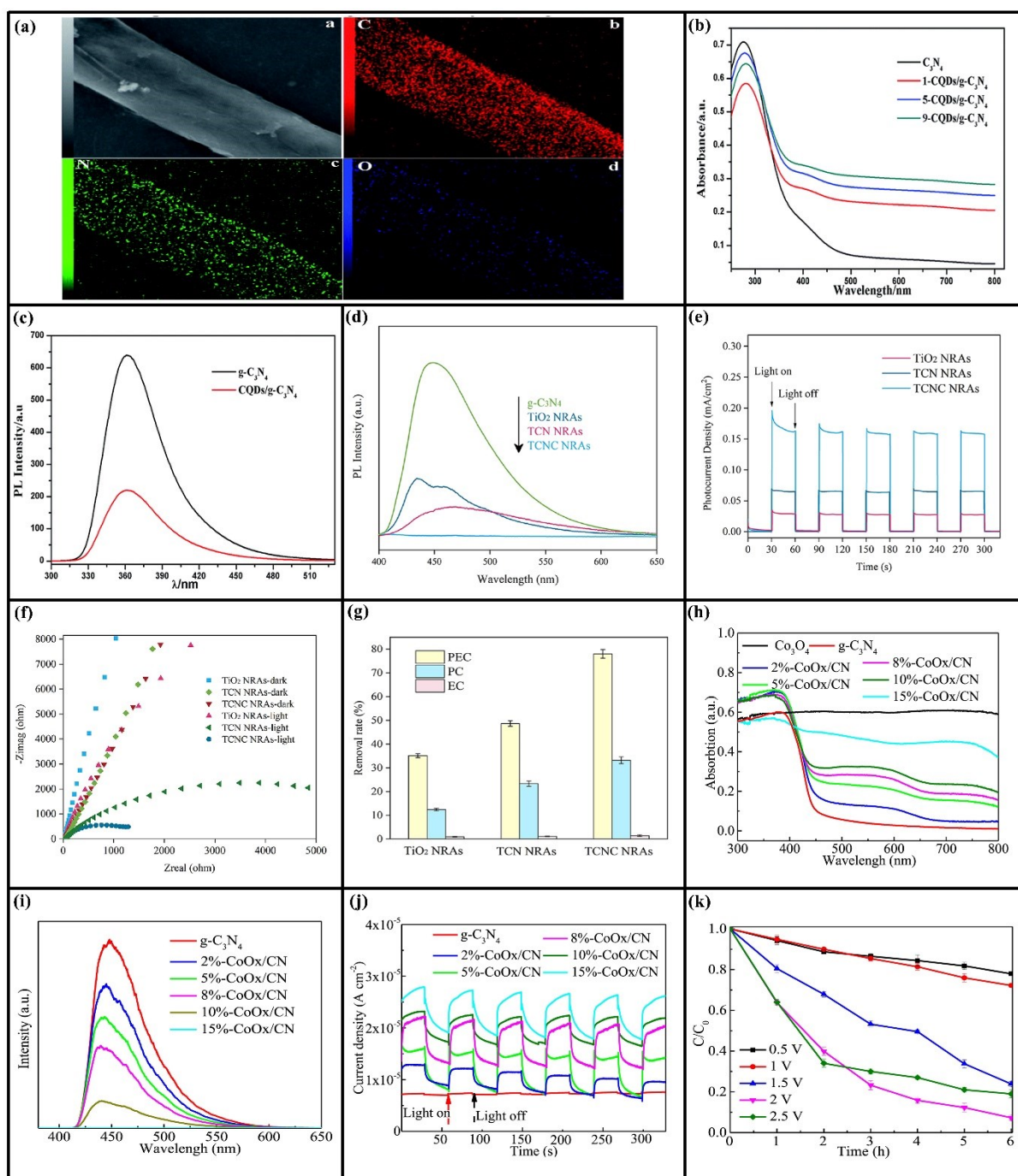
1 Li et al. evaluated the effect of the addition of GCN to PbO₂ on the PC and PEC degradation
2 of reactive brilliant blue KN-R. The PC and PEC degradation percentages of brilliant blue KN-
3 R over the PbO₂ photoanode were 8% and 63%, respectively, while the PC and PEC
4 degradation percentages of brilliant blue KN-R over the GCN-modified PbO₂ photoanode were
5 18% and 91%, respectively. Firstly, PEC process exhibited much higher efficiency than PC
6 process over both photoanodes. Secondly, the introduction of GCN into PbO₂ film has
7 significantly improved the degradation efficiency of KN-R in both PC and PEC processes that
8 was attributed to the change of morphology, hydrophobicity, and surface hydrophilicity of the
9 electrodes which improve the photoelectrochemical active sites. Furthermore, GCN improved
10 the carrier density and oxygen evolution potential of PbO₂ film.²⁴³ Wang et al. fabricated GCN-
11 wrapped TiO₂ nanotube arrays for phenol degradation under simulated sunlight. The as-
12 prepared photoanode showed promising PEC activity which was related to the improved light
13 harvesting and separation efficiency of charge carriers originated from the introduction of g-
14 C₃N₄ to TiO₂. Compared to pure TiO₂, the optimum amount of GCN resulted in a 4-fold
15 increase in the photocurrent density.²⁴⁴

16 In addition to heterojunction structures, the decoration of GCN with conductive materials
17 is another approach which can significantly improve PEC activity. Recently, carbon quantum
18 dots (CQDs) have received considerable attention due to good biocompatibility, unique
19 electrical conductivity, reliable stability, low toxicity, excellent up-converted
20 photoluminescence feature, chemical inertness, suitable catalytic and optical properties, and
21 easy functionalization.²⁴⁶⁻²⁴⁸ Specifically, due to its unique electron-accepting and donating
22 properties, it is considered a potential photosensitizer for the production of semiconductor
23 nanocomposites with efficient capability of separation of charge carriers.²⁴⁶ Zhang et al.
24 fabricated g-C₃N₄ nanosheets decorated with CQDs for PEC degradation of MB. SEM and
25 EDX mapping images of the optimal CQDs/GCN nanocomposite are shown in Figure 13(a).

1 Notably, the uniform distribution of O and C elements clarify well decoration of GCN with
2 CQDs. Photo-absorption ability and PL spectra of CQDs/GCN nanocomposites are illustrated
3 in Figures 13(b) and 13(c), respectively. As evident, all CQDs/GCN nanocomposites exhibited
4 significantly superior visible-light harvest than GCN. Furthermore, the PL intensity of the
5 optimal CQDs/GCN nanocomposite is remarkably lower than that of GCN. Moreover, the
6 optimal CQDs/GCN nanocomposite exhibited strong fluorescence up-conversion properties.
7 The modified properties resulted in the significantly higher PEC activity of the optimal
8 CQDs/GCN nanocomposite (97.2%) than GCN (11.1%) for the decomposition of MB.
9 Noteworthy, the PC activity of the nanocomposite was 57.02% that signifies the crucial role of
10 the applied bias potential on degradation of organic pollutants.²²⁸

11 In another research, Su et al. used GCN to improve the PEC degradation of 1,4-dioxane
12 over TiO₂ nanorod arrays. Further improvement was carried out by the decoration of the
13 TiO₂/GCN film with CQDs. Notably, TiO₂/GCN/CQDs (TCNC NRAs) showed narrower band
14 gap energy and longer lifetime of photogenerated charge carries than both TiO₂ nanorod arrays
15 (TiO₂ NRAs) and TiO₂/GCN nanorod arrays (TCN NRAs). PL spectra of GCN, TiO₂ NRAs,
16 TCN NRAs, and TCNC NRAs are compared in Figure 13(d), clarifying the low recombination
17 rate of charge carries for TCNC NRAs. In addition, compared to TiO₂ NRAs and TCN NRAs,
18 TCNC NRAs exhibited much higher photo-current responses (Figure 13(e)). The
19 corresponding EIS Nyquist plots are shown in Figure 13(f). The arc radius of TiO₂ NRAs was
20 larger than that of TCN NRAs and TCNC NRAs under both dark condition and visible-light
21 irradiation, indicating faster electrode reactions and easier transfer of electrons on the
22 fabricated photoanodes in the order of TCNC>TCN>TiO₂ nanorod arrays. PEC, EC, and PC
23 activity of these photoanodes for the degradation of 1,4-dioxane is compared in Figure 13(g).
24 As evident, PEC process was more efficient than both EC and PC processes over all
25 photoanodes. Besides, the introduction of GCN to TiO₂ followed by decoration with CQDs

1 remarkably improved the degradation efficiency of 1,4-dioxane in both PEC and PC processes.
2 These findings clearly show the crucial role of recombination rate of charge carriers in the
3 higher efficiency of PEC than PC process, and in the improved degradation efficiency of 1,4-
4 dioxane over $\text{TiO}_2/\text{GCN}/\text{CQDs}$ nanocomposites.²⁴⁵ Zhang et al. constructed $\text{Co}_3\text{O}_4/\text{GCN}$
5 photoanodes for the degradation of clofibric acid. Photo-adsorption ability, PL spectra, and
6 photo-current response of bare GCN and $\text{Co}_3\text{O}_4/\text{GCN}$ nanocomposites with different amounts
7 of Co_3O_4 from 2% to 15%, relative to melamine used for synthesis of GCN, are shown in
8 Figures 13(h), 13(i) and 13(j), respectively. Overall, the introduction of Co_3O_4 increased both
9 photo-adsorption ability and photo-current response, and decreased PL intensity compared
10 with bare GCN. Noteworthy, both higher photo-adsorption ability and lower recombination
11 rate of charge carriers were responsible for the superior PEC activity of the optimal
12 nanocomposite (15%- $\text{Co}_3\text{O}_4/\text{GCN}$) than that of GCN for the degradation of clofibric acid. The
13 effect of applying different bias potentials on the PEC activity of 15%- $\text{Co}_3\text{O}_4/\text{GCN}$
14 nanocomposite is shown in Figure 13(k). It is evident that the more the bias voltage, the higher
15 the degradation efficiency was obtained which was related to the reduced recombination rate
16 of charge carriers at increased bias potentials (up to 2 V). Further increase of the voltage from
17 2 V to 2.5 V did not significantly affect the PEC degradation of clofibric acid due to the
18 saturated efficiency of charge separation at 2 V.²⁴²



1
2 Figure 13. (a) SEM and EDX mapping images of the CQDs/GCN nanocomposite.

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7 CQDs/GCN nanocomposites and bare GCN. Reproduced with permission from Ref. ²²⁸

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4 (<https://creativecommons.org/licenses/by-nc/3.0/>). (c) PL spectra of the CQDs/GCN

5 nanocomposites and bare GCN. Reproduced with permission from Ref. ²²⁸ Copyright 2017

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7 (<https://pubs.rsc.org/en/content/articlelanding/2017/ra/c7ra11205j>). This article is licensed

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9 (<https://creativecommons.org/licenses/by-nc/3.0/>). (d) PL spectra of TiO₂ NRAs, TCN NRAs,

10 and TCNC NRAs. Reprinted with permission from Ref. ²⁴⁵ Copyright 2020 Elsevier. (e)

11 Photo-current response of TiO₂ NRAs, TCN NRAs, and TCNC NRAs. Reprinted with

12 permission from Ref. ²⁴⁵ Copyright 2020 Elsevier. (f) EIS Nyquist plots of TiO₂ NRAs, TCN

13 NRAs, and TCNC NRAs. Reprinted with permission from Ref. ²⁴⁵ Copyright 2020 Elsevier.

14 (g) PC, EC, and PEC activity of TiO₂ NRAs, TCN NRAs, and TCNC NRAs. Reprinted with

15 permission from Ref. ²⁴⁵ Copyright 2020 Elsevier. (h) Photo-adsorption ability of bare GCN

16 and Co₃O₄/GCN nanocomposites. Reprinted with permission from Ref. ²⁴² Copyright 2021

17 Elsevier. (i) PL spectra of bare GCN and Co₃O₄/GCN nanocomposites. Reprinted with

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19 and Co₃O₄/GCN nanocomposites. Reprinted with permission from Ref. ²⁴² Copyright 2021

20 Elsevier. (k) Effects of different bias potentials on the PEC activity of 15%-Co₃O₄/GCN

21 nanocomposite. Reprinted with permission from Ref. ²⁴² Copyright 2021 Elsevier.

22

23 **5. Conclusions and future perspectives**

24 With unique physical, chemical and electronic properties, g-C₃N₄ is a promising semiconductor
25 for photocatalytic applications especially under visible light irradiation. Among different

1 methods used for the synthesis of g-C₃N₄, thermal polymerization is considered a promising
2 approach owing to its advantages such as simplicity, low cost, and absence of impurities.
3 Notably, ammonium salts could be effectively used for the simultaneous synthesis of GCN
4 nanosheets and element doping. Although immobilization might reduce the specific surface
5 area, the immobilized catalysts can be effectively used as the photoanode in
6 photoelectrocatalysis with its superior efficiency than either electrocatalysis or photocatalysis
7 for pollutant remediation due to the significantly reduced recombination rate of charge carriers.
8 In addition, some major drawbacks of photocatalysis such as the agglomeration of the
9 photocatalysts and the issues relating to their recovery could be resolved.

10 Further research should explore industrial and cost-effective methods to synthesize and
11 deposit GCN composites, which is essential to bridge the gap between laboratory experiments
12 and commercial-scale applications of photocatalysis. Great research opportunities also exist in
13 using ammonium salts for the large-scale production of elements-doped GNC nanosheets,
14 followed by defect engineering and decoration with economical and promising metal oxides to
15 fabricate S-scheme heterostructures, which can be used for efficient photocatalytic H₂
16 generation and degradation of emerging and persistent organic pollutants in different
17 environments.

18

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21

22 **Conflict of Interest**

23 The authors declare no conflict of interest.

24

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